

1 Ultra-strong optical harmonic generations in layered platinum
2 disulfide in the mid-infrared

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22
23 **Abstract**

24 Nonlinear optical activities (e.g., harmonic generations) in two dimensional (2D) layered materials
25 have attracted much attention due to the great promise in diverse optoelectronic applications such
26 as nonlinear optical modulators, nonreciprocal optical device, and nonlinear optical imaging.
27 Exploration of nonlinear optical response (e.g., frequency conversion) in the infrared, especially
28 the mid-infrared (MIR) region, is highly desirable for ultrafast MIR laser applications ranging
29 from tunable MIR coherent sources, MIR supercontinuum generation, MIR frequency-comb-
30 based spectroscopy to high harmonic generation. However, nonlinear optical effects in 2D layered
31 materials under MIR pump are rarely reported, mainly due to the lack of suitable 2D layered

1 materials. Van der Waals layered platinum disulfide (PtS₂) with a sizable bandgap from the visible
2 to the infrared region is a promising candidate for realizing MIR nonlinear optical devices. In this
3 work, we investigate the nonlinear optical properties including third-and fifth-harmonic generation
4 (THG and FHG) in thin layered PtS₂ under infrared pump (1550-2510 nm). Strikingly, the ultra-
5 strong third-order nonlinear susceptibility $\chi^{(3)}(-3\omega; \omega, \omega, \omega)$ of thin layered PtS₂ in the MIR
6 region was estimated to be over 10^{-18} m²/V², which is about one order of that in traditional
7 transition metal chalcogenides. Such excellent performance makes air-stable PtS₂ a potential
8 candidate for developing next-generation MIR nonlinear photonic devices.

9 **Keywords:** Mid-infrared, 2D materials, platinum disulfide, nonlinear optics, optical harmonic
10 generation

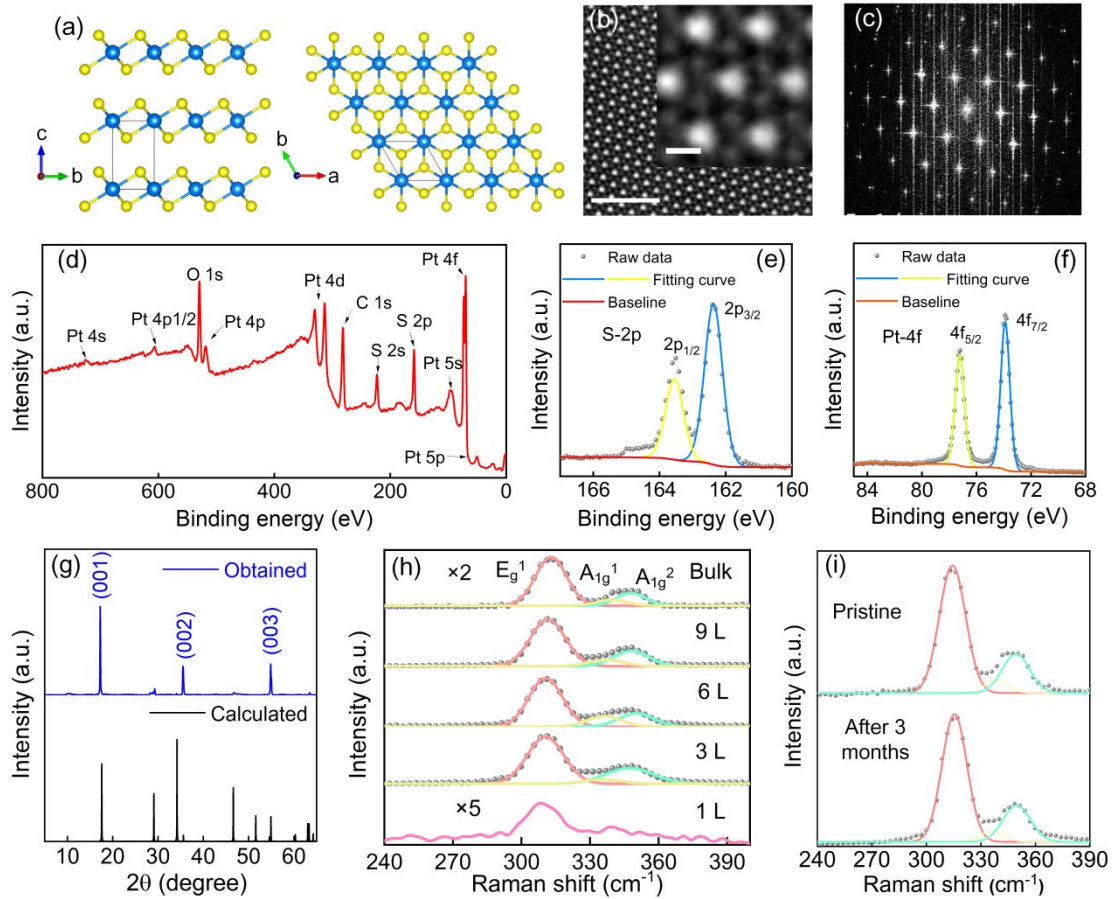
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1 Nonlinear optical activities, such as harmonic generation, are of importance for various
2 applications including frequency conversion,¹ laser technology^{2,3} and optical spectroscopy.^{4,5}
3 Nonlinear optical materials for nonlinear optical response (e.g., frequency conversion) in the
4 infrared especially the mid-infrared (MIR) region are highly desirable due to the emergence of
5 ultrafast MIR laser applications including tunable MIR coherent sources,³ MIR supercontinuum
6 generation,⁶ MIR frequency-comb-based spectroscopy⁷ and high harmonic generation.⁸ Two
7 dimensional (2D) van der Waals layered materials have attracted much attention due to their high
8 nonlinear susceptibilities,⁹⁻¹⁹ good compatibility with nanoscale photonic devices²⁰⁻²⁴ and tunable
9 optical nonlinearity.²⁵⁻³² Harmonic generations in 2D layered materials such as graphene and
10 transition metal chalcogenides (TMDs) have been widely studied in recent years.^{1,33-40} However,
11 most of nonlinear optical responses in 2D layered materials are focused in the visible to near
12 infrared (NIR) region. Few attentions have been paid to the MIR region, mainly due to the lack of
13 suitable 2D layered materials. Black phosphorus (BP) possesses layer-dependent direct bandgaps
14 covering the visible to MIR spectral regions.⁴¹ However, BP is not stable in air, which leads to
15 rapid degradation in ambient environment. Graphene has a giant third-order nonlinear
16 susceptibility ($\chi^{(3)}(-3\omega; \omega, \omega, \omega)$, called $\chi^{(3)}$) in the MIR region ($8 \times 10^{-17} \text{ m}^2/\text{V}^2$ at ~ 3100
17 nm),³³ but graphene suffers from intrinsic limitations for on-off ratio in electronic and photonics
18 devices due to its zero-bandgap nature. Van der Waals layered platinum disulfide (PtS₂), as one
19 kind of group-10 TMDs similar to PtSe₂,⁴² PdSe₂,⁴³ and PtTe₂,⁴⁴ demonstrates great potential for
20 optoelectronic applications due to its high carrier mobility,⁴⁵ widely tunable bandgap (Figure S1 in
21 the Supporting Information),^{46,47} and high stability. Studying the nonlinear optical activity of
22 layered PtS₂ in the infrared region (especially MIR region) is an essential step toward the
23 development of next-generation MIR nonlinear photonic devices. Up to now, there are few works
24 that studied the nonlinear absorption effect in layered PtS₂,⁴⁸⁻⁵⁰ corresponding to the imaginary
25 part of its third-order nonlinear susceptibility ($\text{Im } \chi^{(3)}(\omega; -\omega, \omega, \omega)$). However, up to now, there
26 still lacks experimental investigation of harmonic generations in 2D layered PtS₂.

