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Synthesis of degradable and chemically recyclable polymers using 4,4-disubstituted five-membered cyclic ketene hemiacetal ester (CKHE) monomers†

Xin Yi Oh,  Yicen Ge and Atsushi Goto *

Novel degradable and chemically recyclable polymers were synthesized using five-membered cyclic ketene hemiacetal ester (CKHE) monomers. The studied monomers were 4,4-dimethyl-2-methylene-1,3-dioxolan-5-one (DMDL) and 5-methyl-2-methylene-5-phenyl-1,3-dioxolan-4-one (PhDL). The two monomers were synthesized in high yields (80–90%), which is an attractive feature. DMDL afforded its homopolymer with a relatively high molecular weight ($M_n > 100\,000$, where M_n is the number-average molecular weight). DMDL and PhDL were copolymerized with various families of vinyl monomers, *i.e.*, methacrylates, acrylates, styrene, acrylonitrile, vinyl pyrrolidinone, and acrylamide, and various functional methacrylates and acrylate. Such a wide scope of the accessible polymers is highly useful for material design. The obtained homopolymers and random copolymers of DMDL degraded in basic conditions (in the presence of a hydroxide or an amine) at relatively mild temperatures (room temperature to 65 °C). The degradation of the DMDL homopolymer generated 2-hydroxyisobutyric acid (HIBA). The generated HIBA was recovered and used as an ingredient to re-synthesize DMDL monomer, and this monomer was further used to re-synthesize the DMDL polymer, demonstrating the chemical recycling of the DMDL polymer. Such degradability and chemical recyclability of the DMDL polymer may contribute to the circular materials economy.

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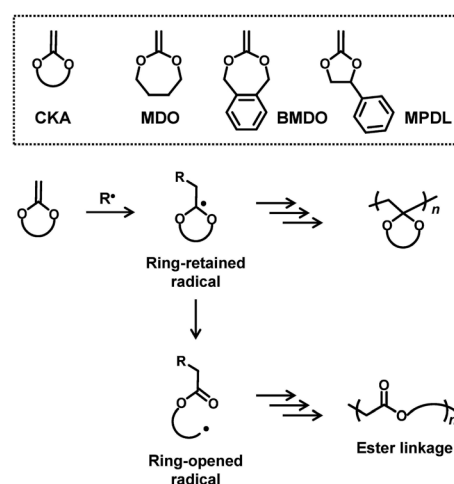
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Introduction

Degradable polymers find numerous applications in, *e.g.*, drug delivery, packaging, and agriculture fields.^{1–9} In the field of radical polymerization, degradable polymers are prepared *via* radical ring-opening polymerization of cyclic ketene acetals (CKAs),^{10–14} for example, gaining significant attention. Examples of CKA are 2-methylene-1,3-dioxepane (MDO),^{15–23} 5,6-benzo-2-methylene-1,3-dioxepane (BMDO),^{24–26} and 2-methylene-4-phenyl-1,3-dioxolane (MPDL)^{27–30} (Scheme 1). CKAs are homopolymerized and co-polymerized with other vinyl monomers, affording degradable ester linkages in the polymer backbones. Mechanistically (Scheme 1), the radical addition to the carbon-carbon double bond of CKA generates a ring-retained radical. The subsequent intramolecular fragmentation (ring opening) generates a ring-opened radical and provides a degradable ester linkage in the backbone. MDO and BMDO are seven-membered ring, and the ring strain would promote the ring opening. BMDO and MPDL produce benzylic ring-opened radicals, and the stabilization of the radical by the phenyl group would

promote the ring opening. The obtained homopolymers and copolymers have been exploited for applications in, *e.g.*, packaging, agrochemical, personal care, and biomedical applications.^{10–13}

In the present work, we are motivated to use cyclic ketene hemiacetal esters (CKHEs) (Scheme 2) as alternatives to CKAs. A

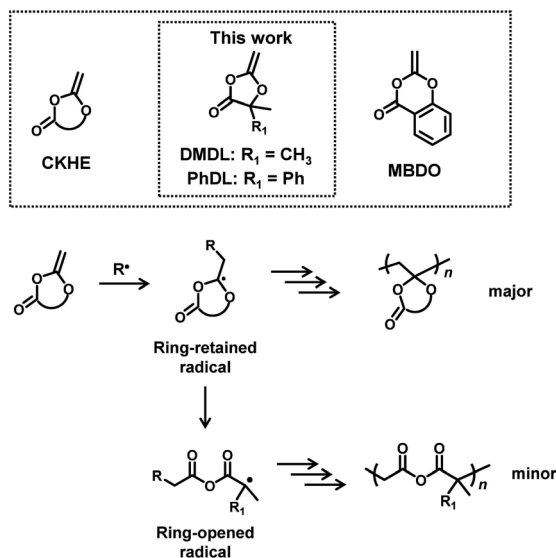


Scheme 1 CKAs monomers and their possible polymerization mechanisms.

