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Contrast and Raman spectroscopy study of single- and few-layered charge density wave material: 2H-TaSe₂

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In this article, we report the first successful preparation of single- and few-layers of tantalum diselenide (2H-TaSe₂) by mechanical exfoliation technique. Number of layers is confirmed by white light contrast spectroscopy and atomic force microscopy (AFM). Vibrational properties of the atomically thin layers of 2H-TaSe₂ are characterized by micro-Raman spectroscopy. Room temperature Raman measurements demonstrate MoS₂-like spectral features, which are reliable for thickness determination. E_{1g} mode, usually forbidden in backscattering Raman configuration is observed in the supported TaSe₂ layers while disappears in the suspended layers, suggesting that this mode may be enabled because of the symmetry breaking induced by the interaction with the substrate. A systematic *in-situ* low temperature Raman study, for the first time, reveals the existence of incommensurate charge density wave phase transition in single and double-layered 2H-TaSe₂ as reflected by a sudden softening of the second-order broad Raman mode resulted from the strong electron-phonon coupling (Kohn anomaly).

The extraordinary properties of graphene as a pioneering two-dimensional (2D) material have created tremendous interest for searching other possible 2D materials with unique electronic and optical attributes. A promising group of such materials is transitional metal dichalcogenides (TMDs) which have weakly coupled layered structure just like graphite¹. Most popular member of these TMDs is molybdenum disulfide (MoS₂). Recent calculations predict that majority of these TMDs are expected to have stable single layer form which may exhibit rich physical properties ranging from metallic to direct or indirect semiconductors², charge density wave (CDW)³ and superconductivity^{4,5} phase. Though bulk TMDs have been extensively studied by Raman spectroscopy^{5,6}, angle-resolved photoemission spectroscopy (ARPES)⁷, and neutron diffraction⁸⁻¹⁰, their potential as 2D-TMD materials is being realized in recent times¹¹.

Means of fabrication of 2D thin layers of TMDs and reliable yet efficient techniques to identify exact number of layers must be developed for further exploring their electrical, optical and other properties, thus eventually realize the scientific and technological potentials of these materials. It is known that Raman spectroscopy is a very reliable tool for studying vibrational properties of materials and has gained new meaning with emergence of 2D materials. Recently, Raman spectroscopy has been widely adopted to probe the number of layers¹², the stacking order^{13,14}, the edge orientations^{15,16}, the molecular doping¹⁷, the strain effects^{18,19} and other properties of 2D materials, such as graphene²⁰, MoS₂²¹ and *etc.*

2H-TaSe₂, a candidate of TMDs, is metallic at room temperature with crystal structure of hexagonal D_{6h}⁴ symmetry²². Metallic TMDs exhibit Peierls transition at certain critical temperatures where the overall energy of the system could be lowered by distorting and creating superlattice over underlying crystal lattice^{3,23}. 2H-TaSe₂ in bulk has transition from normal (metallic) phase to the incommensurate charge-density-wave (ICDW) phase at 123 K, followed by the commensurate charge-density-wave (CCDW) phase transition at ~90 K²².

In this report, we demonstrate the first successful preparation of atomically thin layers of TaSe₂ on 300 nm SiO₂/Si substrates by mechanical exfoliation and identification of their layer numbers by using white light contrast spectroscopy and atomic force microscopy (AFM). Room temperature and *in-situ* low temperature micro-Raman spectroscopy study on single- and few-layered TaSe₂ have been systematically conducted. Like MoS₂, the difference of the vibrational energies between E_{2g}¹ and A_{1g} mode can be used as a reliable and effective way to identify the number of TaSe₂ layers. E_{1g} mode while usually forbidden in TMDs with the same crystal symmetry in backscattering Raman configuration has been observed in the thin (less than 4 layers) TaSe₂ samples



supported on the 300 nm SiO₂/Si substrate. In contrast, the E_{1g} peak does not present in the suspended bilayer and tri-layer TaSe₂. We attribute the appearance of E_{1g} here to the symmetry breaking due to the interaction between the atomic thin layers and the substrate. For single and bilayer TaSe₂ samples, a second-order broad Raman peak caused by the strong electron-phonon coupling shows redshift when the temperature is lowered from room temperature to the liquid nitrogen temperature. It is noticed that a sudden drop of the energy of the second-order Raman mode occurs when the temperature is close to ICDW phase transition point of the bulk TaSe₂, which could be an unambiguous evidence of the existence of ICDW phase in the atomic thin layers of TaSe₂²². The existence and behavior of CDWs in atomically thin materials and their possible applications offer us interesting and unexplored field.