27 In this work, we report optical harmonic generations including THG and fifth-harmonic
28 generation (FHG) in 2D layered PtS₂ under NIR to MIR pump (1550 nm-2510 nm). We find that
29 the thin layered PtS₂ flake exhibits ultra-strong $\chi^{(3)}$ values of over $10^{-18} \text{ m}^2/\text{V}^2$ under MIR pump,

1 which is about one order of that in traditional TMDs (e.g., MoS₂, WS₂). In addition, FHG is also
 2 observed in layered PtS₂ under MIR pump (2510 nm). These findings not only reveal the intrinsic
 3 nonlinear optical properties of PtS₂, but also possess a potential for developing nonlinear
 4 optoelectronic devices in the NIR to MIR regions.



5
 6 **Figure 1.** Electron and phonon characterization of the PtS₂ crystal. (a) The crystal structure of PtS₂ with side (left)
 7 and top (right) views. (b) The HRTEM image of the thin PtS₂ flake. Scale bar: 2 nm. Inset: an enlarged area of
 8 Figure 1 (b). Scale bar: 0.2 nm. (c) The fast Fourier transform (FFT) diffraction patterns. (d) XPS of the PtS₂ flake.
 9 (e) S-2p core level, (f) Pt-4f core level. (g) The XRD pattern of the PtS₂ flake. Upper: obtained; bottom: calculated.
 10 (h) Raman spectra of 1 L, 3 L, 6 L, 9 L and bulk PtS₂. (i) Raman spectra of the PtS₂ flake. Upper: pristine; bottom:
 11 after 3 months.

12 **Results and Discussion**

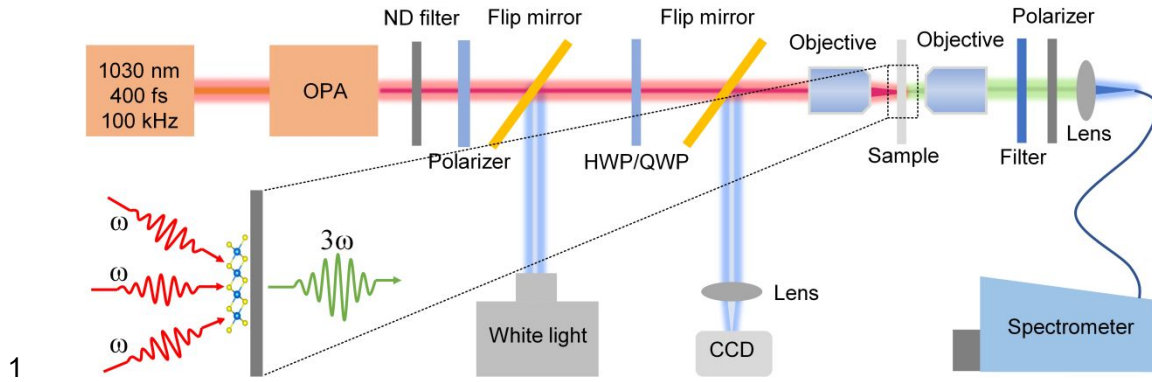
13 **Crystal structure and characterization of van der Waals layered PtS₂**

14 PtS₂ crystal belongs to $P\bar{3}m1$ space group (No. 164). Figure 1a shows the side-view and top-view
 15 of the PtS₂ crystal, which exhibits a tilted and slightly distorted octahedral structure, called 1T
 16 phase with AA arrangement for stacked layers.^{46,51} In this structure, each Pt atom is coordinated

1 by six S atoms in a tilted octahedral structure, which is different from that of Mo in 2H TMD,
2 such as MoS₂ and WS₂, in which each Mo atom is arranged in the site of a triangular-prismatic
3 site. In the side view, PtS₂ exhibits a crystal structure with a monolayer thickness of 8.6 Å.⁴⁶ As
4 shown Figures 1b and 1c, the crystal structure of the layered PtS₂ is verified by high-resolution
5 transmission electron microscopy (HRTEM), showing the 1T structure. To further confirm the
6 elemental composition and valence states in the PtS₂ crystal, X-ray photoelectron spectroscopy
7 (XPS) measurement is conducted (Figure 1d), confirming Pt and S element ratio of ~1:2 and the
8 valence of Pt⁴⁺ and S²⁻. To further investigate the crystal quality of experimentally prepared PtS₂,
9 X-ray diffraction (XRD) characterization is performed. As shown in Figure 1g, the diffraction
10 pattern exhibits three robust peaks at 18°, 36°, and 55°, corresponding to (001), (002), and (003),
11 suggesting a high crystal quality.⁴⁵ The three peaks are well matched with the calculated results
12 shown in the bottom of Figure 1g, suggesting the pure phase. Figure 1h shows the thickness-
13 dependent Raman spectra of 2D layered PtS₂ in 1 L, 3 L, 6 L, 9 L and bulk. For monolayer PtS₂,
14 there is only an apparent E_g¹ phonon mode at 307 cm⁻¹, which is due to the limited number of
15 scattering atoms.⁴⁶ E_g¹ phonon mode undergoes a blue shift, and the A_g¹ together with the A_g²
16 phonon mode emerge with the increase of the layer number. As mentioned, the PtS₂ crystal has
17 good air-stability, which is important for practical applications. To verify this point, we compare
18 the Raman spectra of a pristine layered PtS₂ flake that is measured at times separated by 3 months.
19 As shown in Figure 1i, the Raman spectrum remains unchanged, confirming its good air-stability.

