<table>
<thead>
<tr>
<th><strong>Title</strong></th>
<th>Observation of clusters in Re60Fe30Al10 alloys and the associated magnetic properties</th>
</tr>
</thead>
<tbody>
<tr>
<td><strong>Author(s)</strong></td>
<td>Kong, H. Z.; Ding, Jun; Dong, Zhili; Wang, L.; White, Timothy John; Li, Y.</td>
</tr>
<tr>
<td><strong>Date</strong></td>
<td>2002</td>
</tr>
<tr>
<td><strong>URL</strong></td>
<td><a href="http://hdl.handle.net/10220/8296">http://hdl.handle.net/10220/8296</a></td>
</tr>
<tr>
<td><strong>Rights</strong></td>
<td>© 2002 Institute of Physics Publishing. This is the author created version of a work that has been peer reviewed and accepted for publication by Journal of Physics D: Applied Physics, Institute of Physics Publishing. It incorporates referee's comments but changes resulting from the publishing process, such as copyediting, structural formatting, may not be reflected in this document. The published version is available at: [DOI: <a href="http://dx.doi.org/10.1088/0022-3727/35/5/302">http://dx.doi.org/10.1088/0022-3727/35/5/302</a> ]</td>
</tr>
</tbody>
</table>
Observation of clusters in Re$_{60}$Fe$_{30}$Al$_{10}$ alloys and the associated magnetic properties

Kong H Z$^1$, Ding J$^{1,4}$, Dong Z L$^2$, Wang L$^3$, White T$^2$ and Li Y$^1$

$^1$ Materials Science Department, National University of Singapore, Singapore 119260
$^2$ Environmental Technology Institute, Innovation Center, Nanyang Technology University, Singapore 637723
$^3$ Physics Department, National University of Singapore, Singapore 119260
$^4$ Author to whom correspondence should be addressed. E-mail: masdingj@nus.edu.sg

ABSTRACT

Magnetic properties and microstructure of melt-spun ribbons of RE$_{60}$Fe$_{30}$Al$_{10}$ alloys with RE = Nd, Sm, Dy, Gd and Y were studied in detail. High coercivity values in the range of MA m$^{-1}$ were observed at low temperatures for amorphous ribbons. Presence of Fe-rich clusters and nanoscale rare-earth crystallites in the amorphous matrix in the ribbons were revealed by high-resolution transmission electron microscopy studies. The magnetic transition temperatures were estimated experimentally and compared with fitting results based on the cluster ferromagnetism model (Wang L et al 2001 Phys. Rev. B 64 214410). Possible mechanisms for the magnetic behaviour observed due to the presence of Fe-rich magnetic clusters are discussed.

1. INTRODUCTION

The structural disorder in amorphous magnetic materials has generated a number of interesting magnetic properties such as high Curie temperature, magnetostriction and induced anisotropy, which can be tailored for different applications [2]. The amorphous rare-earth–transition-metal films deposited by magnetron sputtering, e.g. GdTb–Fe, Tb–FeCo and Dy–FeCo, are promising candidates for magneto-optical information storage due to their large perpendicular anisotropy and high Curie temperature [2]. Different origins were proposed to account for the observed anisotropy, e.g. structural inhomogeneities [3], stress-induced anisotropy [4], dipolar interactions [5] and exchange anisotropy [6].

More recently, high coercivity combined with outstanding glass forming ability was observed for (Nd, Pr)–Fe–Al alloys [7]. The hard magnetic properties were suggested to be due to magnetic exchange coupling interaction among the Fe-rich clusters with large random magnetic anisotropy in the relaxed amorphous samples [7]. Fine dark and bright patches of 1–2 nm in diameter spread uniformly in the amorphous Pr$_{90}$Fe$_{10}$ alloy were observed under high-resolution transmission electron microscope (HRTEM) [8]. The fine patches were correlated to ferromagnetic clusters [8]. Results from our previous studies on magnetic behaviour of the amorphous ribbons of rare-earth–iron (RE–Fe)–based alloys [9–12] also suggested presence of Fe-rich magnetic clusters in these ribbons.
The complex magnetic behaviour observed by Ortega-Hertogs et al [13] for bulk glassy rods (2.7mm in diameter) of Nd$_{60}$Fe$_{30}$Al$_{10}$ alloy shows that in a temperature region where the ribbon is paramagnetic, the hysteresis loop is well simulated by a Stoner–Wohlfarth type of loop, supporting the existence of an ensemble of ferromagnetic entities with uniaxial anisotropy, whose easy axes are oriented at random, and remain magnetically isolated by a paramagnetic matrix, the Nd-rich phases [13]. At 2.6 K, the Nd-rich matrix into which the Fe-rich Fe-Nd entities are embedded is antiferromagnetic. The Fe-Nd entities become magnetically coupled to the surrounding Nd-rich matrix via Fe-Nd and Nd–Nd exchange interaction, and the Nd-rich matrix is expected to develop a large crystal field anisotropy [13]. A magnet that is structurally glassy but magnetically granular at the nanoscale and exhibits good isotropic hard magnetic properties is suggested [13].