Results

Layer identification. Thin layers of TaSe₂ samples were transferred onto 300 nm SiO₂/Si substrate through mechanical exfoliation of commercially available bulk 2H-TaSe₂ material using well-known scotch tape method. Number of layers has been confirmed by both white light contrast spectroscopy and AFM.

The white light contrast spectroscopy has been used for identification of layer numbers of atomically thin material such as graphene²⁴ and TMDs²⁵ through comparison of absorption of white light by different number of layers. Atomically thin TaSe₂ samples deposited on 300 nm SiO₂/Si substrate possesses linear relationship between white light absorption and layer number (see Fig. 1f).

AFM measurements done in a tapping mode reveals high concentration of surface adsorbents and impurities in thin TaSe₂ samples (Fig. 2c). According to the first principle calculations, thickness of single layer TaSe₂ is expected to be around 0.65 ± 0.05 nm which is in good agreement with our AFM measurements^{26,27}. The AFM height profile measured at the step of single layer TaSe₂ sitting on TaSe₂ crystal shows ~0.75 nm thickness which is consistent with the theoretical value²⁸. However, the thickness readings of the single layer TaSe₂ directly on the SiO₂/Si substrate range from 1 nm to 1.5 nm which can be attributed to the large surface roughness of the substrate, adsorbents and impurities between substrate and the TaSe₂ layer (Fig. 2d)²⁹.

Identification of layer numbers of thin samples by AFM can be challenging especially if samples have been exposed to ambient air for a long time. On the other hand, white light contrast imaging gives us very precise values for number of layers even for weeks-old samples. Therefore, we conclude that contrast spectroscopy is more reliable and time-efficient methods for identifying number of layers. Recently, Gomez *et al.* published first article for fast identification of TaSe₂ samples using contrast spectroscopy²⁹. However thinnest sample they achieved by exfoliation is 5 layers in thickness.

Raman spectroscopy. In normal (metallic) phase, 2H-TaSe₂ has crystal structure of hexagonal D_{6h}⁴ symmetry. For one-phonon processes, unit cell has following vibrational modes:

$$A_{1g} + 2B_{2g} + E_{1g} + 2E_{2g} + 2A_{2u} + B_{1u} + 2E_{1u} + E_{2u}$$

Among them, A_{1g}, E_{1g}, E_{2g}¹ and E_{2g}² are Raman active modes (Fig. 3)²².

From previous Raman measurements on bulk 2H-TaSe₂, in backscattering Raman configuration, observable Raman peaks are (i) A_{1g} - out of plane vibrational mode at 235 cm⁻¹, (ii) in plane vibrations E_{2g}¹ at 208 cm⁻¹ and E_{2g}² - also known as rigid layer mode with the low energy at around 23 cm⁻¹ and (iii) very broad second-order peak due to two-phonon process at around ~140 cm⁻¹.²² For this crystal symmetry, E_{1g} vibrational mode is forbidden in backscattering Raman configuration³⁰. From previous studies of bulk 2H-TaSe₂ in non-backscattering Raman configuration, the E_{1g} is observed at 137 cm⁻¹.³¹ The two-phonon Raman process can be described as either (i) creation or destruction of two phonons or (ii) simultaneous creation of one-phonon with destruction of another phonon. The second-order Raman peak is related to two-phonon density of states. The intensity of this peak at certain frequency ω depends on density of states of phonon pairs with frequency difference or sum equal to ω. Hence, the second-order Raman peak is a broad peak that arises from overlapped sharp peaks^{31,32}.