20 **THG process in thin layered PtS₂ flakes**

21 Optical harmonic generations in layered PtS₂ are measured using a home-built microscopic system
22 (Figure 2 and Methods). Figure 3a shows the optical microscopic image of a thin PtS₂ flake. The
23 average thickness of the PtS₂ flake is measured to be ~12 nm using the atomic force microscopy
24 (AFM), corresponding to a flake with roughly 14 L. Since PtS₂ belongs to $\bar{3}m$ point group with
25 inversion symmetry, it is expected to exhibit only odd-order (e.g., THG) nonlinear optical
26 effects.⁵² To measure the THG signal, the PtS₂ flake is illuminated by a 1550-nm laser with a spot
27 size of ~2 μm, generated from an optical parametric amplifier (OPA). Figure 3c illustrates THG
28 spectra measured at different pump powers with the central pump wavelength around 1550 nm.
29 The spectrum of fundamental light measured by a Zolix monochromator integrated with a liquid



1
2 **Figure 2.** Schematic of the experimental setup. OPA: optical parametric amplifier, ND filter: neutral density filter,
3 HWP: half waveplate, QWP: quarter waveplate. Inset: principle of the THG process.

4 cooled indium antimonide (InSb) detector is shown in the inset of Figure 3c. The linewidth of the
5 THG spectrum is about 6 nm, which is about three times smaller than that of the fundamental light
6 (~16 nm) due to the nonlinear multiphoton process.⁵³ Figure 3d shows log-log plot of the THG
7 intensity as a function of the pump power. It can be found that the THG intensity is cubically
8 (slope=3) dependent on the pump power, indicating the THG process. Wavelength-dependent
9 THG in the range from 1515 nm to 1644 nm is also measured and shown in Figures 3e and 3f,
10 showing weak spectral dependence of the THG intensity, indicating a lack of resonance absorption
11 in the PtS₂ flake.⁵⁴ Absorption spectra of PtS₂ flakes with thicknesses of 9 nm, 20 nm and 24 nm
12 are also measured (Figure S2 in the Supporting Information), which reveal that there is no clear
13 excitonic transition peak in the visible to MIR region due to the decreased exciton binding energy
14 in the thicker PtS₂ derived from the stronger interlayer interaction and indirect bandgap nature.^{46,55}

15 The polarization angle dependent THG is also measured, as shown in Figure 4a. It can be
16 seen that the THG intensity is independent on the pump polarization, which is due to its $\bar{3}m$ point
17 group for the in-plane symmetry.⁵² Besides, the polarization of the THG signal is also measured
18 by rotating a polarizer between the collective objective lens and the spectrometer. Figure 4b shows
19 change of the THG intensity when the polarizer is rotated from 0 to 360°, which indicates the
20 linear polarization characteristic of the THG signal. In addition, the THG polarization is parallel to
21 the polarization of the fundamental light. To clarify its fundamental principle for polarization
22 measurement, we introduce $\chi^{(3)}$ tensor of the PtS₂ crystal. Here, the pump light is linearly

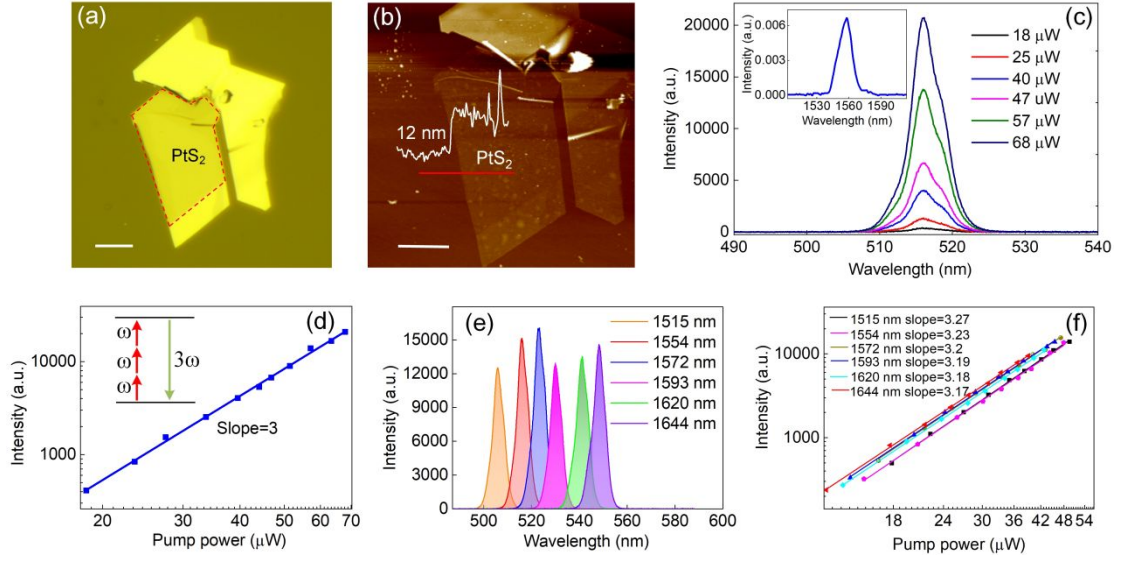


Figure 3. THG processes in the thin layered PtS₂ flake. (a) Optical microscopic image of the PtS₂ flake. Scale bar: 10 μm. (b) The AFM image of the thin PtS₂ flake. Scale bar: 10 μm. (c) THG spectra from the PtS₂ flake depending on the pump power. Inset: 1550-nm pump spectrum. (d) The THG intensity as a function of the pump power. Inset: diagram of the THG process. (e) THG spectra for various pump wavelengths. (f) The THG intensity as a function of the pump power for different pump wavelengths.