Recently, we proposed a model (named the cluster ferromagnetism model) to account for the unique magnetic behaviour of these inhomogeneous amorphous RE–Fe-based ribbons [14]. The magnetic units in this model are magnetic clusters with random distribution of magnetic easy directions, and with strong interaction between the clusters. The model can be expressed in the Hamiltonian here:

$$E = -D \sum_i [\vec{m}_i(T) \cdot \vec{m}_i(T)] - \frac{1}{2} \sum_{i \neq j} J_{ij} [\vec{m}_i(T) \cdot \vec{m}_j(T)] - H \sum_i \vec{m}_i(T)$$

(1)

where $m(T)$ is the magnetic moment of the cluster at temperature $T$, $D$ is the magnetic anisotropy of a cluster, $J_{ij}$ is the exchange interaction coefficient between the $i$th cluster and $j$th cluster, $n_i$ is a random unit vector representing the local axis of easy magnetization direction, $H$ is the applied magnetic field. $m(T)$ can be expressed as $m_s f(T/T_{c\text{cluster}})$, where $m_s$ is the saturation magnetization of the cluster phase at zero temperature or total magnetic moment of a cluster, and $T_{c\text{cluster}}$ is the Curie temperature of the magnetic clusters. The sample is paramagnetic at temperatures above $T_{c\text{cluster}}$ [14]. The Monte Carlo method has been used to simulate magnetization curves after zero-field cooling and magnetic hysteresis loops at different temperatures [1]. The simulation results showed the presence of another two critical temperatures, $T_{\text{block}}$ and $T_{\text{system}}$. $T_{\text{block}}$ is the blocking temperature due to anisotropy energy, while $T_{\text{system}}$ is the freezing temperature due to interaction between the clusters [1]. At temperatures below $T_{\text{system}}$, remanence increases significantly due to cluster interactions that lead to ferromagnetic coupling between clusters. However, the coercivity remains low due to relatively high thermal energy in comparison to magnetic anisotropy energy, which results in easy flip of magnetization vectors. High coercivity requires the freezing of the ferromagnetic state at temperatures below $T_{\text{block}}$ [1]. The Monte Carlo simulation work suggested that if $T_{\text{system}}$ is higher than $T_{\text{block}}$, valid for most amorphous RE and transition metal alloys, the magnetic behaviour as a function of temperature could be divided into four periods [1]:
In this paper, we have performed systematic structural and magnetic studies on amorphous ribbons of RE$_{60}$Fe$_{30}$Al$_{10}$ alloys with RE = Y, Nd, Sm, Gd and Dy. The unique magnetic behaviour and results of microstructure studies that lead to observation of clusters in these ribbons are summarized.

2. EXPERIMENTAL PROCEDURES

Ingots of nominal compositions RE$_{60}$Fe$_{30}$Al$_{10}$ with RE = Nd, Y, Sm, Gd and Dy were prepared by arc-melting the pure constituents (purity 99.5%) under argon atmosphere. Small pieces of the ingots were then re-melted and melt-spun into ribbons using single roller (copper) chill block melt-spinner at wheel surface speeds of 5 and 30 m s$^{-1}$. Part of the ingots was also used for casting into cylinders of diameter 1 mm and length 30 mm by low-pressure copper mould chill casting. The phases present in the samples were determined by x-ray diffraction (XRD) technique with Cu K$_\alpha$ radiation. Magnetic measurements were carried out by vibrating sample magnetometer (VSM) at different temperatures from 5 to 700 K. Microstructure studies were carried out using TEMs operating at 100 and 300 keV.

