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# Ultrafast control of the optical transition in type-II colloidal quantum wells

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## Abstract

Manipulating the optical transition in semiconductors at ultrashort timescales is of both fundamental interest and central importance for emerging photonic applications. Traditionally, this manipulation is realized by electrostatic gating via Stark effects or bandgap renormalizations. Here, we report an ultrafast and all-optical route to engineer the indirect transition in core-crown colloidal quantum wells (CQWs), CdSe/CdTe, with type-II band alignment. Following the intense laser pulse excitation, the indirect band transition energy exhibits a pronounced blueshift-redshift crossover on the picosecond timescale, stemming from the formation and dissipation of transient E-filed that forms upon photoexcitation to compensate for the driving force provided by the band offsets. Both the energy shift and dynamics of the transient E-filed can be modulated optically by tuning the laser pulse excitation fluence. Our finding demonstrates a strong analogy between the type-II heterojunction and a p-n junction with respect to carrier equilibrium processes, which holds promise to facilitate the integration of CQWs within optical switching networks.

**Key Words:** Colloidal quantum wells; Type-II heterojunction; Ultrafast spectroscopy; Optical modulation; CdSe/CdTe;

## Introduction

The investigation and manipulation of optical transition in semiconductors are important topics in condensed matter physics<sup>1-2</sup> and underlie future photonic applications, including optical switching and modulating<sup>3</sup>. To date, successfully demonstrated manipulation schemes<sup>4-10</sup> mostly rely on the Stark effect (i.e., the band is tilted to oppose the external E-filed) or the bandgap renormalization due to electrostatic doping (i.e., the free carriers modify the electrostatic energies to shift the conduction and valence band toward each other). However, these approaches cannot go beyond the limitations of electronics and strongly hinder the practical applications in integrated circuits due to the sacrifice in operation speed and the restriction in the modulation depth. To tackle this issue, all-optical manipulation is one of the most fascinating directions that offers high-speed processing and versatile modulation capability<sup>11-13</sup>. Although pioneer works have demonstrated temporally manipulating various optical properties (e.g., refractive index<sup>11</sup>, carrier temperature<sup>12</sup>, and electro-absorption<sup>13</sup>) in colloidal nanocrystals, ultrafast control of optical transitions in colloidal nanocrystals is still a daunting challenge.

Recent advances in colloidal quantum wells (CQWs), particularly those CQWs with fully type-II band alignment (i.e., conduction band minimum and valence band maximum are residing in two different domains), offer a unique platform to investigate the all-optical manipulation<sup>14-20</sup>. In type II CQWs (e.g., CdSe/CdTe CQWs), when photo-generated excitons/carriers in either domain of the heterojunction diffuse to the interface, ultrafast charge transfer (CT), which occurs in the femtosecond timescale, generates electrons and holes separated in different domains. This gives rise to the n-doping and p-doping for CdSe and CdTe, respectively<sup>14,21</sup>. Benefitting from the strong Coulomb interactions due to quantum confinement and reduced dielectric screening in these few-monolayer thick CQWs, electrons/holes across the interface create a considerable transient E-field to oppose the further charge separation where the driving force is provided by the band offsets<sup>12,22</sup>. Therefore, in close analogy to the carrier equilibrium process in p-n junction or the band bending in organic photovoltaics<sup>23</sup>, built-up and dissipation of the photo-generated transient E-field in type II CQWs provide a good opportunity to examine the all-optical control of band separation in semiconductors. However, despite several studies addressing the dynamics of excitons and carrier separation in type II CQWs<sup>14-16,18,24,25</sup>, the role of photo-generated transient E-field on optical transition remains unexplored experimentally.

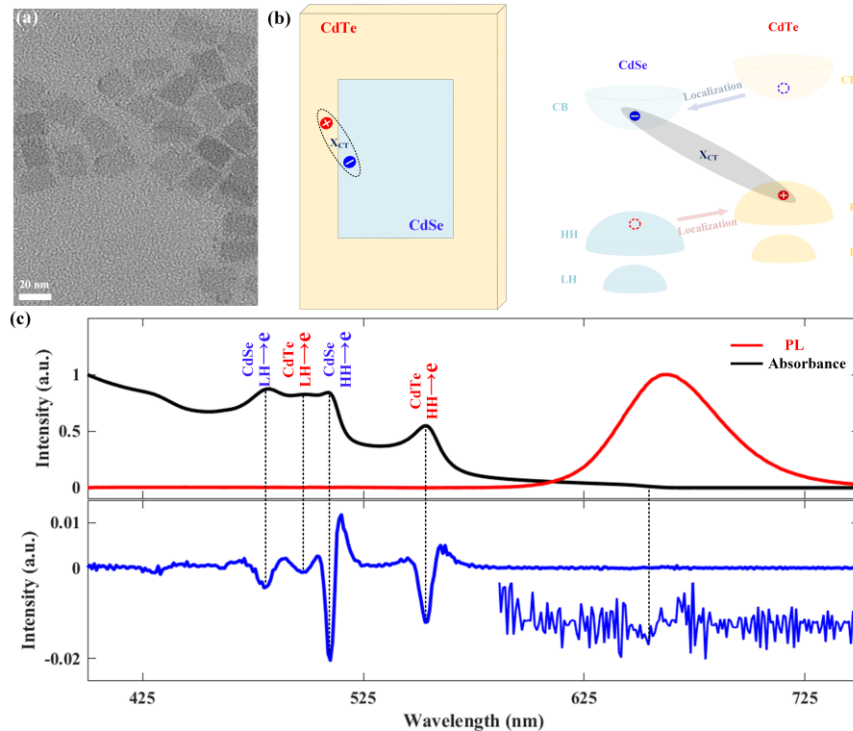
Herein, we demonstrate the crossover control of the optical transition at nano-heterojunctions in CdSe/CdTe CQWs on the ultrafast timescale by using time-resolved (TR) absorbance spectroscopy. In particular, after the intense femtosecond optical excitation (e.g., 2164  $\mu\text{J}/\text{cm}^2$  at 410 nm), we observe a pronounced blueshift ( $\sim 72$  meV) of the indirect band transition energy at the early time stage (with a lifetime of  $\sim 4.51$  ps), followed by a faster redshift (with a lifetime of  $\sim 1.21$  ps) which almost recovers the indirect band transition energy to the static value. We attribute these two different energy shifts to the formation and decay process of the photo-generated transient electric field (*E*-field). Meanwhile, we further demonstrate that the manipulation of optical transition can be photo-controlled by varying the excitation fluence and the blue-shifted energy linearly increases with the cube root of exciton fluence, supporting again that the control of optical transition is caused by the transient *E*-field which is generated by the separation of opposite charges across the interface. Our results may stimulate the future development of all-optical solid-state devices.

