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Morphology, Crystal Structure and Adsorption Performance of Hydrothermally Synthesized Titania and Titanate Nanostructures

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Titania (TiO₂) and sodium titanate nanostructures with controllable phases and morphologies were synthesized by hydrothermal method with titanium disulfide (TiS₂) as the starting material. Sodium titanate nanobelts could be synthesized under a relatively low alkaline concentration (1 mol L⁻¹ NaOH) and short duration (6 hours). At 3 mol L⁻¹ HCl, rutile TiO₂ nanorods were synthesized. Anatase TiO₂ nanoparticles were obtained under pH values ranged between 2 to 13. FTIR analysis confirmed the phase change as the pH of the reacting medium increased from highly acidic to highly alkaline condition. The adsorption of methylene blue (MB) on the as-synthesized sodium titanate nanobelts fitted well with the Langmuir monolayer model, with an adsorption capacity as high as 312.5 mg g⁻¹. The kinetics of MB adsorption was found to be pseudo-second-order kinetic model. In brief, this study demonstrates a simple method to control the phase and morphology of titanium-based oxides. Excellent performance has been shown in the MB adsorption test by the sodium titanate nanostructures.

Keywords: Layered titanate; Titanium oxide; Nanostructures; Hydrothermal Synthesis; Adsorption

Introduction

Titania (TiO₂) is a wide band-gap semiconductor possessing specific chemical properties and stability. It exists in three distinct phases: anatase, rutile and brookite under atmospheric pressure. Both anatase and brookite are metastable while rutile is a thermodynamically stable phase. TiO₂ is widely employed in various applications, especially in photocatalysis^{1,2} and photovoltaics.³ TiO₂ can be synthesized *via* various methods, such as sol-gel⁴ and hydrothermal.⁵ In comparison with other techniques, hydrothermal synthesis is an inexpensive and environment-friendly method⁵ with the ability to control chemical composition and morphology of the synthesized products.⁶

The synthesis conditions affecting the phase composition and morphology of hydrothermally-synthesized TiO₂ are time and temperature of synthesis, and pH of the reacting solution. It has been reported that a gradual phase change from anatase to rutile was observed as the acidity increased.^{7,8} Moreover, the TiO₂ can also be prepared from the TiS₂ by hydrothermal method⁹ under acid condition or by heat treatment.¹⁰ However, to the best of our knowledge, there is no report on subjecting TiS₂ under alkaline condition to synthesize TiO₂ and titanate compounds. There is also no systematic investigation of how a wide range of solution acidity would affect the morphology and crystal structure of the final titanium oxide(s) using TiS₂ as the starting material.

In recent years, titanate nanostructures derived via alkaline conditions have attracted much attention¹¹⁻¹⁴ due to their potential in photocatalysis,^{15,16} ion-exchange,¹⁷⁻¹⁹ and adsorption related applications.^{20,21} Since Kasuga et al. devised a method to synthesize titanate compounds by hydrothermal method,²² there has been a growing interest in producing TiO₂-derived titanate structures, including nanowires^{16,23,24} and nanotubes.^{12,25,26} However, these titanate products were usually synthesized under very high alkaline concentrations (5-20 mol L⁻¹ NaOH), and the reaction time typically ranged from 12 to 48 hours.^{12,19} In this study, various titania and titanate nanostructures with controllable phases and

morphologies were synthesized from TiS_2 . For the first time, we report that sodium titanate nanostructures could be synthesized at 1 mol L^{-1} NaOH concentration in a relatively short duration of 6 hours. Excellent adsorption performance was demonstrated by the as-synthesized titanate nanostructures.

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Experimental Methods

Sample preparation

In a typical procedure, TiS_2 (Aldrich, 99.8%) was added to deionised water, and the pH of the reacting solution was subsequently adjusted by adding either HCl or NaOH of different concentration. The acidity/basicity of the reacting solution ranged from 3 mol L^{-1} HCl to 10 mol L^{-1} NaOH. The above colloid was then transferred into a 20 mL-Teflon-lined stainless steel autoclave until 60% of its volume was filled. The autoclave was maintained at 150°C for 6 hours and then air-cooled to room temperature. The samples were collected via centrifugation, gently washed with deionised water and ethanol, and dried in an oven at 80°C . For comparison purpose, commercial anatase powder (Aldrich, 98%) was also used as the starting material and subjected to the same experimental conditions.