3. RESULTS

3.1. Structural studies

XRD patterns of ribbon melt-spun at 30 m s$^{-1}$ (figure 1) of Nd$_{60}$Fe$_{30}$Al$_{10}$ alloy and Sm$_{60}$Fe$_{30}$Al$_{10}$ alloy show only large amorphous hump, while those of Dy$_{60}$Fe$_{30}$Al$_{10}$ alloy and Gd$_{60}$Fe$_{30}$Al$_{10}$ alloy show a small diffraction peak corresponding to their respective RE phase. Decreasing the melt-spinning speed to 5 m s$^{-1}$ resulted in the formation of respective RE phase with preferred orientation along the (002) and small amount of RE-Al phases, e.g. Al$_2$Gd and AlNd$_3$ phase.

Microstructure of the ribbon of Dy$_{60}$Fe$_{30}$Al$_{10}$ alloy melt-spun at 30 m s$^{-1}$ was studied by TEM operating at 100 keV. TEM bright field images of the ribbon melt-spun at 30 m s$^{-1}$ (figure 2(a)) shows a generally featureless contrast typical of amorphous phase. Glassy nature of the ribbon was also confirmed by the observation of halo ring in the diffraction pattern (figure 2(b)). Upon careful studies, the micrograph revealed the presence of pastel grey shades, indicating inhomogeneity of the amorphous phase and the possible existence of clusters in the ribbon. TEM micrographs of the ribbon melt-spun at 5 m s$^{-1}$ show the presence of large crystallites (about 100 nm) corresponding to the RE phase (in this case, Dy phase) existing in the amorphous matrix.

Ribbon of Dy$_{60}$Fe$_{30}$Al$_{10}$ alloy melt-spun at 5 ms$^{-1}$ was further studied by HRTEM operating at 300keV. HRTEM image (figure 3) of the region classified as amorphous
under conventional TEM shows the presence of small patches of fringes around 3.4 nm in diameter scattered in the amorphous matrix. The spacing of the fringes was around 3.2 Å, which is very close to the $d$-spacing of the (100) plane of the Dy phase (3.11 Å). The precipitation of Dy phase in the amorphous matrix is highly probable since Dy make up a large proportion (more than half) of the alloy. The presence of Dy phase in the ribbon was also supported by the XRD results. The presence of circular white patches around 2.5 nm in diameter were also noticed. There are two possibilities. The first possibility is that these may be due to thickness variation inherent from the melt-spinning process or the ion-milling process during the TEM sample preparation. The size of these patches seemed too small to be caused by un-precision fabrication technique. Difference in atomic weight of the constituting atoms of a feature is another very likely source of contrast in an HRTEM image. The second possibility is hence a variation in the ratio of atoms with high atomic weight (Dy) and atoms with low atomic weight (Fe). These patches were whitish, an indication of possible higher content of the atoms with lower atomic weight such as Fe, as compared to the general amorphous matrix which consists of a large amount of heavy Dy. X-ray energy dispersive spectroscopy (EDX) analysis indicated that the whitish region in the ribbon was Fe-rich. Careful studies of the circular white patches show that they were agglomerates of atoms without any ordered arrangement, i.e. amorphous clusters. Presence of Fe-rich amorphous clusters in the ribbons is concluded.

3.2. Magnetic studies

Hysteresis loops at different temperatures for the ribbons of Dy$_{60}$Fe$_{30}$Al$_{10}$ alloy melt-spun at 30 m s$^{-1}$ are shown in figure 4. At room temperature (290 K), the magnetization curve of the ribbon was a straight line, typical for paramagnetic material. As the temperature decreased to 170 K, the ribbon became superparamagnetic. Observation of superparamagnetism suggested the presence of clusters in the amorphous ribbon. When the temperature decreased to below 140 K, the ribbon exhibited ferromagnetism with low coercivity ($H_c$) (38.84 kA m$^{-1}$ at 80K). Coercivity of the ribbon increased rapidly as temperature decreased further, reaching a high coercivity of 1992.4 kA m$^{-1}$ at 4.2K. The amorphous ribbons of RE$_{60}$Fe$_{30}$Al$_{10}$ alloys with RE as Nd and Sm also exhibit similar series of magnetic behaviour transitions as temperature decreases, i.e. (a) above its Curie temperature, the ribbon is paramagnetic; (b) below this Curie temperature, the ribbon is superparamagnetic; (c) at still lower temperatures, the ribbon becomes ferromagnetic; (d) as temperature further decreases, the ribbon transforms from soft magnetic to hard magnetic. This phenomenon agrees well with the expectation of the cluster ferromagnetism model characterized by three critical transition temperatures, $T_{\text{block}}$, $T_c^{\text{system}}$ and $T_c^{\text{cluster}}$ (as elaborated in section 1).