Figure 4 shows Raman spectra of TaSe₂ samples supported on the 300 nm SiO₂/Si substrate with thicknesses varying from single layer to bulk obtained in backscattering configuration. There are three prominent Raman features, which are A_{1g}, E_{2g} and 2-phonon modes for TaSe₂ samples from single layer to bulk. Moreover, the E_{1g} mode which is usually forbidden in backscattering Raman configuration

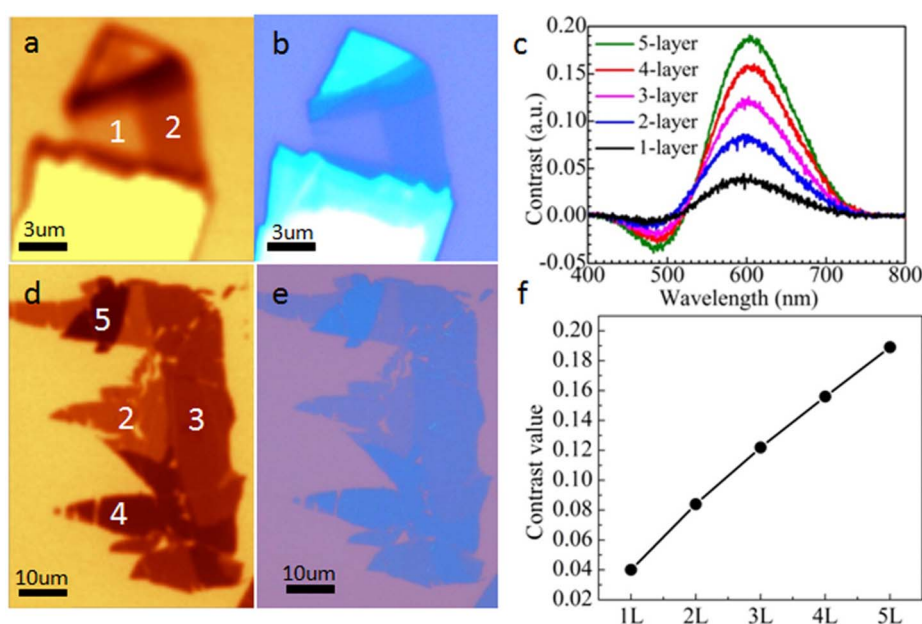


Figure 1 | Layer number identification of atomically thin TaSe₂ samples. (a, d) White light contrast images and (b, e) Optical images of exfoliated TaSe₂ samples with layer number ranging from 1 to 5. (c) White light contrast spectra and (f) extracted contrast values of different number of layers.

