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THE MANUFACTURING OF Cu-Al₂O₃ COMPOSITE PRODUCTS: STUDY OF PROCESS PARAMETERS, STRUCTURE AND MECHANICAL PROPERTIES

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ABSTRACT: The purpose of this work is development of a method of producing the Cu-Al₂O₃ composite product by using the selective laser melting technology. The work includes 3 stages: production of composite powder, selective laser melting, mechanical tests of the obtained material. Method of production of composite powder has 2 stages: 1) production of metal powders by atomization of the liquid melt (Cu) in the gas jet, 2) modification of the surface of the metal particles of the second phase (Al₂O₃ nanopowder) in the planetary mixer. Metal powder was produced on URM-001 gas atomizer (patented by South-Ural State University). The surface modification of metal particles is produced in a planetary mixer KURABO Mazerustar kk250 with 5 different modes. Acceptable mode of modification is selected on the basis of the analysis of composite powders on a scanning electron microscope JSM-6400LV. The roundness of powders of Cu and Cu-Al₂O₃ determined using an optical analyzer OCCHIO 500. The conclusion is made about the suitability of powders for use in selective laser melting. The specimens of Cu and Cu-Al₂O₃ were made on the SINTERSTATION® Pro DM125 SLM machine by using different modes (power output, point distance, exposure time, hatch space). Surface end volume energy densities were used for describe of SLM modes. The structure of the composite material (Cu-Al₂O₃) using the chemical mapping was investigated on scanning electron microscopy. The mechanical tests on samples of Cu and Cu-Al₂O₃ made on the GLEEBLE 3800 Thermal-Mechanical Testing system.

KEYWORDS: Metal matrix composite, Cu-Al₂O₃, additive technologies, selective laser melting.

INTRODUCTION

Nowadays the additive technologies (Selective Laser Sintering/Melting (SLS/SLM), Electron Beam Melting (EBM), Direct Metal Deposition (DMD)) are used in high-tech industries, such as medical industry, machine manufacturing, aerospace industry, Wohlers (2005), Liu (2014). Further development of additive technologies involves the range expansion of materials used. One of the most promising directions is the creation of products from composite materials. Copper have high electrical and thermal conductivity, high corrosion resistance. Al₂O₃ is well-known hard material with high melting temperature. To combine of these physical properties in this work the method of producing the Cu-Al₂O₃ composite product by using the SLM technology is developed. The work includes 3 stages: production of composite powder, selective laser melting, mechanical tests of the obtained material.

RESEARCH METHODOLOGY

Method of production of composite powder has 2 stages: 1) production of metal powders by atomization of the liquid melt (Cu) in the gas jet, 2) modification of the surface of the metal particles of the second phase (Al₂O₃ nanopowder) in the planetary mixer.

Molten metal (copper) atomization is performed using the URM-1 (patented by South-Ural State University) apparatus (Figure 1,a), Safonov et al. (2011). Melting the metal and a subsequent superheat of the liquid melt at 200-250 °C above the melting temperature were performed using the induction melting furnace; further, as a result of excessive pressure in the sealed cavity of the induction furnace, the molten metal is supplied through the melt flow channel to the central aperture of the spray nozzle. The molten metal jet contacts with the gas jet, which results in atomization of the melt (Figure 1,b).