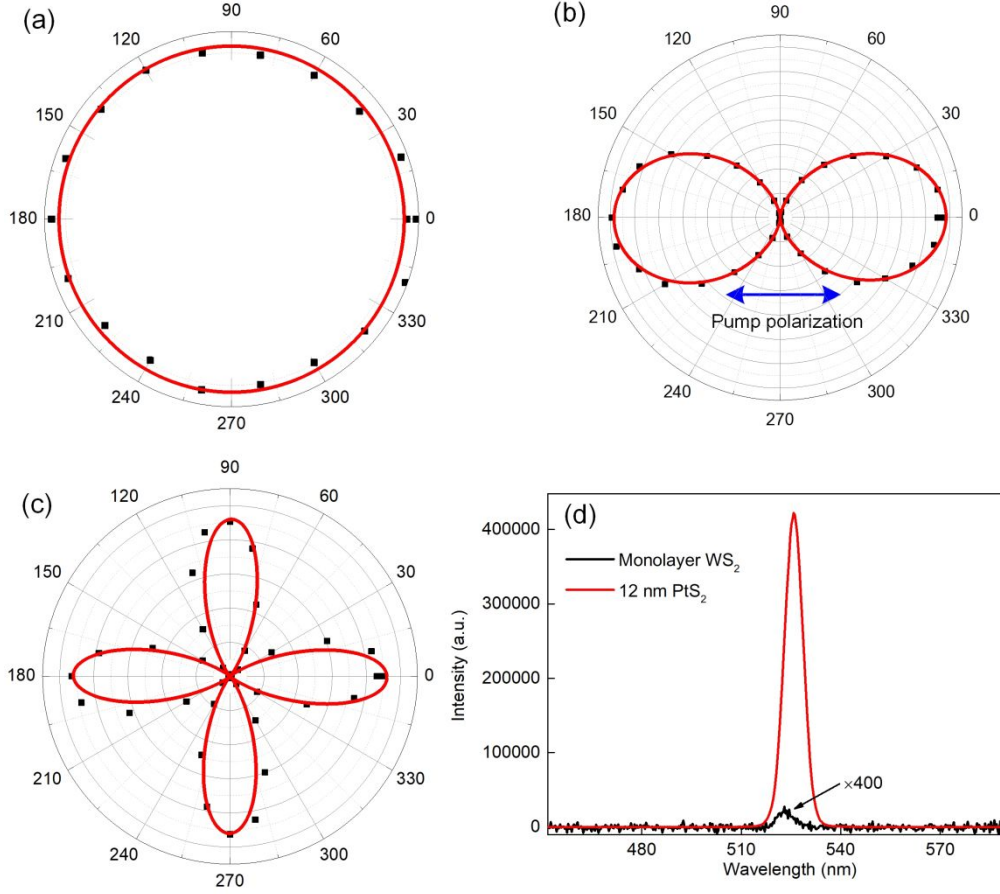
polarized with the fundamental frequency ω , which can be expressed as $\mathbf{E} = E_x \mathbf{x} + E_y \mathbf{y} + E_z \mathbf{z}$. A coordinate is assigned in Figure 1a, where the x -axis can be along the a -axis, the y -axis is perpendicular to the x -axis, and the z -axis is oriented along c -axis. Its $\chi^{(3)}$ tensor can be expressed as:^{52,56}

$$\begin{bmatrix} \chi_{11} & 0 & 0 & 0 & \chi_{15} & \chi_{16} & -\chi_{15} & 1/3\chi_{11} & 0 & 0 \\ 0 & \chi_{11} & 0 & \chi_{16} & -\chi_{15} & 0 & 0 & 0 & 1/3\chi_{11} & \chi_{15} \\ \chi_{31} & 0 & \chi_{33} & 0 & \chi_{35} & 0 & \chi_{35} & -\chi_{31} & 0 & 0 \end{bmatrix} \quad (1)$$

where the first subscript 1, 2, 3 refers to x, y, z , respectively, and the second subscript refers to the following:

$$\begin{array}{cccccccccc} xxx & yyy & zzz & yzz & yzy & xzz & xxz & xyy & xxy & xyz \\ 1 & 2 & 3 & 4 & 5 & 6 & 7 & 8 & 9 & 0 \end{array}$$

The sample is excited along z -axis in our configuration, thus the electric field intensity of the pump light in the z -direction can be negligible. Therefore, the pump light can be further expressed as $\mathbf{E} = E_0(x\cos(\theta) + y\sin(\theta))$, where θ is the polarization angle relative to the x -axis. Therefore, components of the third-order nonlinear polarization ($P^{(3\omega)}$) in the layered PtS₂ crystal can be



1
2 **Figure 4.** Angle dependent THG in the thin PtS₂ flake. (a) The THG intensity depending on the polarization
3 direction of the linear pump light. (b) The THG intensity versus the rotation angle of the polarizer. (c) The THG
4 intensity depending on the pump ellipsometry. The polar-plot angle corresponds to linearly polarized light when θ
5 = $0^\circ+m\cdot 90^\circ$, and to circularly polarized light when $\theta = 45^\circ+m\cdot 90^\circ$. (d) THG spectra from the thin PtS₂ flake and
6 monolayer WS₂ with the pump power of $\sim 49 \mu\text{W}$.

7 expressed as:

8

$$P^{(3\omega)} = \begin{bmatrix} P_x^{3\omega} \\ P_y^{3\omega} \\ P_z^{3\omega} \end{bmatrix} = \varepsilon_0 E_0^3 \begin{bmatrix} \chi_{11} \cos^3(\theta) + \chi_{11} \cos(\theta) \sin^2(\theta) \\ \chi_{11} \sin^3(\theta) + \chi_{11} \cos^2(\theta) \sin(\theta) \\ 0 \end{bmatrix} \quad (2)$$

9 Therefore, the x - and y - components of the THG intensity can be expressed as:

10

$$\begin{aligned} I_x^{3\omega} &\propto (\chi_{11} \cos^3(\theta) + \chi_{11} \cos(\theta) \sin^2(\theta))^2 \\ I_y^{3\omega} &\propto (\chi_{11} \sin^3(\theta) + \chi_{11} \cos^2(\theta) \sin(\theta))^2 \end{aligned} \quad (3)$$

11 The total THG intensity $I=I_x+I_y$. The measured polarization resolved THG (Figure 4a) can be well
12 fitted using Eq. (3). Furthermore, the polarization dependent THG for pump light with elliptical
13 polarization is also studied. The electric field intensity of the pump light with the major axis of the

1 polarization ellipsometry is considered as $E = E_0 \mathbf{e}^\pm$, where $\mathbf{e}^\pm = x \cos(\beta) \pm iy \sin(\beta)$ with β being the
 2 ellipticity angle. Therefore, the electric field intensity of the THG signal can be expressed as (for
 3 left circular pump):

$$4 \quad \mathbf{E}^{3\omega} \propto (\chi_{11} \cos^3(\beta) - \chi_{11} \cos(\beta) \sin^2(\beta)) \mathbf{x} + i(\chi_{11} \cos^2(\beta) \sin(\beta) - \chi_{11} \sin^3(\beta)) \mathbf{y} \quad (4)$$

5 In Eq. (4), $\beta = 0^\circ$ corresponds to linear polarization, while $\beta = 45^\circ$ corresponds to a circular
 6 polarization. The desired elliptical polarization of the pump light is obtained by rotating a quarter
 7 waveplate (QWP) working at 1550 nm before being focused on the sample. Here, the ellipticity of
 8 the pump light is determined by the angle between the fast axis of QWP and the linearly polarized
 9 light. Figure 4c shows polar-plot of the angular dependence of the THG intensity in the layered
 10 PtS₂. It can be found that the THG intensity is maximum for a linearly polarized pump, while it
 11 becomes almost zero in the case of the circularly polarized pump. The fitted red curve using Eq.
 12 (4) agrees well with the experimental data (black dots).