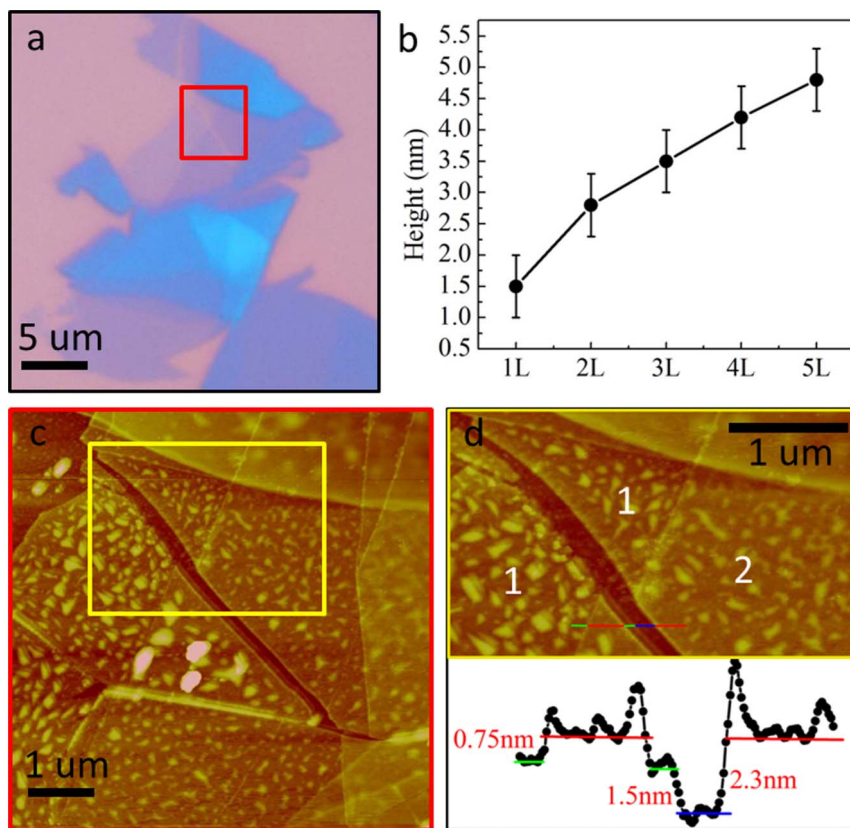


Figure 2 | Layer number identification with AFM. (a) Optical image of single layer and bilayer TaSe₂ (b) heights measured by AFM of samples with different layer numbers (c) AFM image of the area within red square in (a). (d) AFM image (yellow squared area in b) and cross section profile of single layer (green line), bilayer (red line) TaSe₂ samples on Si/SiO₂ (blue line) substrate.

was observed in the thin (less than 4 layers) TaSe₂ samples. To our best effort, E_{1g} peak was not observable in bulk or thin samples of more than 3 layers in thickness. It can be seen from Fig. 4 that the intensity of the E_{1g} mode increases with decreasing number of layers. The reasons that E_{1g} peak of thicker layers does not appear as clearly as it in thin layers could be due to (i) the second order Raman peak intensity increased with the number of layers, which may “suppress” the very weak E_{1g} peak; (ii) the interaction of thicker layers and substrate could be dramatically weaker comparing to the case for very thin layers, this may further weaken the E_{1g} intensity.

Like MoS₂, A_{1g} and E_{2g} peaks approach each other as number of layers decreases. Although the red-shift of the A_{1g} peak is barely observable, nevertheless, a shift within ~1 cm⁻¹ can be detected once the peaks are fit with Lorentz function. From bulk to single layer,

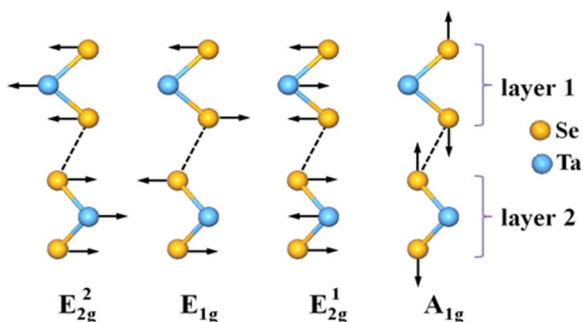


Figure 3 | Vibrational modes for one-phonon processes. A single layer structure corresponds to a sheet of Ta atoms sandwiched between two sheets of Se atoms (trilayer). The bulk 2H polytype contains two trilayers per unit cell.

the E_{2g} mode shows a blue-shift of 6.5 cm⁻¹. These shifts are mainly attributed to: (i) the decrease of the force constant resulted from the weakening of the interlayer Van der Waals force between layers for the A_{1g} mode³³; and (ii) the structure changes or long-range coulombic interlayer interactions for the E_{2g} mode, when the number of layers decreases³⁴.