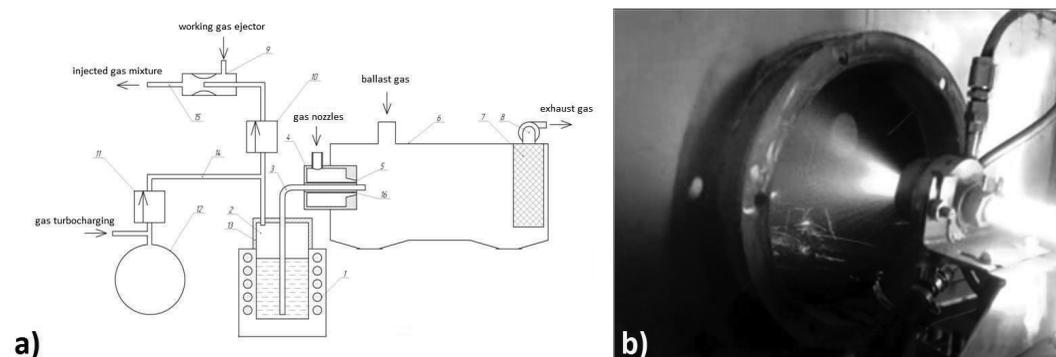


Figure 1. a - scheme of URM-1: 1 – melting furnace, 2 – working cavity, 3 – melt flow channel, 4 – spray nozzle, 5 – gas outlet channel, 6 – bunker, 7 – filter, 8 – ventilator, 9 – ejector, 10,11 – valves, 12 – tank of compressed gas, 13 – cap, 14,15 – channels inlet and outlet of the gas, 16 – central through hole; b - atomization process on URM-1

The surface modification of metal particles is produced in a planetary mixer KURABO Mazerustar kk250 with 5 different modes (Table 1). For characterization of surface modification quality scanning electron microscopy was used.

Table 1. The modes of modification

Mode	Rotation speed of platform, rpm	Rotation speed of container, rpm	Weight of the mixture, g	The proportion of the Cu powder in the mixture, %	The proportion of the Al ₂ O ₃ powder in the mixture, %	Processing time, min
1	1350	1230	10	95	5	2
2	1350	1230	10	95	5	5
3	1350	1230	10	95	5	10
4	1350	1230	20	95	5	15
5	1350	1230	40	95	5	15

The 4x4x4 cubic specimens of Cu and Cu-Al₂O₃ was made on the SINTERSTATION® Pro DM125 SLM machine by using different modes (power output, point distance, exposure time, hatch space). Surface (eq. 1) end volume (eq. 2) energy densities were used for describe of SLM modes (Table 2,3).

$$E_v = \frac{P \cdot ET}{PD \cdot HS \cdot h} \quad (1)$$

$$E_s = \frac{P \cdot ET}{PD \cdot HS} \quad (2)$$

Where P is laser power; ET is exposure time; PD is point distance; HS is hatch space; h is layer thickness.

Structure of fabricated specimens was determined by scanning electron microscopy. JSM-6400LV electron microscope was used. Specimens were polished by using standard procedures and were etched with a FeCl₃, HCl and ethanol mixture.

For tensile tests three cylindrical specimens (diameter – 6 mm, length – 120 mm) was fabricated. Tensile tests were carried out on the GLEEBLE 3800 Thermal-Mechanical Testing system at 573 K temperature.

RESULTS AND DISCUSSION

Obtained copper powder is characterized by an average particle size of about 50 micron, Lykov et al. (2013). The powder was seen to be generally spherical under scanning electron microscopy, as shown in Figure 2,a.

Acceptable mode of modification is selected on the basis of the analysis of composite powders on a scanning electron microscopy (Figure 2). The most effective modification without substantially changing the shape of the particles was achieved in mode 2 (Figure 2,c). In comparison with mode 1 (Figure 2,b) Cu particles for mode 2 were covered more uniformly by Al₂O₃ nanopowder. In mode 3 (Figure 2,d) many Cu particles were mechanically deformed due to high processing

duration (10 min), that may negatively influenced on powder flowability. In modes 4 and 5 (Figure 2,e and f) copper particles did not cover by ceramic nanopowder. So, mode 2 was chosen for produce of composite powder for SLM.