Temperature dependence of inverse susceptibility ($1/\chi$) of the ribbon of Dy$_{60}$Fe$_{30}$Al$_{10}$ alloy melt-spun at 30 ms$^{-1}$ is shown in figure 5(a). It is noticed that instead of a sharp kink signifying the transition from ferromagnetic to paramagnetic state, the inverse susceptibility traced out a gradual bowed curve, extended for about 100 K, a unique characteristic of superparamagnetic state due to the presence of clusters. $T_c^{\text{system}}$ being the temperature at which the sample transforms from ferromagnetic to superparamagnetic
state is characterized by the temperature at which inverse susceptibility deviates from zero (as shown in figure 5(a)). $T_c^{\text{cluster}}$ being the transition temperature from superparamagnetic state is, on the other hand, the lowest temperature at which the inverse susceptibility value follows the linearity defined by the Curie–Weiss law.

According to the results of our Monte Carlo simulation in [1], $T_c^{\text{system}}$ can also be characterized by the temperature at which the remanence increases, as illustrated in figure 5(b) for the ribbon of Dy$_{60}$Fe$_{30}$Al$_{10}$ alloy melt-spun at 30 ms$^{-1}$. $T_{\text{block}}$, on the other hand, is defined experimentally by the peak of typical cusp in the ZFC curves near the temperature at which the bifurcation of the ZFC and FC curves occurs, as shown in figure 5(d). According to the results of our Monte Carlo simulation, $T_{\text{block}}$ can also be characterized by the temperature at which the coercivity value starts to increase rapidly (figure 5(c)). The increase of coercivity at temperatures below $T_{\text{block}}$ is also expected according to the equation below for spherical single-domain particles [15]:

$$H_c = \left( \frac{2K}{M_s} \right) \left[ 1 - \left( \frac{T}{T_{\text{block}}} \right)^{1/2} \right]$$ (2)

The values of $T_{\text{block}}$, $T_c^{\text{system}}$ and $T_c^{\text{cluster}}$ as estimated for the ribbon of Dy$_{60}$Fe$_{30}$Al$_{10}$ alloy and ribbons of other RE$_{60}$Fe$_{30}$Al$_{10}$ alloys are tabulated in table 1 and plotted for different RE elements in figure 6. $T_c^{\text{system}}$ and $T_c^{\text{cluster}}$ of the ribbons melt-spun at 5 m s$^{-1}$ were always higher than those of the ribbons of the same alloy melt-spun at 30 m s$^{-1}$. Ribbon of Nd$_{60}$Fe$_{30}$Al$_{10}$ alloy gave the highest $T_{\text{block}}$, $T_c^{\text{system}}$ and $T_c^{\text{cluster}}$ among all the ribbons melt-spun at 30 m s$^{-1}$. $T_c^{\text{cluster}}$ of ribbon melt-spun at 5 m s$^{-1}$ of Gd$_{60}$Fe$_{30}$Al$_{10}$ alloy was the highest among all the ribbons melt-spun at the same speed.

Similar sequence of magnetic transition were also observed as temperature decreases when Al in Nd$_{60}$Fe$_{30}$Al$_{10}$ alloy is replaced by Si to form Nd$_{60}$Fe$_{30}$Si$_{10}$ alloy and when Fe in Sm$_{60}$Fe$_{30}$Al$_{10}$ is replaced by Co to form Sm$_{60}$Co$_{30}$Al$_{10}$ alloy (table 1). It is noted that the coercivity for the ribbon of Nd$_{60}$Fe$_{30}$Al$_{10}$ alloy melt-spun at 5 m s$^{-1}$ decreased from $\sim$3229kAm$^{-1}$ at 40K to $\sim$1262K at 5K. Decrease of coercivity was also observed for the ribbon of the same alloy melt-spun at 30 m s$^{-1}$. The decrease in coercivity was due to antiferromagnetic transition of the Nd phase (Néel temperature of Nd = 19K [16]).