## Results and Discussion

4-monolayer ( $\sim 1.4$  nm in the vertical confinement direction) CdSe/CdTe core-crown CQWs, as shown in **Fig. 1a**, are synthesized using a previous method with slight modifications (see details in **Methods**)<sup>19</sup>. These CQWs display rectangular shapes and crystallize in a zinc blende structure with CdTe crown grown on the facets of the CdSe core<sup>20</sup>. The band alignment of the core-crown CQWs is illustrated in **Fig. 1b**, showing that the conduction band minimum resides in CdSe and the valence band maximum (i.e., the highest energy level for heavy holes) locates in CdTe. Consequently, the CdSe/CdTe CQW forms a type II heterojunction with a conduction band and a valence band offset of 0.35 eV and 0.54 eV, respectively<sup>26</sup>. In the single-particle picture, this type II heterostructure, driven by the band offsets, will lead to separated electrons and holes in two different domains upon photon excitation, forming an analogous p-n junction at the interface (i.e., CdSe is n-doping and CdTe is p-doping)<sup>23,27,28</sup>. The absorption and photoluminescence (PL) spectra of as-prepared CdSe/CdTe CQWs dispersed in hexane are presented in **Fig. 1c**. Typically, in type II heterostructures (quantum dots<sup>29,30</sup>, nanorods<sup>31,32</sup> or transition-metal dichalcogenides<sup>12,27,33</sup>), the absorption spectrum will exhibit both intra- and inter-domain transition features. As shown in the black curve, four intra-domain excitonic states centered at 480, 512, 501, and 560 nm can be well-resolved, which correspond to the transitions associated with light holes (LH) in CdSe, heavy holes (HH) in CdSe, LH in CdTe and HH in CdTe, respectively. Since the strength of the inter-domain transition is very weak (i.e., the interfacial CT transition is indirect), the absorption spectrum lacks a pronounced feature related to the CT state. To quantify the static energy position of the inter-domain transition, we analyze the second derivative of the absorption spectrum (the blue curve) and infer that the CT transition is located at  $\sim 646$  nm (see the zoom-in part).

While for the emission spectrum (the red curve in **Fig. 1c**), the dominated radiative recombination from CT excitons (peaked at 662 nm) and the fully quenched emission from each domain (expected at  $\sim 512$  nm for CdSe and  $\sim 556$  nm for CdTe) have indicated a very effective charge separation. Recently, several reports<sup>14-16,18,24</sup> have investigated CT dynamics at the domain interface of CdSe/CdTe CQWs, although the role of carrier interaction/excitonic effect on the charge separation process and how carriers across different domains can conserve the momentum are still under debate, a consensus from all reports is that interfacial CT occurs with near-unity efficiency and on an ultrafast sub-picosecond time scale (e.g., 0.48 ps in ref. 14 and 70 fs in ref. 16). We have also noticed that the CT emission shows a large Stokes shift ( $\sim 16$  nm) and broad line-width (full width