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Scheme 2 CKHE monomers and their possible polymerization mechanisms.

CKHE bears a carbonyl group, while a CKA does not. We choose 4,4-disubstituted five-membered CKHEs, *i.e.*, 4,4-dimethyl-2-methylene-1,3-dioxolan-5-one (DMDL) and 5-methyl-2-methylene-5-phenyl-1,3-dioxolan-4-one (PhDL) (Scheme 2). If their ring opening occurs (Scheme 2), a relatively stable polymethacrylate-like radical (a tertiary radical with an ester group and two alkyl groups) is generated. Hence, we hypothesize that the polymerizations of these monomers would prefer ring opening and provide polymers with anhydride degradable linkages in the backbone, which is our initial motivation, although this hypothesis is found to be incorrect as described below.

To the best of our knowledge, there is one report of radical polymerization of a CKHE. The studied CKHE was a six-membered CKHE, *i.e.*, 2-methylene-4*H*-benzo[d][1,3]dioxin-4-one (MBDO (Scheme 2)).³¹ MBDO yielded polymers with ring-retained structures, possibly because its ring-opened radical is a phenyl radical, which is unlikely to generate. The homopolymer of MBDO was insoluble in most of common organic solvents. MBDO was copolymerized with vinyl acetate (VAc), yielding soluble random copolymers. Interestingly, the obtained polymers degraded *via* acid-assisted hydrolysis. After 5 days of the hydrolysis, only small molecules were detected. The authors suggested the backbone (main chain) scission of the polymers. The nuclear magnetic resonance (NMR) analysis showed that the polymers degraded to acetic acid and salicylic acid, which are raw ingredients of MBDO. Therefore, the authors suggested that the polymers are potentially recyclable, although the recycling was not demonstrated experimentally. These results are interesting findings.

In the present paper, we report the use of DMDL and PhDL as CKHEs to synthesize their homopolymers and random copolymers with several co-monomers. DMDL and PhDL are 4,4-disubstituted five-membered CKHEs bearing two methyl groups (DMDL) or a phenyl group and a methyl group (PhDL) at the 4-

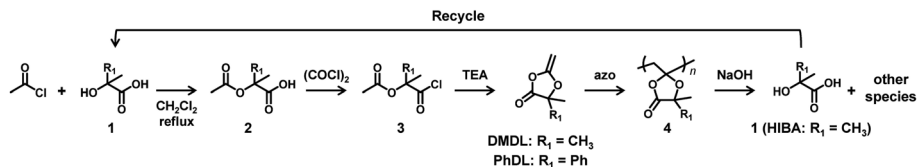
position. Unlike the previously reported MBDO, DMDL was able to generate homopolymers soluble in common organic solvents. DMDL and PhDL were also able to be copolymerized with a range of co-monomers. Mechanistically, contrary to our initial hypothesis, DMDL and PhDL generated polymers with ring-retained structures (Scheme 2) like MBDO, meaning that the activation energy for generating the ring-opened radical is still high for DMDL and PhDL. Nevertheless, despite the ring-retained structure, the PDMDL homopolymers and copolymers degraded *via* the backbone scission in basic conditions, where PDMDL is poly(4,4-dimethyl-2-methylene-1,3-dioxolan-5-one). Importantly, NaOH-assisted degradation of PDMDL homopolymers generated 2-hydroxyisobutyric acid (HIBA), which is a raw ingredient of DMDL (Scheme 3). We recovered HIBA to synthesize the DMDL monomer and PDMDL, demonstrating the recycling of PDMDL (Scheme 3). In this recycling, four of the six carbons of DMDL are recycled *via* HIBA (four carbons). Thus, the polymers serve as (partly) recyclable polymers. Markedly, the degradation of our polymers does not generate mere CO_2 but generates their raw ingredient. This type of polymers are classified to chemically recyclable polymers that can degrade into their raw ingredients or useful oligomers to build other new polymeric materials.² Chemically recyclable polymers are attractive in sustainability and may provide more economic values compared with polymers that can degrade into mere CO_2 .