$M(T, H)/f (T/T_c^{\text{cluster}})$ versus $f (T/T_c^{\text{cluster}})(H + \alpha M (T, H))/T$ plots were constructed based on the magnetization curves of these amorphous RE$_{60}$Fe$_{30}$Al$_{10}$ ribbons taken at temperatures in the superparamagnetic regime (where anisotropy energy term $D \sum_i [\vec{n}_i \cdot \vec{m}_i(T)]^2$ in Hamiltonian for cluster ferromagnetic model can be omitted). By molecular field approximation, the magnetization at temperature $T$ and magnetic field $H$ in the superparamagnetic regime can be expressed as the modified Langevin function with molecular field ($\alpha M$) incorporated below [14]:

$$M(T, H)/f (T/T_c^{\text{cluster}}) = f (T/T_c^{\text{cluster}})(H + \alpha M (T, H))/T$$
where \( N \) is the cluster density, \( \alpha \) is the molecular field coefficient and \( k_B \) is the Boltzmann constant. \( m_s f(T/T_c^{\text{cluster}}) \) is the magnetic moment at temperature \( T (= m(T)) \). A universal curve should be obtained for each ribbon for appropriate values of \( T_c^{\text{cluster}} \) and \( \alpha \). The universal curve for the ribbon of Dy_{60}Fe_{30}Al_{10} alloy melt-spun at 30 m s\(^{-1}\) is shown in figure 7. \( T_c^{\text{cluster}} \) and \( \alpha \) thus obtained, and the \( T_c^{\text{system}} \) and \( m_s \) calculated according to the equations below are listed in table 1.

\[
T_c^{\text{system}} = N m_s f^2 \left( \frac{T_c^{\text{system}}}{T_c^{\text{cluster}}} \right) \alpha \frac{3k_B}{M_0} \quad (4)
\]

\[
m_s = \frac{3k_B}{M_0} \left( \frac{M(T, H)/f(T/T_c^{\text{cluster}})}{f(T/T_c^{\text{cluster}})(H + \alpha M(T, H))/T} \right) \quad (5)
\]

Good agreement between the critical transition temperatures obtained by fitting and estimated experimentally shows that the magnetic behaviour of these amorphous RE_{60}Fe_{30}Al_{10} ribbons can be well described by the cluster ferromagnetism model.

**4. DISCUSSIONS**

**4.1. Effect of RE elements**

\( T_c^{\text{cluster}} \) of ribbon of Gd_{60}Fe_{30}Al_{10} alloy melt-spun at 5 m s\(^{-1}\) was the highest among all the ribbons melt-spun at the same speed. Gd is an S-state RE ion, hence its anisotropy energy is not expected to be high as compared to other non-S-state RE element ions. The high \( T_c^{\text{cluster}} \) observed might be due to the presence of Gd with high Curie temperature (293.4 K). Variation of Curie temperature of RE\(_x\)Fe\(_y\) compounds (e.g. RE\(_2\)Fe\(_{17}\) and ReRe\(_2\)) with RE ion involved also show a maximum value when RE = Gd [17] due to the effect of spin of RE ion (spin of Gd\(^{3+}\) ion = 7/2, largest among all the RE\(^{3+}\) ions).

\( T_c^{\text{system}} \) of the ribbons melt-spun at 5 and 30 m s\(^{-1}\) show similar dependence on RE elements present in the alloy, with maximum when RE = Nd and minimum when RE = Y. According to our Monte Carlo simulation results [1], low \( T_c^{\text{system}} \) is an indication of weak interactions between the c particles. Y is a non-magnetic atom and hence might not contribute any magnetic moment to the interaction.

High \( T_c^{\text{system}} \) of Nd\(_{60}\)Fe\(_{30}\)Al\(_{10}\) ribbons is an indication of the very strong interactions that resulted in ferromagnetic coupling between the magnetic clusters. It is noticed from table 1 that the \( \alpha \) value that signifies the strength of the interaction between the magnetic clusters of Nd\(_{60}\)Fe\(_{30}\)Al\(_{10}\) ribbon is also the highest. Coupling constant \( J_{RE-Fe} \) between the RE spin and Fe spin in RE\(_2\)Fe\(_{14}\)B compounds is highest and hence the interaction is
strongest when $\text{RE} = \text{Nd}$ [18]. If similar RE–Fe interaction exists in our amorphous $\text{RE}_{60}\text{Fe}_{30}\text{Al}_{10}$ ribbons and contributes to the interaction between the Fe-rich clusters, $T_c^{\text{system}}$ that reflects the strength of interaction of the $\text{Nd}_{60}\text{Fe}_{30}\text{Al}_{10}$ ribbon is expected to be higher than other ribbons.