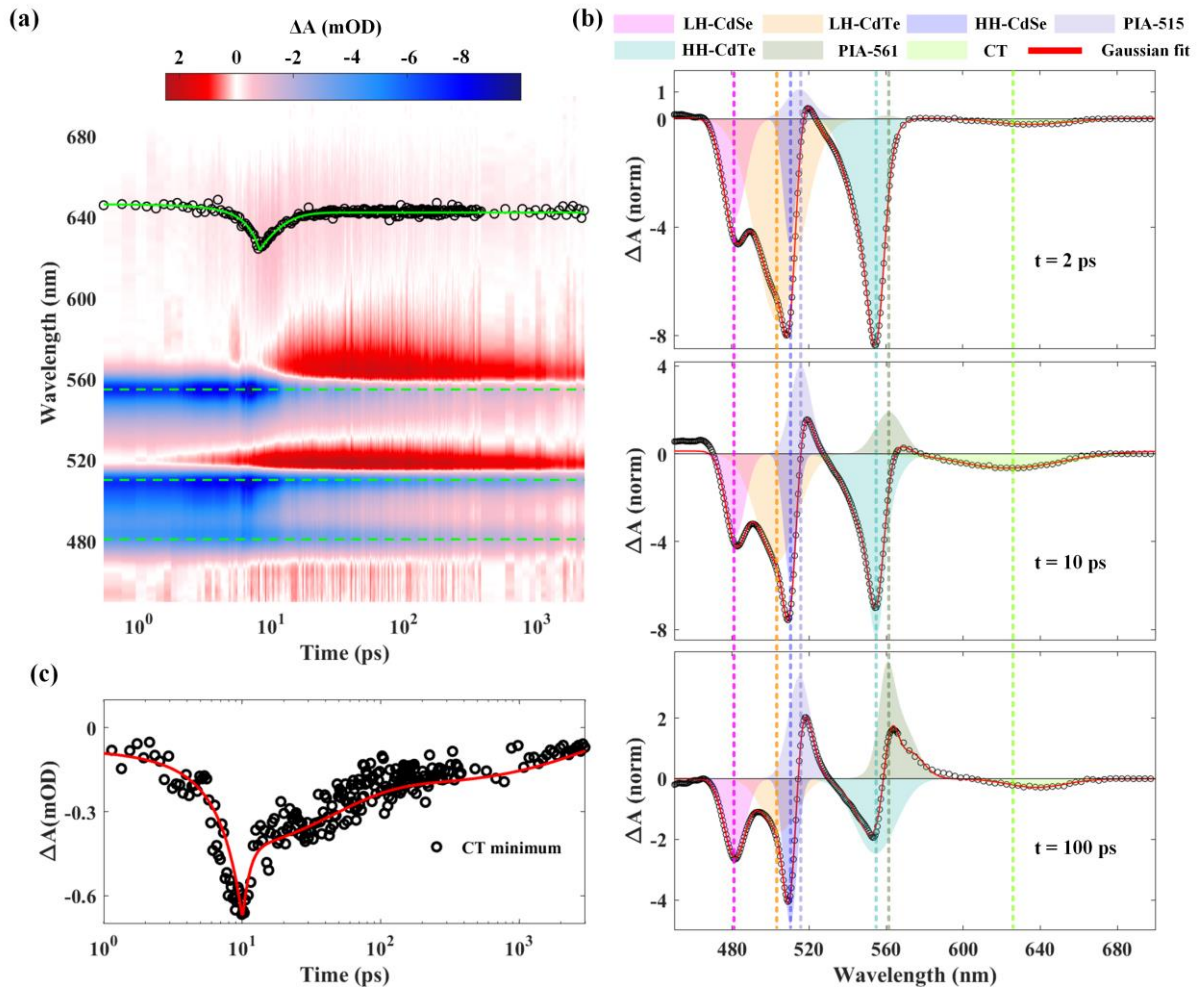
at half maximum:  $\sim 54$  nm), which are in stark contrast to the nearly vanishing Stokes shift (2-4 nm) and the narrow emission (8-12 nm) in single domain CQWs with atomic monolayer thickness confinement<sup>13,17</sup>. This is another fingerprint of fully type II band alignment since the spatially separated electrons/holes are more polarizable and can heavily interact with longitudinal optical phonons through Frolich interactions<sup>34</sup>. In addition, we observe an averaged inter-domain exciton lifetime of  $\sim 54.6$  ns via the time-resolved PL measurement (see **Fig. S1**), an order of magnitude longer than that in intra-domain excitons, again supporting the effective charge separation since the spatially indirect nature leads to a reduced electron/hole wavefunction overlapping<sup>25,33</sup>.



**Figure 1. Band alignment and structure of CdSe/CdTe type II CQWs.** (a) Transmission electron microscopy (TEM) images of the synthesized 4-ML core-crown CdSe/CdTe CQWs with an overall size of  $\sim 22$  nm  $\times$  36 nm. (b) Schematic illustrations of the core-crown structure (the left panel) and the predicted band alignment (the right panel) with a fully type II heterojunction. Photo-generated carriers/excitons in both CdSe and CdTe domains will be localized to the interface with electrons in CdSe and holes in CdTe. These domain-separated electrons and holes have a chance to form the charge transfer excitons ( $X_{cr}$ ). CB: conduction band, HH: heavy-hole, LH: light-hole. (c) Absorption (black) and photoluminescence (red) spectra of CdSe/CdTe CQWs. Absorption second derivative (blue) spectra of the type II CQWs are also presented to help determine the possible transition energies. Inset of bottom panel: a zoom-in of the second derivative curve to show the charge-transfer (CT) state.