13 Monolayer WS₂ ($\chi^{(3)} = 2.4 \times 10^{-19} \text{ m}^2 / \text{V}^2$) is used as a reference sample to estimate $\chi^{(3)}$
 14 value of the layered PtS₂.⁵⁷ Before doing the estimation, we need to consider whether it will be
 15 influenced by the coherent length (l_{coh}). For THG in the bulk materials, there will be phase
 16 mismatch ($\Delta k = k_3 - 3k_1 = \frac{6\pi(n_1 - n_3)}{\lambda_1}$) between the fundamental and the forward propagating
 17 third-harmonic signal, where k_1 and k_3 are wave vectors of the fundamental and the forward
 18 propagating third-harmonic signals, respectively, n_1 and n_3 are refractive indexes (RIs) at
 19 fundamental (λ_1) and THG (λ_3) wavelengths, respectively.⁵⁸ For layered PtS₂, its RIs at $\lambda_1 = 1550$
 20 nm and $\lambda_3 = 516$ nm are about 3.54 and 3.76, respectively.⁵⁹ By calculating using equation:

21 $l_{coh} = |\pi / \Delta k|$,⁵⁸ we find that l_{coh} should be about 1174 nm, which is much larger than the flake
 22 thickness (12 nm), which means that the phase mismatch cannot influence the THG intensity in
 23 our case. For the THG process in the reflective configuration, there is phase mismatch
 24 ($\Delta k = k_3 - 3k_1 = \frac{6\pi(n_1 + n_3)}{\lambda_1}$) between the fundamental and the backward propagating third-
 25 harmonic signal.⁵⁸ Thus, l_{coh} is calculated to be about 35 nm in the reflective configuration, much

1 smaller than that in the transmission configuration. We only do the transmission measurement
 2 since our current setup does not support the reflective configuration. The estimation is performed
 3 using the following equation:^{60,61}

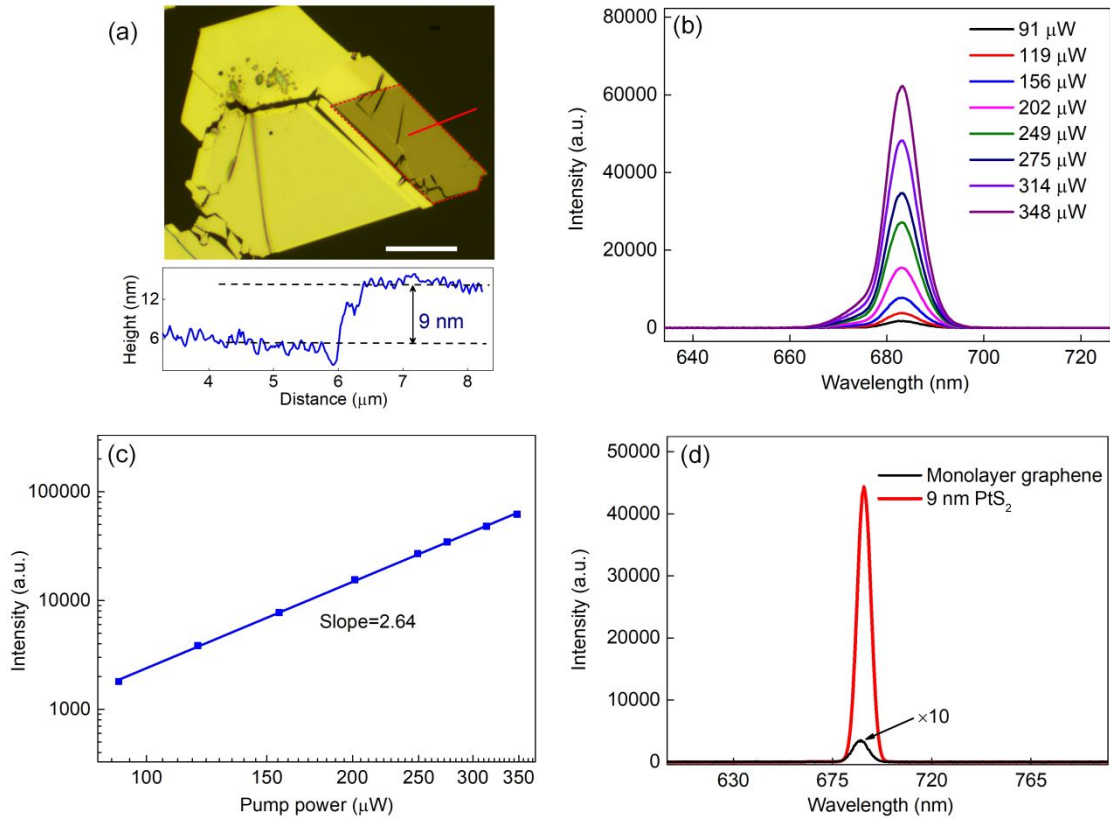
$$4 \quad \chi_{PtS_2}^{(3)} = \frac{l_{WS_2}}{l_{PtS_2}} \sqrt{\frac{THG_{PtS_2}}{THG_{WS_2}}} \chi_{WS_2}^{(3)} \quad (5)$$

5 where THG_{PtS_2/WS_2} are THG intensities of the thin layered PtS₂ flake and monolayer WS₂,
 6 respectively, the thickness of monolayer WS₂ (l_{WS_2}) is about 0.67 nm, the thickness of the PtS₂
 7 flake (l_{PtS_2}) is about 12 nm. As shown in Figure 4d, by comparing to monolayer WS₂ (Figure S4),
 8 $\chi^{(3)}$ value of PtS₂ is estimated to be about $1.3 \times 10^{-18} \text{ m}^2/\text{V}^2$ under 1550-nm pump, which is larger
 9 than that in most of 2D materials.

10 **Optical harmonic generations under MIR pump**

11 In addition to the nonlinear response of layered PtS₂ in the NIR region, it would be desirable to
 12 further investigate its nonlinear activity in the MIR region (over 2000 nm). Figure 5 shows the
 13 THG of the thin layered PtS₂ flake under 2050-nm pump. The thickness of the PtS₂ flake used in
 14 this measurement is ~9 nm, as shown in Figure 5a. Figure 5b shows evolution of the THG
 15 spectrum with the increase of the pump power. The THG wavelength of around 684 nm
 16 corresponds to the pump wavelength of around 2050 nm. As shown in Figure 5c, the linearly fitted
 17 slope of ~2.64 verifies the three-photon nonlinear process. Furthermore, its $\chi^{(3)}$ value is also
 18 estimated using Eq. (5) by comparing to monolayer graphene (Figure S5).³³ As shown in Figure
 19 5d, $\chi^{(3)}$ value of layered PtS₂ around 2050 nm is estimated to be about $2.7 \times 10^{-18} \text{ m}^2/\text{V}^2$ through
 20 the spectral measurements for the thin layered PtS₂ flake and monolayer graphene performed with
 21 the same pump power of ~100 μW .