Characterization

The X-ray diffraction (XRD) patterns were obtained on a Shimadzu X-ray diffractometer using Cu $K\alpha$ radiation ($\lambda = 1.54178 \text{ \AA}$) at ambient temperature (25°C). The accelerating voltage and applied current were 40 kV and 40 mA respectively. The morphologies of the samples were examined by field-emission scanning electron microscopy (FESEM, JEOL JSM-6340F) and transmission electron microscopy (TEM, JEOL JEM-2010). The chemical structure of the as-synthesized products was examined using Fourier-transformed infrared spectroscopy (FTIR, Perkin Elmer FTIR System Spectrum GX) in the range of $400\text{-}4000 \text{ cm}^{-1}$.

Dye Adsorption Test

Methylene blue (MB) was used to determine the sample's adsorption kinetics and capacities. The adsorption tests were carried out in the dark to prevent the degradation of MB under light. The equilibrium adsorption isotherm was determined with MB solution (50 mL) with varying concentration from 20 to 80 mg L^{-1} . In each run, the adsorbent (10 mg) was added to the solutions so that the adsorbent loading was 200 mg L^{-1} . After 4320 minutes, the equilibrium concentration was measured. To understand the adsorption mechanism, the equilibrium adsorption isotherm data were curve-fitted into the well-known Freundlich and Langmuir models:

$$\text{Freundlich: } q = K_F C^{\frac{1}{n}} \quad (1)$$

$$\text{Langmuir: } q = \frac{q_m KC}{1 + KC} \quad (2)$$

where q (mg g^{-1}) is the amount of adsorbed MB, C (mg L^{-1}) is the concentration of MB at equilibrium, q_m (mg g^{-1}) is the maximum adsorption capacity, K_F and n are Freundlich constants, and K is the Langmuir constant. If the adsorption isotherm exhibits Langmuir behaviour, it indicates a monolayer

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adsorption. On the other hand, a good fit into the Freundlich model indicates a heterogeneous surface binding.

For the kinetics experiment, the synthesized powder (20 mg) was added to MB solution (100 mL) with an initial concentration of 100 mg L⁻¹. The adsorbent loading was therefore 200 mg L⁻¹. Stirring was applied throughout the duration of the test (1680 minutes) at a speed of 300 rpm. To investigate the rate at which MB was adsorbed on the sodium titanate nanostructures, two kinetic models were employed to fit the experimental data: the pseudo-first-order kinetic model and the pseudo-second-order kinetic model. The pseudo-first-order kinetic model can be expressed in its integral form as:

$$\ln(Q_{e1} - Q_t) = \ln Q_{e1} - k_1 t \quad (3)$$

where Q_{e1} and Q_t are the amount of dye adsorbed (mg g⁻¹) on the adsorbent at equilibrium and at time t respectively, k_1 is the rate constant for the pseudo-first-order kinetic model (min⁻¹), and t the contact time (minutes). The values of $\ln(Q_{e1} - Q_t)$ were calculated from kinetic data.

The pseudo-second-order kinetic model can be expressed in its integral form as:

$$\frac{t}{Q_t} = \frac{1}{k_2 (Q_{e2})^2} + \frac{t}{Q_{e2}} \quad (4)$$

where k_2 (g mg⁻¹ min⁻¹) is the rate constant for the pseudo-second-order kinetic model, and Q_{e2} the equilibrium amount of dye adsorbed (mg g⁻¹).