To probe whether excitation-generated carriers can affect optical transitions in type II CQWs, we have measured transient absorption (TA) spectra using the pump-probe technique. We start with a weak excitation at 410 nm ( $0.6 \mu\text{J}/\text{cm}^2$ , which corresponds to the number of generated electron-hole pairs  $\langle N_{e-h} \rangle$ :  $\sim 0.01$ , see **Methods**) to avoid any multi-carrier/-exciton effects. As presented in **Fig. S2a**, four intra-domain and one inter-domain bleaching bands can be seen in the TA spectrum at the initial delay time (i.e., 2 ps) with energy locations agreeing well with that extracted from the steady-state absorption spectrum (**Fig. 1c**). Meanwhile, based on the bleaching dynamics in **Fig. S2b**, we can extract the carrier transfer lifetime (i.e., the separation time of electrons and holes in two different domains) as  $\sim 0.47$  ps, agreeing well with previous reports<sup>14-16,18,24</sup>. Next, we focus on the multi-carrier scenario in CdTe/CdSe CQWs with high excitation densities. **Fig. 2a** shows the transient absorption signals in false color after an excitation fluence of  $2164 \mu\text{J}/\text{cm}^2$  ( $\langle N_{e-h} \rangle$ :  $\sim 417$ ) and the data reveal two distinct behaviors for inter- and intra-domain transitions after performing the multiple Gaussian fittings<sup>5</sup>. For CdSe-LH/-HH and CdTe-HH, the transition energy remains almost constant in the measured time window (as indicated by the green dashed lines, also see the extracted peak positions in **Fig. S3**). In contrast, the CT transition band blueshifts gradually on the first  $\sim 10$  ps after photoexcitation. At later times (i.e., after  $\sim 10$  ps), the blueshift trend diminishes and turns into a redshift, which eventually restores the CT transition energy to the static value (i.e., 646 nm). This observation can be appreciated better by looking at three different spectral cuts (**Fig. 2b**) with the delay time right at (the blue curve), before (the green curve), and after (the red curve) the shift turning point. For a quantitative analysis of the CT transition energy, we have depicted the energetic peak

position for each spectral cut in **Fig. 2a** (black circles), which shows the maximum energy shift for CT transition crossover is  $\sim 23$  nm ( $\sim 72$  meV). More importantly, the transition crossover dynamics can be extracted by exponentially fitting this peak energy trace of CT state, which reveals an averaged blueshift lifetime of  $\sim 3.85$  ps and an averaged redshift lifetime of  $\sim 17.23$  ps.

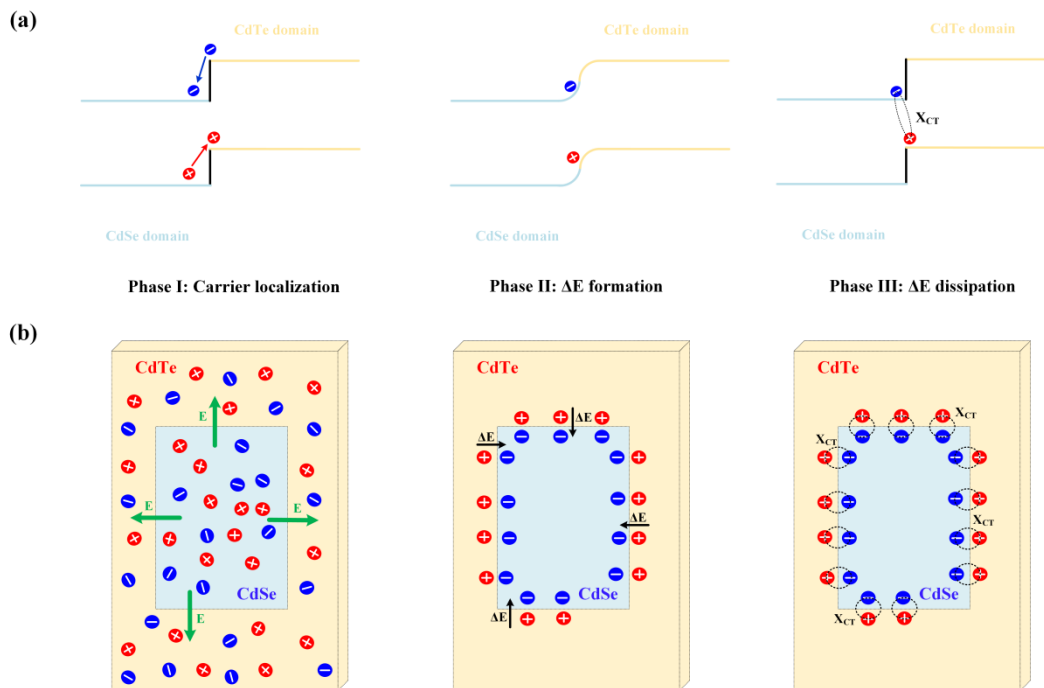


**Figure 2.** Time-resolved absorbance spectroscopy of CdSe/CdTe CQWs with the excitation fluence of  $2164 \mu\text{J}/\text{cm}^2$  ( $\langle N_{e-h} \rangle: \sim 417$ ). (a) The pseudocolor plot at room temperature. The horizontal axis, vertical axis, and color scale represent the pump-probe time delay, probe photon energy, and transient absorbance signal, respectively. Three green dashed lines mark the peak wavelength of the direct transitions in CdSe and CdTe domains. The black dots indicate the wavelength position of the bleaching peak of the CT state extracted by Gaussian fittings, and the green curve is an exponential fitting. (b) Transient absorbance spectra at three representative pump-probe delays, which are fitted by multiple Gaussian functions to determine the peak positions. The vertical dotted lines are located at the peak positions of corresponding transition bands when  $t = 10$  ps. (c) Decay kinetics probed at the minimum wavelength of the CT state. Black dots: measurements. Red curve: exponential fitting.