22 In the next step, optical harmonic generation is studied at a pump wavelength of around 2510
 23 nm. As shown in Figure 6a, we can see that THG around 840 nm and FHG around 502 nm are
 24 observed with a pump power of ~588 μW . It can be found that the FHG intensity is much smaller
 25 than the THG intensity, which is due to a fact that FHG needs the higher pump power.^{8,62} Figures



1

2 **Figure 5.** THG of the thin PtS₂ flake under 2050-nm pump. (a) Upper: optical microscopic image of the PtS₂ flake;
 3 bottom: AFM measurement result. Scale bar: 30 μm. (b) THG spectra depending on the pump power. (c) The THG
 4 intensity as a function of the pump power. (d) THG spectra from the PtS₂ flake and the monolayer graphene
 5 measured at the pump power of ~100 μW.

6 6b and 6c represent evolution of THG and FHG spectra with the increase of the pump power.

7 Furthermore, $\chi^{(3)}$ value around 2510 nm is also estimated by comparing to the monolayer

8 graphene,³³ as shown in Figure 6d. We find that layered PtS₂ has a $\chi^{(3)}$ value of about 2.1×10^{-18}

9 m^2/V^2 in the MIR region. The THG and FHG intensities as a function of the pump power are also

10 measured, as shown in Figure 6e. It can be found that the THG intensity exhibits a nearly cubic

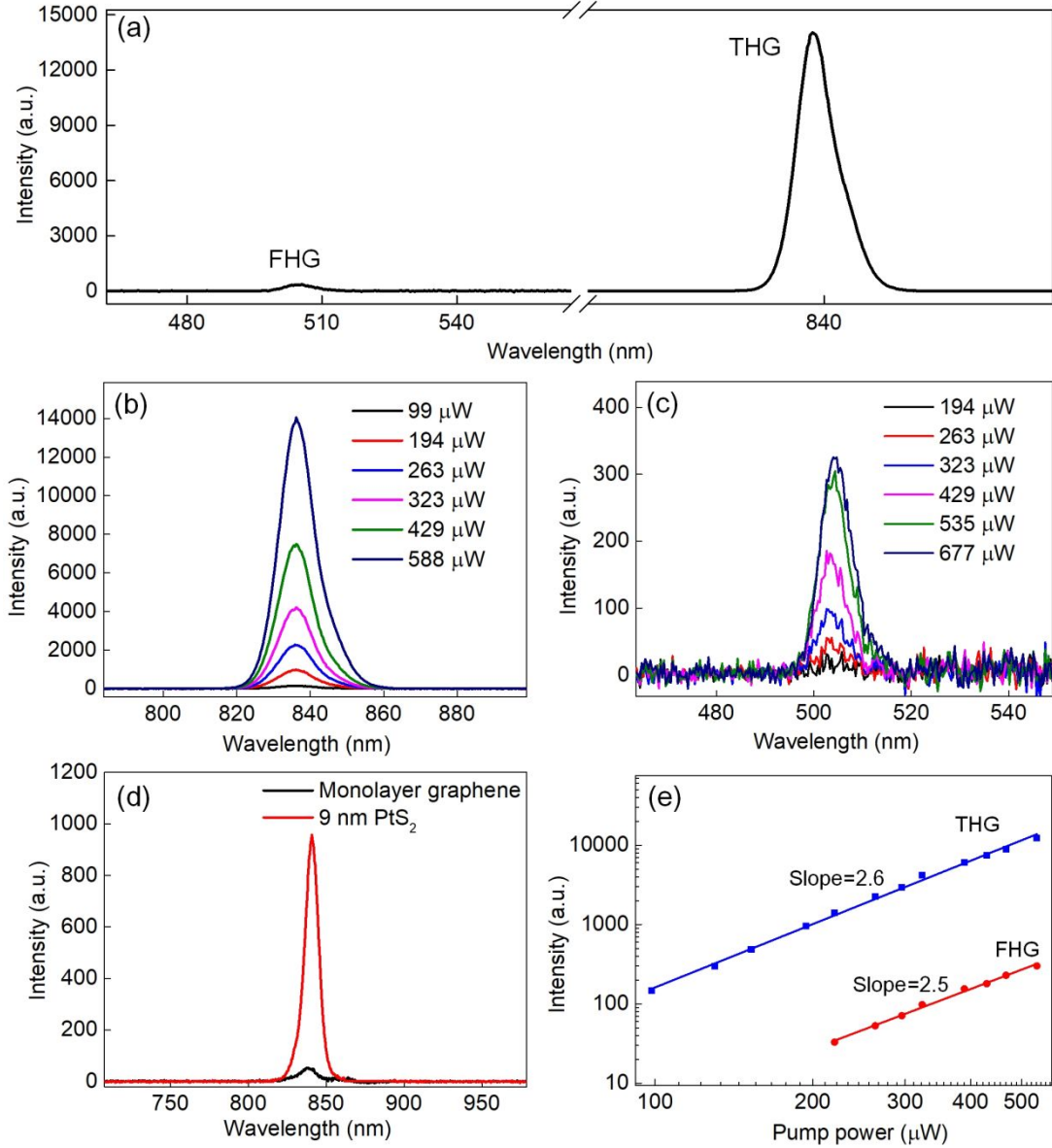
11 dependence on the pump power, but the FHG intensity follows the power dependence of $I^{2.5}$,

12 which shows the non-perturbative character of the HHG process.^{8,63} Besides, the THG intensity as

13 a function of the pump power for different thicknesses is plotted in Figure 7a. It is noted that the