Results and discussion

Synthesis of DMDL and PhDL

DMDL was previously synthesized in the field of synthetic organic chemistry³² but has not been utilized in polymerization. According to the literature,³² we synthesized DMDL *via* an intramolecular cyclization of commercially available compound **3** ($R_1 = \text{CH}_3$ in Scheme 3) (1 equiv.) using triethylamine (TEA) (1.2 equiv.) in dichloromethane under reflux for 5 h. The conversion determined with ^1H NMR was nearly 100%. There was no report for the synthesis of PhDL. We synthesized PhDL from commercially available compound **1** ($R_1 = \text{Ph}$ in Scheme 3) by referring to the synthesis of a similar compound.^{33,34} The compound **1** (1 equiv.) was reacted with acetyl chloride (2.8 equiv.) in dichloromethane under reflux for 2 h, giving compound **2** ($R_1 = \text{Ph}$ in Scheme 3) in a 96% yield after purification (recrystallization). The purified compound **2** (1 equiv.) was subsequently reacted with oxalyl chloride (1.1 equiv.) in a mixture of dichloromethane (99%) and dimethyl formamide (1%) at room temperature overnight, giving compound **3** ($R_1 = \text{Ph}$ in Scheme 3) in a 99% conversion (as determined with ^1H NMR). The crude compound **3** (1 equiv.) was used for its intramolecular cyclization in the presence of TEA (4.7 equiv.) in dichloromethane under reflux for 2 h, yielding PhDL (Fig. S1 and S2 in ESI†). Both DMDL and PhDL were purified by distillation, and we obtained pure DMDL and PhDL in 80% and 90% yields (based on the amount of compound **3**), respectively. The high yields of these two monomers are attractive features in practical use.





Scheme 3 Synthesis of monomer, polymerization, hydroxide-assisted degradation of polymer, and recycle of HIBA.

In a synthetic point of view, the 4-position of compound 3 must be di-substituted. If the 4-position is not di-substituted (but mono-substituted or non-substituted), the compound 3 is unstable, spontaneously generating an undesired compound (a ketene).³⁵ This requirement is also another reason why we studied 4,4-disubstituted DMDL and PhDL. The stability of the DMDL monomer (8.9 wt%) was studied in DMSO-*d*₆ (89.3 wt) under moisture (1.8 wt% water) at room temperature. DMDL decomposed to form compound 2 (Scheme 3) *via* hydrolysis (Fig. S3 in ESI†). The extent of the DMDL hydrolysis was 5%, 11%, and 64% for 0.5 h, 12 h, and 4 days, respectively (ESI).

Homopolymerizations of DMDL and PhDL

We conducted a homopolymerization of DMDL (100 equiv., 50 wt%) using 2,2'-azoisobutyronitrile (AIBN) (1 equiv.) as a radical initiator in toluene (50 wt%) at 70 °C for 12 h (Table 1, entry 1). The monomer conversion reached 82% (as determined with ¹H NMR), yielding a PDMDL with $M_n = 10\,300$ and $\mathcal{D} (= M_w/M_n) = 1.61$, where M_n and M_w are the number- and weight-average molecular weights, respectively, and \mathcal{D} is the dispersity. The M_n and \mathcal{D} values are not absolute values but PMMA-calibrated GPC values, where PMMA is poly(methyl methacrylate). The obtained PDMDL was soluble in organic solvents such as chloroform, tetrahydrofuran, and dimethylsulfoxide. Fig. 1a and b show the ¹H NMR spectra (CDCl₃) of DMDL (monomer) and PDMDL (polymer) (after purification *via* reprecipitation in hexane/diethyl ether (1/1 (v/v)) (non-solvent)), respectively. The vinyl protons (*a*) of DMDL appearing at 3.62 and 3.70 ppm were converted to the backbone methylene protons (*a'*) of PDMDL appearing at 2.17–3.25 ppm. For PDMDL (Fig. 1b), the integration ratio of methylene protons (*a'*) and dimethyl protons (*b'*) was nearly 1 (*a'*) to 3 (*b'*), agreeing with the PDMDL structure.