All of the adsorption experiments were carried out at ambient temperature (25 °C) and at a standardized pH value range of 9.5-10. The initial pH value was close to this range, but slightly differed due to the difference in the MB concentration. Therefore minor adjustment was carried out to unify the starting pH of different solutions. The amount of MB adsorbed by the adsorbent, q , was calculated from the difference between the initial and the final MB concentration as follows:

$$q = \frac{(C_o - C_f)V}{M} \quad (5)$$

where q is the amount of MB adsorbed (mg g⁻¹), C_o and C_f are the initial and final concentrations of MB remaining in the solution (mg L⁻¹), V is the volume of the MB solution (L), and M is the mass of the adsorbent added (g). The experimental data reported was obtained by averaging the data from five parallel experiments conducted simultaneously under the same experimental conditions. The quantitative determination of the MB concentration in solution at different time intervals was performed by measuring the intensity change of a specific absorption peak of the sample solution under a Perkin-Elmer UV-Vis-NIR Lambda 900 spectrophotometer.

Results and Discussion

Phase Composition

Figure 1. XRD patterns of starting material TiS_2 (curve (A)) and TiS_2 -derived products under different acidity/basicity. Curve (B): $3 \text{ mol L}^{-1} \text{HCl}$, (C): $\text{pH}=2$, (D): $\text{pH}=4$, (E): $\text{pH}=11$, (F): $\text{pH}=13$, (G): $\text{pH}=14$, (H): $5 \text{ mol L}^{-1} \text{NaOH}$ and (I): $10 \text{ mol L}^{-1} \text{NaOH}$. Label A stands for anatase, B for brookite, R for rutile and T for titanate. The inset shows a magnified image of the characteristic brookite peak at $\theta \approx 30.7^\circ$.

Figure 2. XRD patterns of anatase TiO_2 -derived products under different acidity/basicity. Curve (A): $3 \text{ mol L}^{-1} \text{HCl}$, (B): $\text{pH}=14$, (C): $5 \text{ mol L}^{-1} \text{NaOH}$, (D): $10 \text{ mol L}^{-1} \text{NaOH}$. Label A stands for anatase, R for rutile and T for titanate.

Figure 1 presents the XRD patterns of the as-synthesized TiO_2 nanostructures using TiS_2 as the starting material (curve (A), Figure 1) hydrothermally treated for 6 hours under different acidity/basicity. When $3 \text{ mol L}^{-1} \text{HCl}$ was used as the reacting solution, rutile was the main phase. In the pH range of 2 to 11, anatase phase was present, though for the product synthesized at pH 11, trace amount of brookite phase was also detected (curve (E), Figure 1). At $\text{pH} = 13$, anatase still remained as the predominant phase, with a small amount of sodium titanate, as seen from the weak peak at $\theta \approx 9^\circ$. When 1 mol L^{-1} or higher concentration NaOH was used as the reacting solution, only sodium titanate phase was present. In comparison, when commercial anatase TiO_2 (A- TiO_2) powder was used as a starting material and synthesized under different acidity/basicity in the pH range of 2 to 13, there was no chemical reaction at all (the XRD data not shown). Only trace amount of rutile was formed in the

samples subjected to 3 mol L⁻¹ HCl and in pH 14 medium (Fig. 2). When 5 mol L⁻¹ NaOH was used as the reacting solution, trace amount of rutile and titanate products were synthesized (Fig. 2). All of the strong peaks corresponded to the un-reacted starting material A-TiO₂. It was only when A-TiO₂ was subjected to 10 mol L⁻¹ NaOH reacting solution that it was completely transformed to sodium titanate (Fig. 2). This observation agrees with what have been reported by other researchers,^{19, 21, 23-25} where high alkaline concentration (≥ 5 mol L⁻¹ NaOH) is needed to transform the starting material TiO₂ to titanate.

10 Figure 3. A schematic of proposed formation mechanism of titania and titanate samples synthesized under different acidity/basicity

The formation mechanism of different phases from TiS₂ under different acidity/basicity is proposed in Figure 3. When TiS₂ is added into aqueous solution, hydrolysis reaction occurs which leads to the formation of Ti(OH)_x^{4-x} complex in the solution:

Under highly acidic condition in aqueous medium (e.g. 3 mol L⁻¹ HCl), the number of OH⁻ coordinated to Ti⁴⁺ centres is limited, leading to the formation of a more positively charged Ti(OH)_x^{4-x}. A more positively charged Ti(OH)_x^{4-x} encourages corner-sharing of the TiO₆ octahedra, favoring the formation of rutile phase.²⁷ Similarly, when the solution pH ranges from 2 to 13, there is an increase in the number of OH⁻ coordinated to Ti⁴⁺ centres, and hence the complex Ti(OH)_x^{4-x} would be less positively charged, which in turn encourages edge-sharing of TiO₆ octahedra. Such an arrangement favors the formation of anatase phase.²⁷ For titanate formation, the dissolution/crystallization process of the precursor takes place.²⁸ Therefore, the bond-breaking of Ti-S (or Ti-O, for TiO₂) is the limiting step for TiS₂, as the formation of Ti(OH)_x^{4-x} is required in order for titanate layers to form by ololation of Ti(OH)_x^{4-x}. It has been reported that the bond energy for Ti-S (418±3 kJ mol⁻¹) is lower than Ti-O (672±9 kJ mol⁻¹),²⁹ which may facilitate the formation of titanate in lower concentration of NaOH (1 mol L⁻¹) compared to that using TiO₂ as precursor.

Figure 4. FTIR spectra of the as-synthesized powder. Curve (A) is for rutile synthesized in 3 mol L⁻¹ HCl, (B) for anatase synthesized in pH 11, and (C), (D), (E) are for sodium titanate synthesized in pH 14, 5 mol L⁻¹ NaOH and 10 mol L⁻¹ NaOH respectively.

Figure 4 presents the FTIR spectra of the as-synthesized TiO₂ and titanate products formed under different HCl and NaOH concentrations. Hydroxyl groups could be observed for the sodium titanate products due to the existence of a bending vibration of O-H-O at a wavenumber (ν) of 1630 cm⁻¹ and a strong stretching vibration of O-H group at ν =3400 cm⁻¹. The relatively high intensity of the peaks associated with the hydroxyl groups suggested that large amount of hydroxyl groups were located on the surface and in the layers of the sodium titanate products.³⁰ It could also be inferred from Figure 4 that the amount of hydroxyl groups present decreased as the pH of the synthesis condition decreased, as the intensity of the spectra at ν =1630 cm⁻¹ and ν =3400 cm⁻¹ was reduced from the one that was synthesized at a higher pH to one at a lower pH. Two sharp peaks at ν =500 cm⁻¹ and ν =900 cm⁻¹ observed for sodium titanate products (curves (C)-(E)) were attributed to the stretching of Ti-O found in titanate products.^{30, 31} As expected, the spectra of the rutile (curve (A)) and anatase (curve (B)) products did not contain the peaks at ν =500 cm⁻¹ and ν =900 cm⁻¹.

Figure 5. FESEM images of hydrothermally-synthesized TiO_2 and titanates in (a) $3 \text{ mol L}^{-1} \text{ HCl}$, (b) $\text{pH}=13$, (c) $\text{pH}=14$ using TiS_2 as precursor, and (d) titanate nanotubes in $10 \text{ mol L}^{-1} \text{ NaOH}$ using A-TiO_2 as precursor.

5 Morphology

The morphologies of the synthesized titania and sodium titanate are shown in Figures 5, 6 and 7. It is clearly shown that the acidity/basicity of the reacting medium affects the morphology of the as-synthesized products. From Figures 5a and 6a, rutile TiO_2 synthesized using $3 \text{ mol L}^{-1} \text{ HCl}$ possessed nanorod morphology, with an average length of 100 nm and average diameter of 10 nm. Anatase TiO_2 synthesized in the pH range of 2 to 13 were agglomerated nanoparticles (Figs. 5b and 6c). The morphologies of titanate products synthesized from 1 mol L^{-1} , 5 mol L^{-1} and $10 \text{ mol L}^{-1} \text{ NaOH}$ solutions consisted of nanobelt or nanosheet type morphology (Figs. 5c and 7a). The average width of the nanobelts was 7 nm. However, the morphology of sodium titanate synthesized from $10 \text{ mol L}^{-1} \text{ NaOH}$ with A-TiO_2 as the starting material (Figs. 5d and 7c) resembled nanotubular-like structure, which is in good agreement with many previous reports.^{18, 25, 32} The mechanism for the nanotube formation has been explained as the folding of nanosheet under asymmetrical surface tension.^{33, 34} However in the case of TiS_2 as the starting material, the same condition ($10 \text{ mol L}^{-1} \text{ NaOH}$) produced nanobelt (or nanosheet) structure instead. An apparent explanation could be that the presence of sulfur ions may have modified the surface tension, such that there is insufficient drive for the wrapping action.