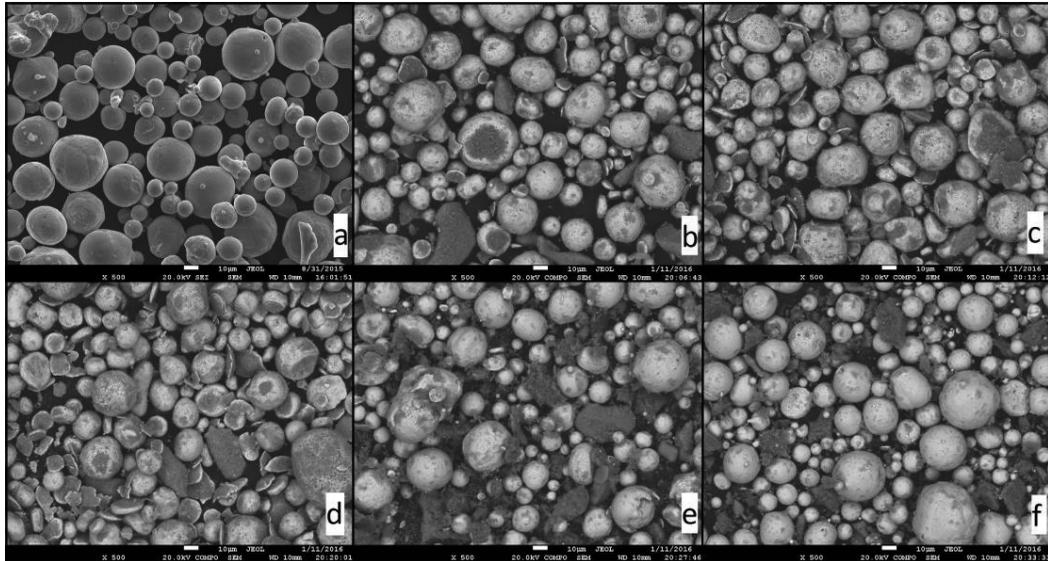


Figure 2. Images of powders (raster electron microscope JSM-6400LV): a – Cu, b – mode 1, c – mode 2, d – mode 3, e – mode 4, f – mode 5

SLM process parameters were chosen based on preliminary experiments of selective laser melting of pure copper powder. SLM process parameters presents in Table 2.

Table 2. SLM modes (Cu-Al₂O₃)

Mode	P, W	ET, μ s	PD, μ m	HS, μ m	h, μ m	Diameter of the laser spot, μ m	Base plate preheating, °C	E _s , J/mm ²	E _v , J/mm ³
1	200	400	50	50	50	35	140	32	640
2	200	300	50	50	50	35	140	24	480
3	200	250	50	50	50	35	140	20	400
4	200	200	35	35	50	35	140	23	457

The structure of the SLM processed composite material structure presents in Figure 3 in chemical mapping diagrams for Cu, Al and O. Analysis of chemical maps shown that second face is distributed uniformly for all used SLM process modes. Various energy densities does not influence on second face distribution and morphology significantly.

Results of tensile tests presents in Table 3. For fabrication of tensile test specimens SLM mode 1 with highest energy density was chosen. Ultimate tensile strength of SLM fabricated composite material was 110 MPa at 573 K test temperature that less in comparison with cast pure copper. That is explained by porosity of SLM produced material due to low laser radiation absorption and high thermal conductivity of copper, Lykov et al. (2016).