To elucidate the dependence of the Raman features as a functional of the number of layers, the A_{1g}, the E_{2g} and the two-phonon peaks

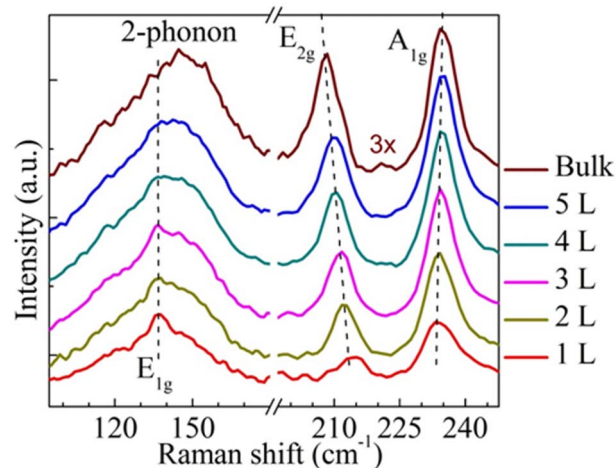


Figure 4 | Raman spectra of TaSe₂ samples, ranging from single layer to bulk in room temperature (in backscattering Raman configuration). As number of layers decrease, the E_{2g} and the A_{1g} peaks approach each other. The E_{1g} peak is easily observed at 137 cm⁻¹ for samples below 4 layers in thickness.

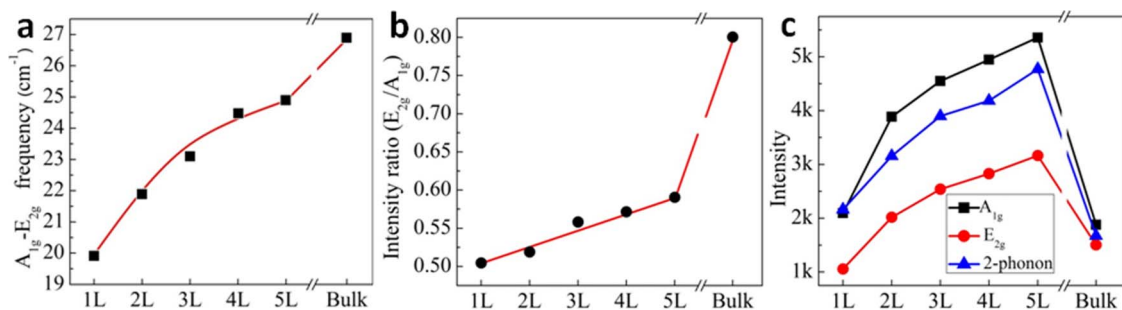


Figure 5 | Identification of number of layers by fitted Raman features. (a) Energy difference between the A_{1g} and the E_{2g} modes saturates to bulk value in inversely exponential manner (b) relative intensity ratio of the E_{2g} peak to the A_{1g} peak in Raman spectrum shows linear relationship with increasing layer number (c) Signal-to-noise ratio of Raman spectrum improves with increasing layer number due to interference enhancement and starts to decline towards bulk value after passing through maximum point at certain layer number.

are fitted by Lorentz function and the fitting results are plotted in Fig. 5. Similar to MoS_2^{34} , the energy difference of the A_{1g} and the E_{2g} is the smallest for the single layer, however, increment for this energy difference decreases with increasing layer number (Fig. 5a). The intensity increases with increasing number of layers and reaches a maximum value at certain number of layers (above 5 layers), then declines towards bulk value (Fig. 5c). This behavior is attributed to optical interference effect between thin layers and SiO_2/Si substrate¹². On the other hand, the relative intensity of the E_{2g} mode to the A_{1g} mode decreases more rapidly with decreasing number of layers (Fig. 5b). Observation of these associations between different layers could be another reliable way to identify the number of layers of thin TaSe_2 samples.