14 THG intensity increases as the thickness is increased from 9 nm to 270 nm. However, the intensity

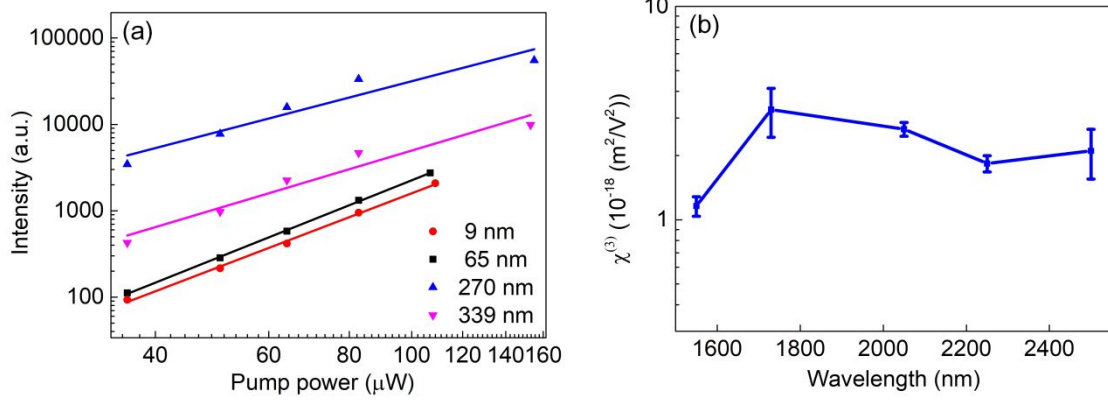
15 decreases when the thickness further increases to 339 nm, which is attributed to the optical



1

2 **Figure 6.** THG and FHG processes of the PtS₂ flake under 2510-nm pump. (a) Typical harmonic generation
3 spectra including THG and FHG. (b) THG spectra depending on the pump power. (c) FHG spectra depending on
4 the pump power. (d) THG spectra of the PtS₂ flake and the monolayer graphene under 2510-nm pump. (e) THG
5 and FHG intensities depending on the pump power.

6 absorption induced decay when the thickness is large enough.⁶⁴ Finally, we also measure $\chi^{(3)}$
7 value of layered PtS₂ around 1730 nm and 2250 nm, respectively. As shown in Figure 7b, by
8 comparing to the monolayer graphene, layered PtS₂ has $\chi^{(3)}$ values of about $3.3 \times 10^{-18} \text{ m}^2/\text{V}^2$
9 around 1730 nm and about $1.8 \times 10^{-18} \text{ m}^2/\text{V}^2$ around 2250 nm, which means that as an air-stable
10 layered material, layered PtS₂ possesses broadband ultra-strong $\chi^{(3)}$ from NIR to MIR regions.



1

2 **Figure 7.** Thickness and wavelength dependent THG processes. (a) Pump power dependent THG intensity with

3 different thicknesses (2510-nm pump). (b) Measured $\chi^{(3)}$ values of the thin layered PtS₂ flakes from NIR to MIR

4 regions.

5 Comparison of the nonlinear susceptibility of layered PtS₂ with well-studied 2D layered

6 materials and bulk materials is presented in Table 1. It can be found that the nonlinear

7 susceptibility of layered PtS₂ is larger than that in most 2D layered materials and common bulk

8 materials (e.g., Si₃N₄, fused silica). In particular, $\chi^{(3)}$ values of layered PtS₂ are about one order of

9 traditional TMDs, demonstrating that layered PtS₂ is a good candidate for MIR nonlinear

10 photonics. Furthermore, its nonlinear susceptibility is also much larger than that in common bulk

11 materials used in photonic devices, such as Si₃N₄ and fused silica, with good stability in the air.

12 Finally, the nonlinear RI (n_2) of layered PtS₂ is also studied, which is derived from the Kerr

13 nonlinearity, relevant to the real part of the third-order nonlinear susceptibility

14 ($\text{Re } \chi^{(3)}(\omega; -\omega, \omega, \omega)$). **The nonlinear Kerr effect induced n_2 should be measured by the closed-**

15 **aperture (CA) Z-scan method.**⁶⁵ The value of n_2 in layered PtS₂ is up to 1.2×10^{-14} m²/W at 1030

16 nm, which is larger than that in most of van der Waals 2D layered materials (Table S1 in the

17 Supporting Information). **Based on the measured n_2 at 1030 nm, the $\text{Re } \chi^{(3)}(\omega; -\omega, \omega, \omega)$ value**

18 **of PtS₂ is calculated to be 5.3×10^{-16} m²/V² according to the equation:**⁶⁶

19
$$n_2(\text{m}^2/\text{W}) = \frac{3}{4\epsilon_0 c n_0^2} \text{Re } \chi^{(3)}(\omega; -\omega, \omega, \omega),$$
 where ϵ_0 is the vacuum permittivity, n_0 is the

20 **linear RI and c is the speed of light in a vacuum. The $\text{Re } \chi^{(3)}(\omega; -\omega, \omega, \omega)$ value is larger than**

1 $\chi^{(3)}(-3\omega; \omega, \omega, \omega)$ of PtS₂, which is because the band edge absorption of PtS₂ at 1030 nm
2 is much larger than that at the wavelength larger than 1550 nm (Figure S2a in the Supporting
3 Information). In short, both $\text{Re } \chi^{(3)}(\omega; -\omega, \omega, \omega)$ and $\chi^{(3)}(-3\omega; \omega, \omega, \omega)$ verify the ultra-
4 strong third-order nonlinear susceptibility of layered PtS₂.

5

1 **Table 1.** Typical $\chi^{(3)}$ values studied by the THG process for low dimensional materials and other
 2 common bulk materials used in photonics

Material	$\chi^{(3)} (10^{-19} m^2 / V^2)$	Pump parameters	Ref.
FRP	26.72	1300 nm, 200 fs, 2 kHz	67
MoS ₂	3.6	1560 nm, 150 fs, 50 MHz	57
WS ₂	2.4	1560 nm, 150 fs, 50 MHz	57
Graphene	800	3100 nm, 150 fs, 80 MHz	33
Graphene	50	1771 nm, 150 fs, 80 MHz	33
Graphene	1.5	1560 nm, 150 fs, 89 MHz	68
BP	1.64	1560 nm, 100 fs, 8 MHz	69
hBN	0.084	1064 nm, 150 fs, 80 MHz	53
Cd ₃ As ₂	2.3	1560 nm	70
Si ₃ N ₄	0.28	1064 nm, 70 ps, 1 kHz	71
Fused silica	0.002	1064 nm	72
PtS ₂	13	1550 nm, 150 fs, 100 kHz	This work
	33	1730 nm, 150 fs, 100 kHz	This work
	21	2510 nm, 150 fs, 100 kHz	This work

3 FRP: fibrous red phosphorus; BP: black phosphorus; hBN: hexagonal boron nitride

4 **Conclusion**

5 In conclusion, optical harmonic generations including (THG and FHG) of layered PtS₂ from the
 6 NIR to MIR regions (1550 nm-2510 nm) are experimentally studied. The layered PtS₂
 7 demonstrates ultra-strong $\chi^{(3)}$ values of over $10^{-18} m^2/V^2$ under MIR pump, which is about one
 8 order of that in traditional TMDs. Finally, demonstration of high performance fifth-harmonic
 9 nonlinear response is also realized in layered PtS₂. We believe van der Waals layered PtS₂ with its
 10 good air-stability and large nonlinear responses holds great promise for nonlinear optoelectronic
 11 applications including nonlinear wavelength conversion, nonlinear photonic waveguide.