Due to the lack of phonon bottleneck (hot carrier relaxes within 100 fs)<sup>8,35</sup>, the relatively slow carrier recombination rate ( $> 50$  ns), and the ultrafast charge transfer rate ( $< 1$  ps) in CdSe/CdTe CQWs, we can further check the CT transition crossover dynamics by analyzing bleaching signals at the minimum CT transition wavelength (i.e., 623 nm). As shown in **Fig. 2c**, the bleaching dynamics at 623 nm can be well described by a single-exponential built-up process with a lifetime of  $\sim 4.51$  ps and a bi-exponential recovery process with time constants of  $\sim 1.21$  ps and  $\sim 3.47$  ns, respectively. Comparing the time trace of peak energies presented in **Fig. 2a** with the bleaching dynamics shown in **Fig. 2c**, we can conclude that (i) the blueshift-redshift crossover (at  $\sim 9.54$  ps in **Fig. 2a**) occurs at the same time as the maximum bleaching signal (at  $\sim 10.12$  ps in **Fig. 2c**) and (ii) the built-up lifetime of bleaching dynamics ( $\sim 4.51$  ps in **Fig. 2c**) is consistent with the blue-shifting time constant ( $\sim 3.85$  ps in **Fig. 2a**). These coincidences have verified the determining process of CT energy shifting dynamics and implied that the observed CT transition crossover is highly related to the photo-induced carrier/exciton dynamics. It should be noted that the recovery process extracted from bleaching dynamics at 623 nm deviates from red-shifting kinetics in the time trace of peak energies, which can be understood as follows: (i) The slow

recovery process with a lifetime of  $\sim 3.47$  ns is close to the fast decay channel resolved in the time-resolved PL measurement (see **Fig. S1**), which can be related to the recombination of inter-domain excitons ( $X_{CT}$ ). Please note that the time trace of peak locations reflects the change of CT transition energy and has a negligible connection with exciton recombinations; as a result, the slow recovery process in **Fig. 2c** is absent in the redshifting process presented in **Fig. 2a**; (ii) the redshift process with a lifetime of  $\sim 1.21$  ps obtained in **Fig. 2c** is more reliable since the accuracy of dynamics evolution in **Fig. 2a** is limited due to the ignorance of time-dependent carrier population, a fitting error to determine the energy peak and the spectra overlapping with the photo-induced absorptions<sup>36</sup>.

After the observation of an ultrafast modulation of the CT transition, we now attempt to figure out the possible mechanism responsible for the blueshift-redshift crossover of CT transitions. Since we do not observe a universal transition shift (i.e., no change of intra-domain transitions) at high excitation density, the measured transition crossover of the CT state with a considerable energy shifting should be related to the process at the heterojunction. Also, such transition crossover of CT state cannot be due to a combination of those general effects because they would also cause the change of intra-domain transitions, such as enhanced Coulomb screening<sup>37</sup>, pump-induced lattice heating<sup>38</sup>, bandgap renormalization<sup>7</sup> and Burstein–Moss effect<sup>39</sup>. In addition, we have noticed that the reversible carrier trapping by the structure defects can involve in such transition crossover. Despite the carrier trapping occurring in the ps range, the reversed process (i.e., the carrier releasing) is relatively slow and typically with a lifetime of several  $\mu\text{s}$ <sup>40</sup>, which is inconsistent with our observation. More interestingly, accompanying the CT transition crossover, the built-up of the intra-domain bleaching signals becomes much slower at high excitation density (see **Fig. 2a** and **Fig. S4**), which is in striking contrast with the fast built-up process at low excitation fluence (**Fig. S2**). Since two already-known channels that contribute to the built-up of intra-domain bleaching signal (i.e., optical excitation and charge transfer) occur on the picosecond timescale, we suggest that at high excitation density, there should be a new dynamic process to against and slow down the ultrafast charge transfer. Hence carrier separation becomes less efficient or incomplete, leading to the observation of pure domain emission at the high excitation fluence (an emission peak from the CdTe domain emerges in addition to the main type-II emission at  $\sim 9000\mu\text{J}/\text{cm}^2$ )<sup>24</sup>.



**Figure 3. Schematic illustration of the optical transition modulation processes in CdSe/CdTe CQWs with ‘Energetic’ (a) and ‘Spatial’ (b) representations.** In Phase I, after photoexcitation, when the generated carriers/excitons diffuse to the interface, driven by the band offsets, the electrons and holes will be separately localized in CdSe and CdTe domains, respectively. In Phase II, the electron/hole charge separation across the interface leads to a transient  $E$ -field build-up, which compensates for the band offsets and opposes the charge separation process. While in Phase III, when carriers approach equilibrium, these separated charges are bounded into the CT excitons. Therefore,  $\Delta E$  disappears due to these neutral excitons. Note that, we do not depict the domain excitons in these schematics for clarity.