4.2. Effect of melt-spinning speed

$T_c^{\text{system}}$ and $T_c^{\text{cluster}}$ of the ribbons of all alloys melt-spun at 5 m s$^{-1}$ were higher than those melt-spun at 30 m s$^{-1}$. $T$ system corresponds to the strength of magnetic interaction between the magnetic clusters; higher the $T_c^{\text{system}}$, stronger the interaction. Cooling rate experienced by the ribbons melt-spun at 5 ms$^{-1}$ is lower than the ribbons melt-spun at 30ms$^{-1}$. Formation of larger cluster or higher density of cluster at the lower cooling rate might decrease the average separation distance between the Fe-rich clusters which were scattered in the amorphous matrix, and hence promote the interactions between the magnetic clusters. Another possibility is that the RE crystallites of about 3 nm formed only at low melt-spinning speed of 5ms$^{-1}$ might act as some kind of medium for the interaction between the Fe-rich clusters. Furthermore, concentration of Fe in the remaining amorphous phase might be enriched as RE phase precipitated out from the amorphous matrix and hence facilitating formation of the magnetic Fe-rich clusters. $T_c^{\text{system}}$ is determined mainly by the composition of the clusters. Presence of a certain amount of RE atoms and Al atoms in the Fe-rich clusters are expected. Formation of RE crystallites at low melt-spinning speed (5 m s$^{-1}$) might decrease the amount of RE atoms in the clusters, leaving the clusters rich in an element of high Curie temperature—Fe (1040.2 K), thus elevating the Curie temperature of the whole cluster.

4.3. RKKY interactions

The Monte Carlo simulation results show that in order to have high coercivity, the anisotropy of the cluster system must be high [1]. There are two possibilities. The first possibility is that the clusters might be hard magnetic themselves. The second possibility is that there is some interaction between the RE small crystallites and the Fe-rich clusters that resulted in high anisotropy. From the magnetic behaviour and high $T_c^{\text{system}}$ observed, strong interaction between the clusters is suggested. If the interaction existing in the system was dipole interaction, molecular field coefficient $\alpha$ should be equal to, and not greater than, 1. However, according to our Langevin fitting results, $\alpha$ for the $\text{RE}_{60}\text{Fe}_{30}\text{Al}_{10}$ ribbons are much greater than 1 (as listed in table 1) suggesting that RKKY-like interaction [19–21] might be present. Hence, the interactions between these amorphous Fe-rich magnetic clusters might involve certain degree of indirect exchange interaction mediated by the surrounding RE crystallites.

4.4. Coercivity mechanism

Table 1 shows high coercivity values (well above 1000 kA m$^{-1}$) were observed for ribbons based on Nd, Sm, Dy melt-spun at 30 m s$^{-1}$. Coercivity values of Y- and Gd-based ribbons remain low (at below 100 kA m$^{-1}$) at temperatures down to 5 K. Simulated
hysteresis loops using the Monte Carlo method based on cluster ferromagnetism model shows that coercivity is determined mainly by the magnetic anisotropy energy of the cluster phase at low temperatures [1]. Generally, the main contribution to magnetic anisotropy in permanent magnet materials like RE–Fe–B and RE–Co is due to the single-ion crystal field-induced anisotropy of the magnetic RE atoms which arises as a consequence of the interaction between the electrostatic field of the charges surrounding the 4f electrons and the asymmetric charge-cloud of these 4f electrons [22]. The difference in shape of the 4f charge-cloud is reflected by the sign of the second-order Stevens factor ($\alpha_J$) [23,24]. This crystal field-induced anisotropy is absent in materials based on Gd that has a zero $\alpha_J$ [22]. It was noticed that coercivity of the largely amorphous RE$_{60}$Fe$_{30}$Al$_{10}$ ribbon based on Gd with zero $\alpha_J$ remains low. In contrast, the $\alpha_J$ of the RE material on which the ribbons with high coercivity are based (e.g. Nd, Sm and Dy) are none zero (either positive or negative). Results of amorphous (Gd$_{0.75}$RE$_{0.25}$)$_{19}$Co$_{81}$ films based on the single-ion anisotropy model [25] also show an increase of uniaxial anisotropy constant $K_u$ and hence anisotropy energy with increasing $\alpha_J$ of RE (either positively or negatively). A relationship between the value of $\alpha_J$ (either zero or non-zero) and the magnitude of coercivity at low temperature is suggested.