As mentioned, there are two possible polymerization pathways, *i.e.*, the ring-retaining and ring-opening pathways (Scheme 2). Fig. 1c and d show the ¹³C NMR spectra (CDCl₃) of DMDL and PDMDL, respectively. Through the polymerization, the vinyl carbon (*b*) of DMDL (Fig. 1c) appearing at 157 ppm can be converted to the backbone quaternary carbon (*b'*) *via* the ring-retaining pathway (Fig. 1d) or to a backbone anhydride carbonyl carbon *via* the ring-opening pathway (Scheme 2). The ¹³C NMR spectrum of PDMDL (Fig. 1d) shows a clear signal for the characteristic backbone quaternary carbon (*b'*) at 106 ppm, meaning the occurrence of the ring-retaining pathway. The signal of the ester carbonyl carbon (*a*) of DMDL appearing at 173 ppm (Fig. 1c) was only slightly sifted to 174 ppm (*a'*) in Fig. 1d) in PDMDL, supporting the ring retention. If the ring-opened structure was also formed, an additional signal for the anhydride carbonyl carbon should have appeared, as anhydride

carbonyl carbons generally appear at 160–170 ppm.³⁶ However, we observed only one signal (for the ester carbonyl carbon for the ring-retained structure) but did not observe other signals in the region at 160–180 ppm, where any carbonyl carbons are to appear. The result means that the obtained PDMDL predominantly contained the ring-retained monomer units and that the ring-opened monomer units were minor units if any present. A possible reason why the ring-opened radical was hardly generated (Scheme 2) is the presence of the trigonal carbonyl carbon in CKHE (DMDL). The trigonal carbonyl carbon can make the CKHE ring more planar than the CKA ring (without the trigonal carbonyl carbon), and the planar conformation may enhance the stabilization of the ring-retained radical.³⁷

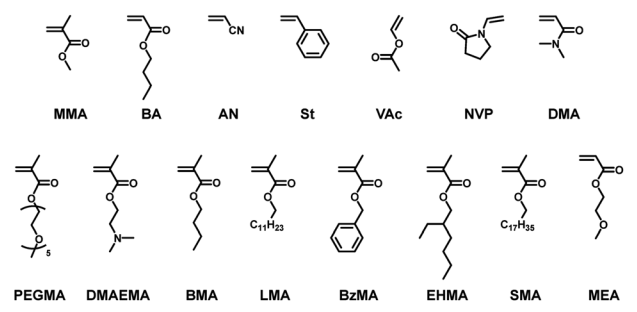
Instead of the solution polymerization in toluene (Table 1, entry 1) (described above), we carried out a bulk polymerization of DMDL (100 equiv.) using AIBN (0.5 equiv.) at 70 °C for 1.5 h (Table 1, entry 2). The absolute monomer concentration increased, and the viscosity of the polymerization solution increased, suppressing radical-radical termination by the so-called gel effect and yielding a higher molecular weight PDMDL with $M_n = 104\,000$ and $\mathcal{D} = 2.04$ at a 79% monomer conversion.

Despite the higher molecular weight, the polymer was still soluble in organic solvents. We measured the glass transition temperature (T_g), crystallization temperature (T_c), melting temperature (T_m), and 50%-decomposition temperature (T_d) of this polymer (PDMDL) using differential scanning calorimetry (DSC) and thermogravimetric analysis (TGA). The T_g , T_c , T_m , and T_d values were 41 °C, 117 °C, 186 °C, and 269 °C, respectively (Fig. S4 in ESI†).

The homopolymerization of PhDL (using AIBN) did not proceed at 70 °C. At an elevated temperature of 120 °C, the homopolymerization of PhDL (using *t*-butyl peroxybenzoate (TBPB)) slowly proceeded but yielded only an oligomer with $M_n < 1000$ (Table 2, entry 1) (Fig. S5 in ESI†). In PhDL, one of the two methyl groups of DMDL is replaced by a phenyl group (Scheme 2). The steric hindrance of the bulky phenyl group might hinder the successive connection of the ring-retained monomer units. Another possibility is that the phenyl group of PhDL stabilizes the ring-opened radical and facilitates the formation of the ring-opened radical but that the radical might be too stable to polymerize. Also, the ring-opened radical might generate polymers with anhydride linkages but the anhydride linkages might be cleaved during the polymerization at a high temperature of 120 °C. These are not definitive but possible explanations. Due to the low molecular weight, the obtained oligomer was difficult to unequivocally analyze, and the exact explanation is not clear at this moment.