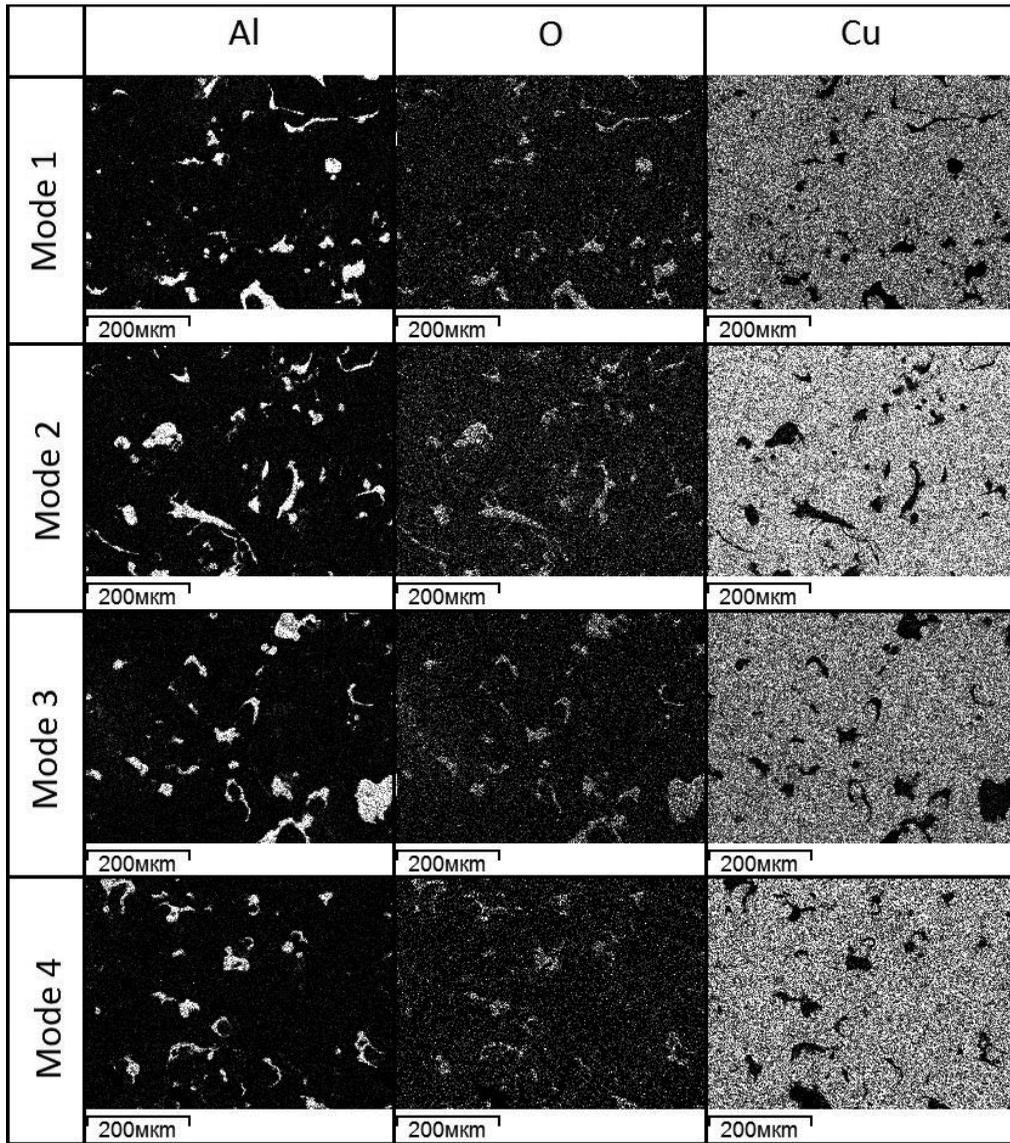


Figure 3. Chemical mapping of SLM produced Cu-Al₂O₃ composite material

Table 4. Ultimate tensile strength - at 573 K

Material	UTS, MPa
Cu-Al ₂ O ₃ (SLM)	110
Cu (cast)	150

CONCLUSIONS

In this paper the development of a method of producing the Cu-Al₂O₃ composite product by using Selective Laser Melting Technology is presented. The mode for producing the composite

powder by planetary mixer was determined. 4x4x4 mm specimens for analysis of composite material structure were fabricated using four SLM process modes with different energy densities. Analysis composite material structure shows, that second phase (Al_2O_3) is distributed uniformly for all used SLM process modes. Various energy densities does not influence on second face distribution and morphology significantly. Tensile tests at 573 K test temperatures show that UTS of SLM fabricated composite material (110 MPa) less in comparison with cast pure copper (150 MPa), that is explained by porosity of SLM produced material due to low laser radiation absorption and high thermal conductivity of copper.

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