In addition, considering the large magnetostriction observed for RE–Fe compounds [26], magnetostriction may contribute to the magnetic anisotropy and hence high coercivity values observed for these ribbons of RE$_{60}$Fe$_{30}$Al$_{10}$.

5. CONCLUSIONS

Ribbon of Dy$_{60}$Fe$_{30}$Al$_{10}$ alloy melt-spun at 5 m s$^{-1}$ contained Dy crystallites of diameter around 3 nm, and Fe-rich clusters with diameter of about 2.5 nm. Amorphous ribbons of RE$_{60}$Fe$_{30}$Al$_{10}$ alloys were paramagnetic at temperatures above $T_c^{\text{cluster}}$, superparamagnetic at temperatures between $T_c^{\text{system}}$ and $T_c^{\text{cluster}}$, ferromagnetic with low coercivity at temperatures between $T_{\text{block}}$ and $T_c^{\text{system}}$, ferromagnetic with high coercivity at temperatures below $T_{\text{block}}$ (except RE = Gd and Y), a sequence expected from the cluster ferromagnetism model. $T_c^{\text{system}}$, $T_c^{\text{system}}$ and $T_{\text{block}}$ for ribbons of RE$_{60}$Fe$_{30}$Al$_{10}$ alloys estimated from temperature dependence of inverse susceptibility, reduced magnetization and coercivity agreed well with those obtained from fitting by modified Langevin function based on the cluster ferromagnetism model. Nd$_{60}$Fe$_{30}$Al$_{10}$ ribbons exhibit the highest $T_c^{\text{system}}$ among all the ribbons. $T_c^{\text{system}}$ and $T_{\text{cluster}}$ of the ribbons melt-spun at 5 m s$^{-1}$ were higher than those of the ribbon melt-spun at 30 ms$^{-1}$. Ribbons of RE$_{60}$Fe$_{30}$Al$_{10}$ alloys with RE of non-zero second-order Stevens factor ($\alpha_J$) exhibit high coercivity at temperatures below $T_{\text{block}}$. 
REFERENCES

List of Figures

Figure 1  XRD patterns of ribbon of alloys RE$_{60}$Fe$_{30}$Al$_{10}$ with RE = Nd, Sm, Gd and Dy melt-spun at (a) 5 m s$^{-1}$ and (b) 30 m s$^{-1}$.

Figure 2 (a) TEM bright field image and (b) diffraction pattern of selected area of the ribbon of alloy Dy$_{60}$Fe$_{30}$Al$_{10}$ melt-spun at 30ms$^{-1}$.

Figure 3 HRTEM lattice image of the ribbon melt-spun at 5 ms$^{-1}$ of alloy Dy$_{60}$Fe$_{30}$Al$_{10}$ (circle 1: amorphous Fe-rich cluster, circle 2: Dy nuclei).

Figure 4 Hysteresis loops of ribbon of alloy Dy$_{60}$Fe$_{30}$Al$_{10}$ melt-spun 30 ms$^{-1}$ at different temperatures.

Figure 5 Temperature dependence of (a) inverse susceptibility ($1/\chi$), (b) reduced magnetization ($M_r/M_{\text{max}}$) and (c) coercivity ($H_c$) for the ribbon of alloy Dy$_{60}$Fe$_{30}$Al$_{10}$ melt-spun at 30 ms$^{-1}$. Inset (d) shows the ZFC/FC curves of the ribbon.

Figure 6 $T_c^{\text{cluster}}$, $T_c^{\text{system}}$ and $T_{\text{block}}$ for ribbons melt-spun at (a) 5ms$^{-1}$ and (b) 30ms$^{-1}$ of RE$_{60}$Fe$_{30}$Al$_{10}$ alloys. (c) Summary of magnetic transitions found in RE$_{60}$Fe$_{30}$Al$_{10}$ alloys with RE = Sm, Gd and Dy.