Figure 6. TEM images of hydrothermally-synthesized TiO₂ in (a) 3 mol L⁻¹ HCl and (c) pH 11, (b) high-magnification image of (a), and (d) high-magnification image of (c).

Figure 7. TEM images of hydrothermally-synthesized titanates in (a) 10 mol L⁻¹ NaOH using TiS₂ as precursor, and (c) 10 mol L⁻¹ NaOH using A-TiO₂ as precursor, (b) high-magnification image of (a), and (d) high-magnification image of (c).

5 Adsorption Kinetics

To investigate the rate of adsorption of MB by the various titania and titanate nanostructures, adsorption kinetics tests were carried out. Figure 8 illustrates the relative MB concentration change with time in the first 120 minutes when the initial concentration was 100 mg L⁻¹. It is clear that the rates of adsorption of the as-synthesized rutile and anatase samples (TiS₂-3 mol L⁻¹ HCl and TiS₂-pH 11 respectively) were negligible, whereas substantial amounts of MB were adsorbed by the as-synthesized sodium titanate nanobelts (TiS₂-pH 14, TiS₂-5 mol L⁻¹ NaOH and TiS₂-10 mol L⁻¹ NaOH). After 1680 minutes, the percentages of MB adsorbed by TiS₂-pH 14, TiS₂-5 mol L⁻¹ NaOH and TiS₂-10 mol L⁻¹ NaOH titanate nanobelts were approximately 37%, 45% and 53% respectively.

Figure 8. MB concentration change with time during adsorption by the titania and titanate nanostructures

Table 1. Parameters of pseudo first- and second-order adsorption kinetics for MB adsorption onto various titanate nanostructures.

Adsorbent	Experimental Q_{e1}^a	Pseudo-first-order kinetics			Pseudo-second-order kinetics		
		k_1^b	$Q_{e1, cal}^c$	R^2	k_2^d	$Q_{e2, cal}^e$	R^2
TiS ₂ -pH 14	253.12	0.0032	169.12	0.59	4.52×10^{-4}	166.67	0.99
TiS ₂ -5 mol L ⁻¹ NaOH	277.90	0.0042	107.33	0.36	1.73×10^{-3}	227.27	0.99
TiS ₂ -10 mol L ⁻¹ NaOH	296.40	0.0056	97.67	0.42	1.16×10^{-3}	263.16	0.99

^a Q_{e1} (mg g⁻¹)

^b k_1 (min⁻¹)

^c $Q_{e1, cal}$ (mg g⁻¹)

^d k_2 (g mg⁻¹min⁻¹)

^e $Q_{e2, cal}$ (mg g⁻¹)

Based on the fitting of kinetic data to the pseudo-first-order kinetic model, the R^2 values (Table 1) for the adsorption of MB on all the as-synthesized titanate nanostructures were extremely low, ranging from 0.36 to 0.59. This indicates that the adsorption kinetics does not follow the pseudo-first-order kinetic model. On the other hand, when the MB adsorption kinetics was modelled to the pseudo-second-order kinetic model, the R^2 values were close to unity for all as-synthesized titanate nanostructures (Table 1). This indicates that the kinetics of adsorption of MB on the as-synthesized titanate nanostructures could be described more accurately by the pseudo-second-order kinetic model. Similar kinetics model has also been observed in the adsorption of MB onto rice husks³⁵ and date pits.³⁶

Figure 9. Adsorption isotherms of various titania and titanate nanostructures in MB. Y-axis refers to the amount of MB adsorbed by the nanostructures, X-axis refers to the equilibrium concentration of MB.

Table 2. Langmuir and Freundlich isotherm parameters for MB adsorption onto various titanate nanostructures.