Table 1 Homo-polymerizations of DMDL and random copolymerizations of DMDL with co-monomers



Entry	Co-monomer (M)	Initiator	Solvent ^a	[M] ₀ /[DMDL] ₀ /[initiator] ₀ (equiv.)	T (°C)	t (h)	Conv (M/DMDL) ^b (%/%)	M _n ^c	D ^c	F _{DMDL} ^d (%)
1	None	AIBN	Toluene	0/100/1	70	12	NA/82	10 300	1.61	100 (100)
2	None	AIBN	Bulk	0/100/0.5	70	1.5	NA/79	104 000	2.04	100 (100)
3	MMA	AIBN	Toluene	50/50/1	70	7	99/23	19 000	1.78	18 (19)
4	BA	AIBN	Toluene	50/50/1	70	1	98/56	130 000	1.53	37 (36)
5	St	AIBN	Toluene	50/50/2	80	12	79/NA ^e	5100	1.46	8 (NA)
6	AN	AIBN	EC	50/50/1	70	1	80/44	77 000	1.54	34 (35)
7	VAc	AIBN	Toluene	50/50/1	70	12	84/48	11 000	1.60	35 (36)
8	NVP	AIBN	DMF	50/50/1	70	2	96/28	29 000	1.81	19 (23)
9	DMA	AIBN	Toluene	50/50/1	70	1	100/39	91 000	1.95	28 (29)
10	PEGMA	V65	Toluene	50/50/1	60	4	92/NA ^e	39 000	2.22	17 (NA)
11	DMAEMA	AIBN	Toluene	50/50/1	70	4	98/20	27 000	1.62	17 (17)
12	BMA	AIBN	Toluene	50/50/1	70	5	93/11	46 000	1.54	10 (11)
13	LMA	AIBN	Toluene	50/50/1	70	3	85/15	42 000	1.76	16 (15)
14	BzMA	AIBN	Toluene	50/50/1	70	3	92/26	39 000	1.76	19 (22)
15	EHMA	AIBN	Toluene	50/50/1	70	2	97/18	38 000	1.95	13 (16)
16	SMA	AIBN	Toluene	50/50/1	70	2	77/29	86 000	1.96	28 (27)
17	MEA	AIBN	Toluene	50/50/1	70	1	100/54	82 000	2.00	32 (35)

^a Dilution with 50 wt% solvent for entries 1 and 3–17. EC is ethylene carbonate. DMF is *N,N*-dimethylformamide. ^b Monomer conversion obtained with ¹H NMR. ^c PMMA-calibrated THF-GPC values for entries 1, 3–5, 7, and 12–17 and PMMA-calibrated DMF-GPC values for entries 2, 6, and 8–11.

^d F_{DMDL} value determined by the ¹H NMR analysis of the purified polymer. In the parenthesis, F_{DMDL} value calculated from the monomer conversions of DMDL and co-monomer determined using ¹H NMR is given. ^e Monomer conversion of DMDL was unable to determine because the DMDL monomer peak overlapped with the co-monomer peak in the ¹H NMR spectrum.

Random copolymerizations of DMDL and PhDL with various co-monomers

We carried out a random copolymerization of DMDL (50 mol%) with methyl methacrylate (MMA) (50 mol%) in toluene at 70 °C (Table 1, entry 3). Fig. 2a shows the plots of the conversions of MMA (square) and DMDL (circle) vs. time. MMA was consumed faster than DMDL. After 7 h, the conversions of MMA and DMDL were 99% and 23%, respectively, yielding a random copolymer of MMA and DMDL with $M_n = 19\ 000$ and $D = 1.78$. The fraction (F_{DMDL}) of DMDL in the copolymer was 18%. The ¹H and ¹³C NMR analysis showed that the polymer contained both MMA and DMDL units and the DMDL units were incorporated in the ring-retained form (Fig. S6 and S7 in ESI†). The monomer reactivity ratio was determined by the Fineman-Ross method³⁸ (Fig. S8–S12 in ESI†) to be $r_{MMA} = 4.60$ and $r_{DMDL} \approx 0$ (Table 3, entry 1). The result means that the MMA terminal radical reacts with both MMA and DMDL and 4.60 times more reacts with MMA than DMDL and that the DMDL terminal radical virtually exclusively reacts with MMA.