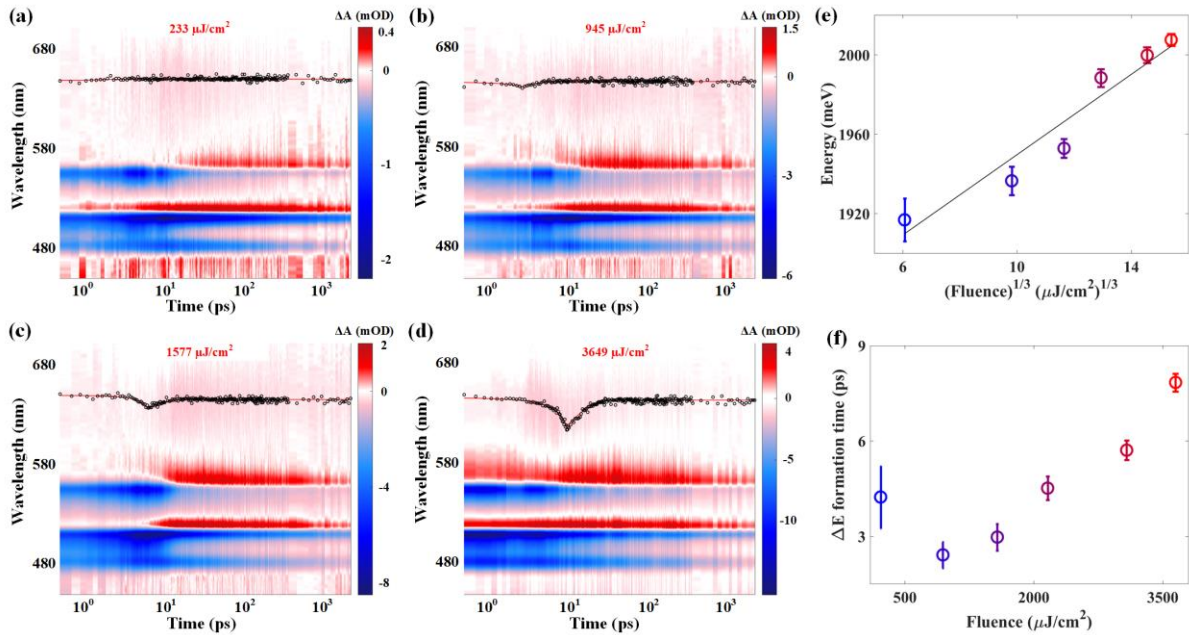
In light of the above discussion, combined with the type II band alignment in CdSe/CdTe CQWs, the most reasonable hypothesis to interpret the observed CT transition crossover is the formation and dissipation of a

transient  $E$ -field ( $\Delta\varepsilon$ ) across the interface. As schematically illustrated in **Fig. 3**, the type-II heterostructure in CdSe/CdTe CQWs functions similarly to that of a p–n junction in photovoltaics or photodectors<sup>23</sup>: At Phase I (the carrier transfer process), in the presence of a built-in potential resulting from type II band offsets, photo-generated hot electrons and holes in one domain are efficiently transferred to the CdSe domain and CdTe domain within one picosecond ( $\sim 0.47$  ps) and, eventually, leading to n-doping (p-doping) for CdSe (CdTe). While at Phase II (the carrier accumulation process at the crown-core interface), hot carriers diffuse to the hetero-interface and cool down through interaction with longitudinal optical phonon modes. At this stage, the accumulation of opposite charges across the interface generates a transient  $\Delta\varepsilon$  that counteracts the further charge separation driven by the built-in  $E$ -field. In other words, the transient electric field will prevent the continuous accumulation of the same charges at one side of the interface which is very similar to repulsive interactions among same charges<sup>24,25</sup>. Consequently, the conduction band minimum and valence band maximum near the interface starts to bend and subsequently cancel out the initial built-in potential, leading to the blueshift of CT transition. Please note the timescale of the blueshifting process in our works ( $\sim 4$  ps) is consistent with the crown-to-core carrier diffusion lifetime ( $\sim 5$  ps) extracted by Raj<sup>16</sup> and Li<sup>15</sup>. It is also worth mentioning that the presence of  $\Delta\varepsilon$  at molecular donor/acceptor interfaces<sup>41</sup> or the p-n/Schottky junctions<sup>22</sup> has already been observed and probed via femtosecond optical spectroscopy. Later, at Phase III (the CT exciton formation process), carrier-phonon interaction further localizes the electron–hole pair at the crown-core interface, forming the ‘cold’ CT excitons with a timescale of  $\sim 1.21$  ps<sup>16,23</sup>. As a result,  $\Delta\varepsilon$  disappears due to the formation of neutral excitons, and therefore, the band alignment restores to the initial scenario. Due to the relatively slow recombination dynamics (see **Fig. S1**, lifetime:  $>50$  ns) in CdSe/CdTe CQWs and the relatively small energy shifting value due to the Burstein-Moss effect (several meV with our excitation condition), we argue that the contribution of reduced carrier population to the redshift of CT transition should be negligible. Furthermore, in Phase I, these photo-generated electron–hole pairs are more likely to diffuse to the interface in the form of domain excitons and then dissociate into the unbounded carriers due to the presence of a built-in potential since it can avoid the strict parallel momentum conservation, as argued by Mishchenko *et al.*<sup>42</sup>. Here, we have simplified these processes and depicted the unbounded electron-hole pairs only instead of domain excitons for clarity in these schematics. Please note that the above explanation is also supported by the absence of CT transition crossover when only pumping the lowest energy CT exciton (see the data in **Figure S5-6**). Without carrier diffusion due to the 590 nm excitation, the transient  $\Delta\varepsilon$  (Phase II in **Figure 3** of our manuscript) cannot be built up, leading to the time-independent CT transition energy.