12

13

1 **Methods**

2 **Sample Preparation**

3 PtS₂ single crystals were synthesized via the chemical vapor transport method. The platinum
4 powder, red phosphorus powder, and sulfur powder with a molar ratio of 1:1:3 were sealed in a
5 silica tube under a high vacuum ($<10^{-2}$ Pa). The sealed tube was loaded in a horizontal furnace,
6 whose reaction zone and growth zone were heated to 800 °C and 750 °C within one day,
7 respectively. After two weeks, the shiny crystals were obtained in the cold end of the silica tube.
8 The thin PtS₂ flakes were mechanically exfoliated from the bulk crystal. The monolayer WS₂
9 flakes were exfoliated from bulk WS₂ crystals (2D Semiconductors, USA). The monolayer
10 graphene was obtained by chemical vapor deposition. The characterization of monolayer WS₂ and
11 graphene could be found from part S3 and part S4 in the Supporting Information, respectively.

12 **Sample Characterization**

13 The thicknesses of layered PtS₂ flakes were confirmed through the AFM method. The crystal
14 structure of the exfoliated PtS₂ flakes was studied by HRTEM (JEOL-2100F). The element
15 compositions of the PtS₂ crystal were measured through XPS (Kratos AXIS Supra spectrometer
16 with a dual anode Al K α (1486.6 eV) X-ray monochromatic source). The XRD patterns were
17 measured using a XRD Bruker D8 Advance Powder with Cu-K α target at the angle of 5°-60°. The
18 thickness-dependent Raman spectra were collected by a confocal Raman spectrometer (WITec
19 alpha 300 RAS) equipped with a 532-nm laser. The absorption spectra of the layered PtS₂ flakes
20 were measured using a Fourier-transform infrared spectrometer (FTIR) equipped with a silicon
21 detector, an InGaAs detector, and a mercury cadmium telluride (MCT) detector. The micro-
22 absorption measurement was carried out by putting the layered PtS₂ flakes between two aligned
23 reflective objectives for light focusing and collection, respectively. The light was focused on the
24 samples by the objective for the absorption measurement. The absorption spectra were obtained by
25 comparing the transmission spectra between the samples and a substrate.¹⁹

26 **Experimental Setup**

27 Optical harmonic generations in layered PtS₂ were measured using a home-built optical system, as
28 shown in Figure 2. The sample was excited by an OPA system (pulse width 150 fs, repetition rate
29 100 kHz). The femtosecond laser beam first passed through a ND filter, a polarizer, and HWP, and
30 then was focused by an objective lens (50 \times , NA 0.45) to a spot with a diameter of ~ 2 μ m. QWP

1 was used to generate elliptical light. The MIR pump light was focused by a reflective objective
2 (40×, NA 0.5). The harmonic signals were collected by the other objective lens (20×, NA 0.45),
3 filtered by a short-pass filter, and recorded by a spectrometer integrated with a cooling detector. A
4 polarizer was put after the collective objective lens to check the polarization of the THG signal.
5 Monolayer WS₂ and monolayer graphene were used as references to estimate
6 $\chi^{(3)}(-3\omega; \omega, \omega, \omega)$ value of PtS₂ with the same pump intensity of around 100 GW/cm². For
7 example, the pump intensity at 1550 nm for estimation of $\chi^{(3)}(-3\omega; \omega, \omega, \omega)$ is 104 GW/cm² and
8 the pump intensity at 2050 nm is 96 GW/cm². It is worthy to highlight that $\chi^{(3)}(-3\omega; \omega, \omega, \omega)$
9 is the intrinsic nonlinear optical property of the materials, which is independent on the pump
10 intensity.

11 **First-Principles Calculations**

12 The strongly constrained and appropriately normed (SCAN) meta-generalized gradient
13 approximation (METAGGA) exchange-correlation potential⁷³ implemented in VASP package⁷⁴
14 was employed for electronic structure calculations and structural relaxations with a criterion of
15 0.01 eV/Å. SCAN+rVV10⁷⁵ scheme was chosen to accurately evaluate the weak van der Waals
16 interaction among layers. Gamma-centered k-point meshes of 15×15×9 and 15×15×1 were set for
17 bulk and layers, respectively. A cutoff energy of 500 eV was set for all density functional theory
18 (DFT) calculations. Using maximally localized Wannier functions, the Hamiltonian of tight-
19 binding Hamiltonian was constructed using Wannier90 package⁷⁶ to reproduce the band structure
20 of DFT.

21 By solving G0W0 and Bethe-Salpeter equations (BSE) implemented in Yambo code,⁷⁷ the
22 layer dependent exciton Bohr radius were evaluated in consideration of the electron-hole
23 interactions for the many body Green's function. In G0W0 calculations, the random integration
24 method was employed and truncated coulomb potential at 32 Å in nonperiodic direction was
25 adopted to avoid problematic image effects in 2D slabs. The plasmon pole approximation was
26 used to obtain of the G0W0 band gap, which was employed in BSE as the scissor operator to
27 avoid the underestimated band gap in usual DFT calculations. The local field effect was also
28 considered with a screening cutoff of 3 Ry.

29 **Supporting Information**

1 Supporting Information is available from online.
2 Band-structures of PtS₂, linear optical absorption spectra of layered PtS₂ with different
3 thicknesses, optical characterization of monolayer WS₂, optical characterization of monolayer
4 graphene, nonlinear refractive index (n_2) of PtS₂.

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10 **Conflict of Interest**

11 The authors declare no conflict of interest.

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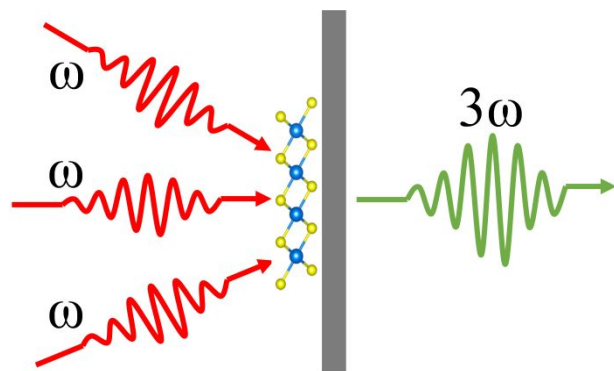
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1 TOC for the manuscript

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