As noticed in Figure 4, the intensity of the E_{1g} mode increases with decreasing number of layers. To further investigate the origin of this anomaly, we performed Raman measurements on suspended tri-layer TaSe_2 samples because the signal-to-noise ratio of suspended tri-layer TaSe_2 is acceptable for data analysis as the Raman signal of the suspended samples is considerable weaker than that of the supported samples. The temperature-dependent Raman spectra of supported 3-layer TaSe_2 (Supplementary Fig. S1b) and suspended 3-layer TaSe_2 (Supplementary Fig. S1c) are from the same sample as illustrated in the optical image of Supplementary Figure S1a measured under the same condition. It can be seen from this figure that for the supported sample, the E_{1g} peak becomes more and more pronounced with decrease of the temperature because the noise and the background of the 2-phonon peak is minimized at low temperature. And the strength of the E_{1g} Raman peak of the supported sample is comparable (in many cases stronger than) to that of the E_{2g} peak. However, even at low temperatures, we did not observe any peak at around the location of the E_{1g} peak which is comparable to the E_{2g} peak. Therefore, we conclude that observation of the forbidden E_{1g} mode in the supported thin TaSe_2 samples might be due to the broken symmetry originating from the interaction with SiO_2/Si substrate. Similar feature, appearance of new Raman inactive peaks due to symmetry breaking has been also observed in few quintuples of thin layered topological insulators^{35,36}.

As mentioned before, 2H- TaSe_2 in bulk has CDW phase transition at low temperatures within liquid nitrogen temperature region²². To investigate the CDW-influenced Raman scattering response together with the Raman feature evolutions with temperature, we performed the temperature-dependent Raman measurements on TaSe_2 samples from single layer to bulk. Figure 6 shows representative temperature-dependent Raman spectra of bilayer TaSe_2 sample in the temperature range of 80 K to 300 K. It can be seen that both the A_{1g} and the E_{2g} modes blue shift with decreasing temperature. Moreover, in our low temperature measurements, it is noticed that the intensities of the E_{2g} mode and the broad two-phonon peak decline with decreasing temperature. Therefore, for thin layers, at temperatures below 120 K, the

fitting results of these two peaks have large error bar in values due to the deteriorated signal-to-noise ratio. In contrast, the intensity of the forbidden E_{1g} mode increases with decreasing temperatures. This may be explained as (i) declining background signal from the two-phonon process that enhances the visibility of this peak and (ii) strain induced effects due to difference in thermal expansion coefficients of substrate and thin films^{37,38}.

The results of our temperature-dependent Raman measurements on TaSe_2 samples from single layer to bulk are summarized in Figure 7. It clearly shows that the energy of both the A_{1g} and the E_{2g} modes increases with decreasing temperatures. Phonon softening with increasing temperature or vice versa is a common behavior, just like in MoS_2^{39} and graphene⁴⁰. This effect can be also observed through laser heating at different laser powers (see Supplementary Fig. S2). Temperature-dependent-shift in phonon energy is related to anharmonic terms in the lattice potential energy. This shift in one-phonon modes depends on anharmonic coupling of the phonons and the thermal expansion of the crystal⁴¹. In addition, in the case of TaSe_2 , energy of the A_{1g} and the E_{2g} modes is also related to electronic susceptibility. CDW phase transition is accompanied by increase in electronic susceptibility which can be observed through relatively sudden stiffening of phonons below critical phase transition temperature (~ 123 K). And it is observed in bulk that this sudden stiffening is more prominent for the E_{2g} mode than the A_{1g} ^{42,22}. However, in our measurements, as can be seen from Figure 6, we cannot observe this sudden stiffening for the E_{2g} because at temperatures below 150 K, E_{2g} peak intensity decreases dramatically and therefore the fitting results of the E_{2g} peak have large error bar in values due to the poor signal-to-noise ratio.

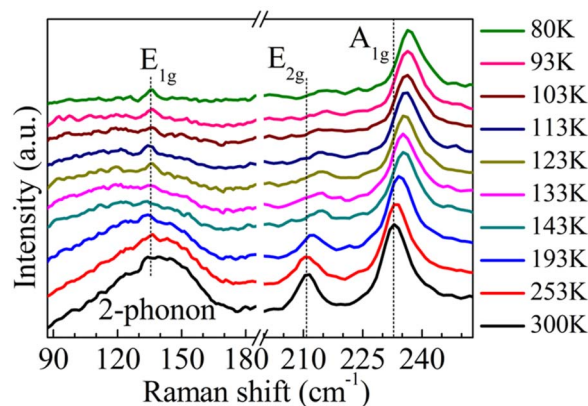


Figure 6 | In-situ low temperature micro-Raman spectra of bilayer TaSe_2 sample.