To further prove that the observed CT transition crossover is indeed due to the photo-induced transient  $\Delta\varepsilon$ , we show that the CT transition characteristics can be gradually modulated by varying the excitation fluence to change the photo-generated carrier density so as the strength and dynamics of  $\Delta\varepsilon$ . **Fig. 4a-d** and **Fig. S7** show the corresponding two-dimensional plots of TA spectra for the pump fluences ranging from a relatively low value, with which we can barely see the modulation ( $233 \mu\text{J}/\text{cm}^2$ , i.e.  $\langle N_{e-h} \rangle$ :  $\sim 40$ ), to the highest value that can be reached in our lab ( $3649 \mu\text{J}/\text{cm}^2$ , i.e.  $\langle N_{e-h} \rangle$ :  $\sim 700$ ). We see that there is no energy shifting of intra-domain transitions for different carrier densities, consistent with our discussion above (also see the extracted peak values in **Fig. S8-9**). Similar to our previous analysis in **Fig. 2c**, we have extracted the maximum energy position of CT transition, the formation, and the dissipation lifetime of  $\Delta\varepsilon$  versus excitation fluence. As shown in **Fig. 4e**, the peak energy of CT transition can be well-fitted by the equation:  $E_p = E_0 + a \cdot F^{1/3}$ , where  $E_0$  is the static value and  $a$  is a fitting parameter. This cube root dependence on excitation fluence is in good harmony with the behavior of  $\Delta\varepsilon$  caused band bending predicted theoretically<sup>43</sup>. When approaching the carrier equilibrium (i.e., at the CT transition crossover turning point), the electron density  $n_e$  and hole density  $n_h$  accumulated at the interface can be related to fluence  $F$  as:  $n_e \cdot n_h \propto F$ . These charged carriers generate an approximately triangular potential well at the heterojunction with the transient  $\Delta\varepsilon$  proportional to  $n_e$  (i.e.,  $F^{1/2}$ ) and the ground state quantization energy in such a triangular potential well is shifted proportionally with  $\Delta\varepsilon^{2/3}$ . Thus the band bending energy is expected to increase proportionally with the cubic root of  $F$ .

**Fig. 4f** summarizes the measured formation times of  $\Delta\varepsilon$ , i.e., the built-up lifetime of bleaching at the minimum CT transition wavelength, as a function of the excitation fluence. We observe a slower formation process at a higher excitation density (except the first data point in which there is a substantial fitting error due to the subtle CT transition shift with low excitation fluence). Indeed, because of the strong carrier density dependence of equilibrium conditions, we do expect a prolonged carrier diffusion and accumulation time for a higher pump fluence; and the result in **Fig. 4f** is a necessary consequence of the slower dynamic balance at the heterojunction with more intensive excitation. While the dissipation lifetime of  $\Delta\varepsilon$  exhibits an irregular trend with increasing

fluence (see **Fig. S10**), which at this stage is still a puzzle, further studies are required to investigate how electron-hole pairs collapse to the lowest CT state. With low fluences, there is an up-and-down fluctuation for the dissipation time, reflecting a competition among multiple involved processes to form the ‘cold’ CT excitons. With a further increase of fluence larger than  $2164 \mu\text{J}/\text{cm}^2$ , an accelerated dissipation process of  $\Delta\epsilon$  is observed, which we speculate to be the result of a stronger carriers/phonon interaction to localize the electron–hole pair at high carrier density (i.e., each domain is more polarized). Another reason could be the contribution of the multi-exciton Auger effect to deplete the carriers. Though Auger lifetime in type II CQWs is very slow due to the apparent momentum mismatch in CT excitons<sup>44</sup>, which is the main reason for the excellent optical gain performance<sup>24</sup>, under high enough fluence as in our experiments, multi-exciton Auger effect might not be ignored anymore.



**Figure 4. Optical transition crossover following various excitation fluences.** (a-d) The pseudocolor plots of TR absorbance spectroscopy at increasing excitation fluences. The black dots indicate the wavelength position of the bleaching peak of the CT state. (e) Blue-shifted energy of the CT states as a function of the cubic root of excitation fluence. The solid line is a linear fitting and the error bars are from the fitting uncertainties to obtain the peak position from CT bleaching signals. (f) The formation time of the transient  $E$ -field as a function of excitation fluence and the error bars come from the exponential fitting uncertainties to extract the built-up dynamics of bleaching signals at CT minimum.

## Conclusion

In summary, we have demonstrated, for the first time, the ultrafast modulation of indirect transition in type II colloidal quantum wells by an all-optical approach. The observed blueshift-redshift crossover of the indirect transition energy is closely analogous to the carrier equilibrium process in a p-n junction: the transient  $E$ -field created by the separated charges at the heterojunctions opposes the localization process driven by the band offset. Through transient absorption spectroscopy, we have revealed that the formation and decay of the transient  $E$ -field occur on the picosecond timescale and are controllable via adjusting the excitation fluence. Our results suggest that type II colloidal quantum wells, with their remarkable optical properties and preferable solution-processability, hold great promise for the development of novel optical switching and modulators.