We studied *n*-butyl acrylate (BA) as another family of monomer. We carried out a random copolymerization of DMDL

(50 mol%) with BA (50 mol%) in toluene at 70 °C (Table 1, entry 4). Fig. 2b shows the plots of the conversions of BA (triangle) and DMDL (circle) vs. time. Similar to MMA, the co-monomer BA was consumed faster than DMDL. After 1 h, the conversions of BA and DMDL were 98% and 56%, respectively, yielding a random copolymer of BA and DMDL with $M_n = 130\ 000$ and $D = 1.53$ (Fig. S13 in ESI†). The monomer reactivity ratios were $r_{BA} = 0.83$ and $r_{DMDL} \approx 0.03$ (a range of 0–0.10 with an experimental error) (Table 3, entry 2). These values means that the BA/DMDL system ($r_{BA} = 0.83$ (Table 3, entry 2)) has a more alternating tendency than the MMA/DMDL system ($r_{MMA} = 4.60$ (Table 3, entry 1)) but that the DMDL terminal radical still virtually exclusively reacts with BA over DMDL ($r_{DMDL} \approx 0.03$ (Table 3, entry 2)).

We also studied styrene (St), acrylonitrile (AN), and vinyl acetate (VAc) as other families of monomer (Table 1, entries 5–7). We carried out random copolymerizations of DMDL (50 mol%) with the co-monomer (50 mol%) in toluene (for St and VAc) or ethylene carbonate (for AN), yielding polymers with $M_n = 5100$ –77 000 and $D = 1.46$ –1.60 at $F_{DMDL} = 8$ –35% (Fig. S14–S16 in ESI†). For the St and AN systems, the monomer



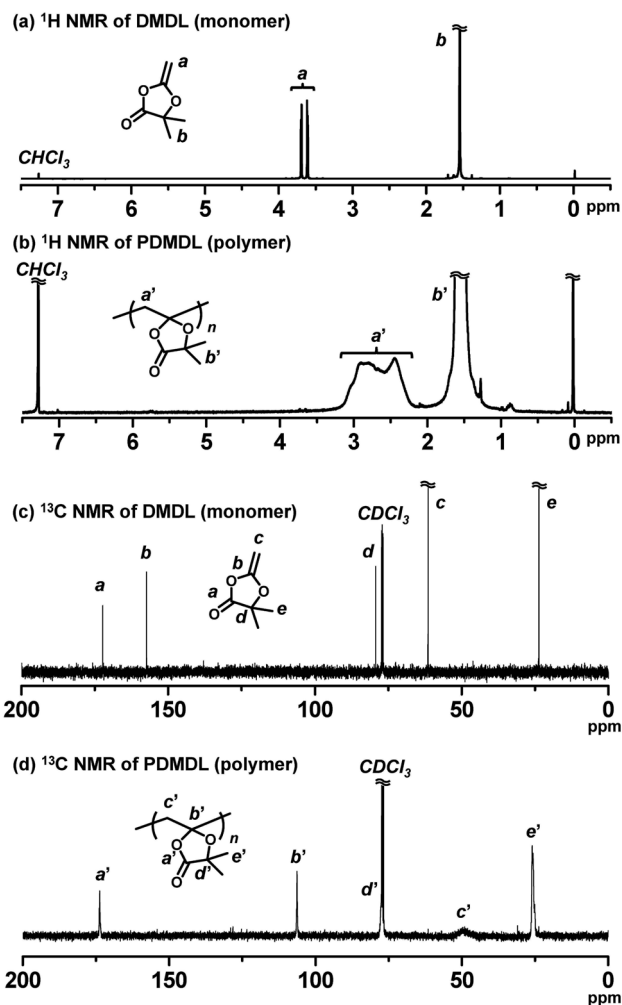


Fig. 1 ^1H NMR spectra of (a) DMDL and (b) PDMDL. ^{13}C NMR spectra of (c) DMDL and (d) PDMDL. The synthesis of PDMDL is given in Table 1 (entry 1). 400 MHz for ^1H NMR (100 MHz for ^{13}C NMR) in CDCl_3 .