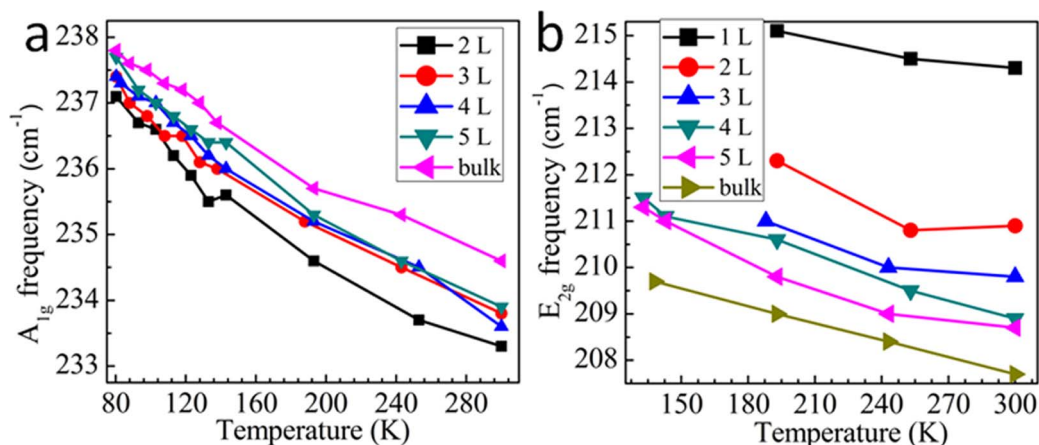


Figure 7 | Layer dependent behavior of the A_{1g} (a) and the E_{2g} (b) modes at low temperatures.

The second-order Raman peak is a very unique optical feature of CDW materials like TaSe₂ and carries rich physics. The Kohn anomaly is an anomaly in the dispersion curve of a phonon branch in a metal. As discussed earlier, the second-order broad Raman peak arises from two-phonon scattering process involving longitudinal acoustic (LA) phonons^{43,44}. The LA phonons in 2H-TMD compounds exhibit a Kohn anomaly which results in high two-phonon density of states. The presence of a Kohn anomaly in a LA mode is due to Fermi surface nesting and strong electron-phonon coupling required for a CDW transition⁸.

Bulk TaSe₂ has Peierls phase transition from metallic to ICDW phase at ~123 K and another transition into CCDW phase at ~90 K. In CCDW phase, 2H-TaSe₂ has superlattice with $3a \times 3a \times c$ unit cell and the same hexagonal D_{6h}^4 symmetry. Below 90 K, four new Raman modes appear at energy range lower than 85 cm⁻¹ in Raman spectrum which are directly associated with the creation of the CDW superlattice over underlying crystal lattice⁴⁵. However, in our measurements, we could not observe these peaks and the E_{2g}^1 rigid-layer mode, because our Rayleigh rejection filter has a cut-off point above 80 cm⁻¹. From previous work done on Raman study of bulk TaSe₂ in CDW phase, we know that on approach to transition temperature from normal to ICDW phase the broad peak due to two-phonon process broadens, exhibits a red-shift and is significantly reduced in intensity^{46,31,22}.

As depicted in Figure 8, an overall softening of the 2-phonon peak can be clearly seen for single, bilayer and bulk TaSe₂. More importantly, a sudden drop of the frequency of this two-phonon mode appear for all the three samples at ~120 K, close to the critical temperature for the phase transition from normal metallic to ICDW as previously reported²². This clearly demonstrates the existence of the ICDW phase in the atomic thin layers of TaSe₂. Recent theoretical studies suggest that CDW distortion for TaSe₂ might be stronger in thin layers than in bulk, however, critical transition temperature should be similar to bulk²⁸. Layer dependent critical transition temperatures cannot be determined in this work due to severely diminished intensity of broad Raman peak which results in large error of values during peak fitting. More details including the low wavenumber modes will be exploited and reported in our future works.