## Methods

**Materials.** Cadmium oxide [CdO] (99.9%), cadmium acetate dihydrate [Cd(OAc)<sub>2</sub>·2H<sub>2</sub>O] (>98%), Myristic acid (>99%), tellurium (Te) (99.99%), selenium (Se) (99.99%), oleic acid (OA) (90%), 1-octadecene (ODE), trioctylphosphine (TOP), hexane, ethanol (EtOH), were purchased from Sigma Aldrich.

**Synthesis of 4 ML CdSe CQWs.** In typical 4 ML CdSe CQWs synthesis, 77 mg of CdO, 340 mg of Myristic acid and 28 mL of ODE were added in a 50 mL round bottom flask and the solution was degassed at 100 °C. Then the solution was heated up to 285 °C under argon flow until it becomes a colourless transparent solution and it was cooled down to 90 °C. In another flask 24 mg of Se powder was dissolved in 2 mL of ODE by using ultrasonication. At 90 °C, the Se-precursor solution was injected into Cd-precursor solution and the temperature was set to 235 °C. After the temperature reached 195 °C, 160 mg Cd(OAc)<sub>2</sub>·2H<sub>2</sub>O was added swiftly. At 235 °C, the reaction was maintained for



10 minutes and it was terminated with the addition of 1 mL OA injection. The NPL solution was cooled down to room temperature by using a water bath and 5 mL hexane was injected to increase the solubility of NPLs. Obtained CdSe CQWs solution was washed with the addition of EtOH at 6000 rpm for 5 minutes. Precipitates re-dissolved in hexane.

**Synthesis of 4 ML CdSe/CdTe Core-Crown CQWs.** For the synthesis of CdSe/CdTe Core-Crown NPLs, a similar method in the literature was used<sup>19</sup>. In a 50 mL round bottom flask, 1 ml of CdSe NPL solution, 30 mg of Cd(OAc)<sub>2</sub>·2H<sub>2</sub>O, 0.45 μL of OA and 4 of mL ODE were mixed with a magnetic stirrer and was degassed at 100 °C. Then, under argon flow, the temperature increased to 215 °C and at this temperature, 0.03 M TOP-Te solution in ODE was injected with a rate of 4 ml/h by using a syringe pump. After a certain amount of precursor injection (according to the desired crown size), the reaction was cooled down to room temperature by using a water bath and 3 mL hexane was injected. Products were washed with EtOH addition and centrifuged at 6000 rpm for 5 minutes. Precipitates re-dissolved in hexane.

**Steady-state optical properties of CdSe/CdTe CQWs.** Absorption spectra of CQWs in hexane are measured by a UV-VIS spectrophotometer (Shimadzu, UV-1800). PL spectra of CQWs in hexane are recorded using a spectrofluorophotometer (Shimadzu, RF-5301PC, excitation wavelength: 355 nm).

**Time-resolved photoluminescence of CdSe/CdTe CQWs.** Time-resolved PL (trPL) measurements are performed with a streak camera from Optronics. The 400-nm pump laser pulses for trPL are generated from a 1000 Hz regenerative amplifier (Coherent Libra™). The beam from the regenerative amplifier has a center wavelength at 800 nm, a pulse width of around 150 fs and is seeded by a mode-locked Ti-sapphire oscillator (Coherent Vitesse, 80MHz). 400-nm pump laser was obtained by frequency doubling the 800-nm fundamental regenerative amplifier output using a BBO crystal. All measurements are performed in CQWs dispersed in hexane at room temperature in ambient air (53±2% humidity) conditions.

**Determination of  $\langle N_{e-h} \rangle$  in CdSe/CdTe CQWs.** Considering the single-photon pumping regime (i.e., excitation wavelength: 410 nm), we adopt this equation<sup>45</sup> to calculate the number of excitation-generated electron-hole pairs  $\langle N_{e-h} \rangle$ :  $\langle N_{e-h} \rangle = f \cdot \sigma / (h\nu)$ , where  $f$  is the pump fluence (μJ/cm<sup>2</sup>),  $\sigma$  is the absorption cross-section (cm<sup>2</sup>), and  $h\nu$  is the photon energy (μJ) of pumping pulse. Here, we calculate  $\sigma$  based on the absorbance data using the method reported by Yu *et al.*<sup>46</sup>. The extracted absorption cross-section at 410 nm is  $9.32 \cdot 10^{-14}$  cm<sup>2</sup>.

**Transient absorption spectroscopy of CdSe/CdTe CQWs.** TA spectroscopy is performed using a Helios™ setup (Ultrafast Systems LLC) and in transmission mode with chirp-correction. The white light continuum probe beam (in the range of 400-800 nm) is generated from a 3 mm sapphire crystal using 800 nm pulse from the regenerative amplifier as mentioned in trPL measurement. The pump beam spot size is ~0.5mm. The probe beam passing through the sample was collected using a detector for UV-Vis (CMOS sensor). All measurements are performed at room temperature in solution (hexane).

## Supporting Information

Time-resolved PL; Supplemental data of time-resolved absorbance spectroscopy excited at 410 nm; Gaussian fitting results; Time-resolved absorbance spectroscopy excited at 590 nm; The dissipation time of the transient E-field.

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## Notes

The authors declare no competing financial interests.

The raw data and processed data are available via reasonable request to the corresponding authors.

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## The TOC graphic