Adsorbent	Experimental q_m^a	K^b	Langmuir q_{ml}^c	R^2	K_F^d	Freundlich n	R^2
TiS ₂ -pH 14	254.76	0.88	227.27	0.99	98.90	3.60	0.69
TiS ₂ -5 mol L ⁻¹ NaOH	290.19	0.72	303.03	0.99	113.04	2.83	0.83
TiS ₂ -10 mol L ⁻¹ NaOH	297.24	0.70	312.50	0.99	111.09	2.45	0.84
TiO ₂ -10 mol L ⁻¹ NaOH	244.95	0.45	256.41	0.99	84.56	2.95	0.72

^a q_m (mg g⁻¹)

^b K (L mg⁻¹)

^c q_{ml} (mg g⁻¹)

^d K_F (mg¹⁻ⁿ Lⁿ g⁻¹)

Adsorption Isotherms

To investigate the MB adsorption capacities of the as-synthesized titanate nanostructures, adsorption isotherm tests were carried out. Figure 9 presents the adsorption isotherms of MB with various titania and titanate compounds. It is clear that there was negligible adsorption of MB on TiS₂-3 mol L⁻¹ HCl. The results of fitting the isotherm curves to Freundlich and Langmuir models were summarized in Table 2. Regression coefficients (R^2) for different titanate compounds were all close to unity for the Langmuir model, indicating that the Langmuir model describes well the MB adsorption behaviour. On the other hand, the R^2 values for all the titanate compounds were less than 0.9 when the data were fitted with the Freundlich model. This illustrates that the adsorption of MB by the as-synthesized titanate compounds is mainly governed by monolayer adsorption, as described by the Langmuir model. From Table 2, among the titanate compounds tested in the adsorption of MB, TiS₂-10 mol L⁻¹ NaOH had the highest q_m (297.24 mg g⁻¹) and q_{ml} (312.50 mg g⁻¹). On the contrary, q_m and q_{ml} of the TiO₂-10 mol L⁻¹ NaOH was significantly smaller. It could be inferred that the number of available sites for

adsorption on the titanate nanobelts was larger than the number of available sites for adsorption on the titanate nanotubes. The adsorption mechanism may be due to electrostatic force of attraction between the adsorbent and MB.³⁷⁻³⁹ It was stated that the dominating surface charge for titanate in alkaline pH was negative. As the isotherm adsorption experiments were carried out at pH 9.5-10, the dominant surface charge for the titanate nanostructures was negative. This phenomenon was also observed in our previous works.^{20, 40} MB, a cationic dye, can be attracted by titanate nanostructures via electrostatic force of attraction. Compared with existing data in the literature, the sodium titanate nanostructures synthesized in this work exhibited much larger adsorption capacities (Table 3).

10 Table 3. Comparison of the maximum monolayer adsorption of MB on various adsorbents.

Adsorbent	$q_{ml} (mg\ g^{-1})$
Rice husks ³⁵	40.58
Raw date pits ³⁶	80.29
Calcined Titanate Nanotubes ⁴¹	133.33
Titanate Nanobelts (Present work)	312.50

Conclusion

15 Phase- and morphology-controllable titania and titanates nanostructures were hydrothermally synthesized using TiS_2 as precursor. While sodium titanate could be synthesized from TiO_2 at concentrations of 5-20 mol L^{-1} NaOH for a relatively long duration, by using TiS_2 as the starting material, sodium titanate could be synthesized when the concentration of NaOH was 1 mol L^{-1} with a shorter processing time. This is due to the relatively weaker Ti-S bond strength as compared to Ti-O, which translates to less drastic condition required to break up the Ti-S bond to enable the formation of titanate layers by ololation of $Ti(OH)_x^{4-x}$. It was also observed that the as-synthesized products exhibited different morphologies when different acidity/basicity of the reacting medium was used. The equilibrium isotherm test data fitted well with the Langmuir model, which indicated that the MB was adsorbed to the titanate surface by monolayer adsorption. The adsorption capacities of the as-synthesized sodium titanate nanostructures were also much better than the adsorption capacity of sodium titanate nanotubes synthesized from TiO_2 . Pseudo-second-order kinetic model was found to describe the adsorption kinetics well. The strong adsorption demonstrated by the synthesized titanate nanostructures deserves further attention as they can be potentially applied in the removal of cationic dyes in water.

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Notes and References

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