Discussion

We have demonstrated for the first time, the mechanical exfoliation of single layer TaSe₂ and layer number identification for thin samples using well-known AFM measurement and white light contrast spectroscopy techniques. Room temperature Raman measurements for atomic thin layers of TaSe₂ reveal resembling features to MoS₂ and

other layered TMDs. We have additionally observed the forbidden E_{1g} peak in backscattering configuration in all thin layers, particularly below 4-layered samples, which is probably due to the symmetry breaking induced by the interaction between thin TaSe₂ layers and substrate. Low temperature Raman measurements show similar properties for thin layered samples to the previous bulk studies of TaSe₂ and indicate the existence of ICDW phase in the atomic thin layers of 2H-TaSe₂. The low Raman signal-to-noise ratio at low temperatures may be an obstacle for detailed “error-free” analysis of the data. Meanwhile, a narrow band Notch filter could benefit to the direct observation of low-wavenumber peaks related to the CDW superlattice.

Methods

Sample fabrication. Samples are isolated by scotch-tape technique and deposited on 300 nm SiO₂/Si substrates. For suspended samples these substrates were chemically etched with grid of holes ~3 μm in diameter. Then suitable samples are searched and located under an optical microscope.

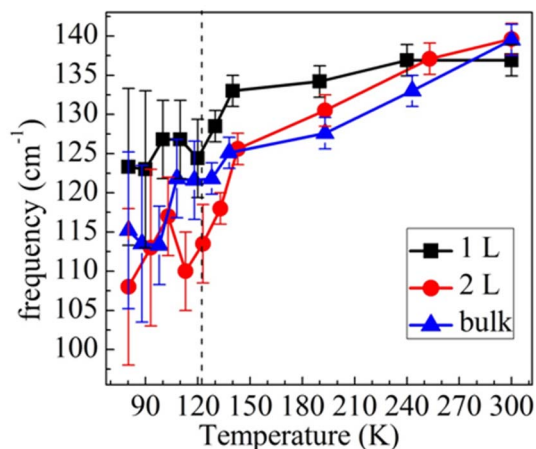


Figure 8 | Observation of phase transition through sudden softening of two-phonon peaks. In bulk samples, upon ICDW phase transition temperatures, the broad peak due to 2-phonon process broadens, softens in energy and intensity also decreases (black dashed line represents ICDW phase transition point for bulk at ~123 K). This broad peak is fitted by Lorenz function to obtain frequency corresponding to the center of the peak (the data below 120 K has huge error due to severely diminished Raman signal). As temperature approaches to Peierls phase transition point, 2-phonon peak for single layer and bilayer samples softens similar to bulk sample.



Layer number identification. Isolated samples are mapped by AFM in tapping mode and white light contrast spectroscopy. We conducted the white light contrast spectroscopy using WITEC CRM200 system with a 150 lines/mm grating and a 100x objective lens.

Raman spectroscopy. Micro-Raman measurements have been conducted by a Renishaw system with 50x, 80x long working distance objective lens and a 2400 lines/mm grating under 532 nm laser excitation. The laser power is regulated below 3 mW and the laser spot size is $\sim 1.5 \mu\text{m}$. For room and low temperature Raman measurements, a temperature-controlled evacuated chamber cooled by liquid nitrogen is used for preventing the potential oxidation and local heating of the TaSe₂ samples under laser irradiation. The TaSe₂ samples are very easily burned if Raman measurements carried out in oxygen-rich ambient. Detailed description of laser burning related effects during Raman measurements can be found in the Supplementary Information (see Supplementary Fig. S2).

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Author contributions

T.Y. and C.X.C. initiated the project, conceived and designed the experiments; P.H. conducted Raman measurements, analyzed the data; C.Y.Q. performed sample exfoliation and preparation of suspended samples; All author contributed to contrast spectroscopy and AFM measurements, and discussed the results, commented on the manuscript.

Additional information

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