Figure 7 $M / f (T / T_c^{\text{cluster}})$ versus $f (T / T_c^{\text{cluster}})(H + \alpha M) / T$ universal curves for the ribbon of alloy Dy$_{60}$Fe$_{30}$Al$_{10}$ melt-spun at 30ms$^{-1}$.
List of Tables

Table 1  $T_{\text{block}}$, $T_c^{\text{system}}$ and $T_c^{\text{cluster}}$ of the ribbons of RE$_{60}$Fe$_{30}$Al$_{10}$ alloys estimated experimentally and by fitting.
Figure 1
Figure 2
Figure 4
Figure 5
Figure 6
Figure 7
Table 1. $T_{\text{bulk}}$, $T_e^{\text{prompt}}$, and $T_e^{\text{cluster}}$ of the ribbons of RE$_{20}$Fe$_{30}$Al$_{10}$ alloys estimated experimentally and by fitting.

<table>
<thead>
<tr>
<th></th>
<th>$v$ (m s$^{-1}$)</th>
<th>$H_e$ at 5 K (or as-stated) (kA m$^{-1}$)</th>
<th>$T_e^{\text{prompt}}$ (K)</th>
<th>$T_e^{\text{cluster}}$ (K)</th>
<th>$T_{\text{max}}$ (K)</th>
<th>$\alpha$ (μm)</th>
<th>$m_e$ ($\mu_0$)</th>
<th>$T_e^{\text{prompt}}$ (K)</th>
<th>$T_e^{\text{cluster}}$ (K)</th>
</tr>
</thead>
<tbody>
<tr>
<td>D$<em>{20}$Fe$</em>{30}$Al$_{10}$</td>
<td>5</td>
<td>232 (at 80 K)</td>
<td>260</td>
<td>413</td>
<td>275</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
</tr>
<tr>
<td></td>
<td>30</td>
<td>1992.4</td>
<td>140</td>
<td>300</td>
<td>65</td>
<td>12</td>
<td>47</td>
<td>135</td>
<td>320</td>
</tr>
<tr>
<td>Gd$<em>{20}$Fe$</em>{30}$Al$_{10}$</td>
<td>5</td>
<td>3.62 (at 290 K)</td>
<td>383</td>
<td>623</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
</tr>
<tr>
<td></td>
<td>30</td>
<td>7.67</td>
<td>260</td>
<td>413</td>
<td>—</td>
<td>21</td>
<td>660</td>
<td>306</td>
<td>500</td>
</tr>
<tr>
<td>Sm$<em>{20}$Fe$</em>{30}$Al$_{10}$</td>
<td>5</td>
<td>2677.95</td>
<td>413</td>
<td>533</td>
<td>215</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
</tr>
<tr>
<td></td>
<td>30</td>
<td>3305.2</td>
<td>260</td>
<td>443</td>
<td>125</td>
<td>18</td>
<td>880</td>
<td>—</td>
<td>630</td>
</tr>
<tr>
<td>Y$<em>{20}$Fe$</em>{30}$Al$_{10}$ [1]</td>
<td>5</td>
<td>1.95</td>
<td>215</td>
<td>—</td>
<td>—</td>
<td>61</td>
<td>600</td>
<td>135</td>
<td>500</td>
</tr>
<tr>
<td></td>
<td>30</td>
<td>6.21</td>
<td>6</td>
<td>355</td>
<td>13</td>
<td>3</td>
<td>150</td>
<td>8</td>
<td>360</td>
</tr>
<tr>
<td>Nd$<em>{20}$Fe$</em>{30}$Al$_{10}$ [1]</td>
<td>5</td>
<td>3239 (at 40 K)</td>
<td>460</td>
<td>570</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
</tr>
<tr>
<td></td>
<td>50</td>
<td>1262 (at 5 K)</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
</tr>
<tr>
<td></td>
<td>30</td>
<td>2645.7 (at 80 K)</td>
<td>390</td>
<td>545</td>
<td>—</td>
<td>475</td>
<td>15</td>
<td>397</td>
<td>520</td>
</tr>
<tr>
<td></td>
<td>30</td>
<td>597 (at 20 K)</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
</tr>
<tr>
<td>Nd$<em>{20}$Fe$</em>{30}$Si$_{10}$</td>
<td>30</td>
<td>2648.2 (at 80 K)</td>
<td>443</td>
<td>533</td>
<td>338</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
</tr>
<tr>
<td>Sm$<em>{20}$Co$</em>{30}$Al$_{10}$</td>
<td>30</td>
<td>1012.8 (at 20 K)</td>
<td>40</td>
<td>50</td>
<td>45</td>
<td>—</td>
<td>—</td>
<td>—</td>
<td>—</td>
</tr>
</tbody>
</table>