reactivity ratios were $r_{\text{St}} = 1.08$ and $r_{\text{DMDL}} \approx 0.01$ (0–0.10 with an experimental error) (Table 3, entry 3) and $r_{\text{AN}} = 0.28$ and $r_{\text{DMDL}} \approx 0.04$ (0–0.10 with an experimental error) (Table 3, entry 4). The r_{St} and r_{AN} values mean that the St terminal radical reacts nearly equally with DMDL and St, whereas the AN terminal radical reacts 3.6 (= $1/0.28$) times more with DMDL than AN. The nearly zero r_{DMDL} values mean that the DMDL terminal

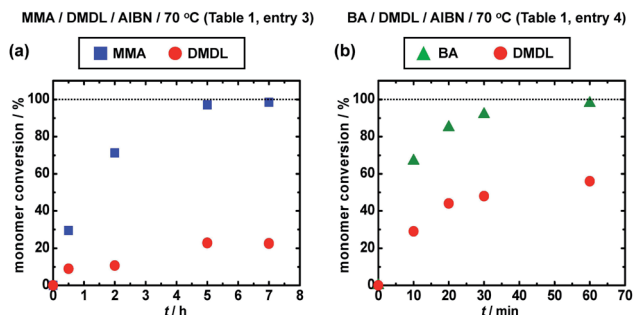


Fig. 2 Plots of monomer conversion vs. time for (a) the MMA/DMDL/AIBN (50/50/1) system (70 °C) (Table 1, entry 3) and (b) the BA/DMDL/AIBN (50/50/1) system (70 °C) (Table 1, entry 4). The solvent was toluene (50 wt%) for both systems. The symbols are indicated in the figure.

Table 3 Monomer reactivity ratios^a of co-monomers (M_1) and DMDL (M_2) determined by the Fineman–Ross method

Entry	M_1	T (°C)	r_1^a	r_2^a
1	MMA	60	4.60	Close to zero
2	BA	60	0.83	0.03^b
3	St	80	1.08	0.01^b
4	AN	70	0.28	0.04^b
5	VAc	70	1.65	0.29

^a $r_1 = r_{12}/r_{11}$ and $r_2 = r_{21}/r_{22}$. ^b $r_{\text{DMDL}} = 0\text{--}0.10$ with an experimental error.

radical predominantly reacts with St and AN over DMDL. The VAc system had $r_{\text{VAc}} = 1.65$ and $r_{\text{DMDL}} = 0.29$ (Table 3, entry 5). The r_{DMDL} value is not nearly zero in the VAc system (Table 3, entry 5), in contrast to the mentioned four systems (Table 3, entries 1–4). The result means that the DMDL terminal radical reacts with VAc slowly and hence can react with DMDL to some extent in the VAc system.

In the literatures, random copolymerizations of CKAs and vinyl co-monomers also showed similar tendency, *i.e.*, slower consumption of CKAs than the co-monomers. The $r_{\text{co-monomer}}$ values are typically 2–8 and the r_{CKA} values are 0.01–1 (Table S1 in ESI†).^{29,39–43} For MBDO (previously reported CKHE monomer) (Scheme 2), its random copolymerization with VAc had $r_{\text{VAc}} =$

Table 2 Homo-polymerization of PhDL and random copolymerizations of PhDL with co-monomers

Entry	Co-monomer (M)	Initiator ^a	Solvent ^b	$[M]_0/[PhDL]_0/[initiator]_0$ (mM)	T (°C)	t (h)	Conv (M/PhDL) ^c (%/%)	M_n^d	D^d	F_{PhDL}^e (%)
1	None	TBPB	Toluene	0/100/1	120	12	NA/37	<1000	NA	100 (100)
2	MMA	AIBN	Toluene	50/50/1	70	7	93/15	15 000	1.83	11 (14)
3	MMA	TBPB	Toluene	50/50/1	120	5	94/33	21 000	1.45	24 (26)
4	BA	AIBN	Toluene	50/50/1	70	3	96/39	47 000	2.08	26 (29)
5	BA	TBPB	Toluene	50/50/1	120	3.5	96/48	13 000	2.63	31 (33)

^a TBPB is *t*-butyl peroxybenzoate. ^b Dilution with 50 wt% solvent. ^c Monomer conversion obtained with ^1H NMR. ^d PMMA-calibrated THF-GPC values. ^e F_{PhDL} value determined by the ^1H NMR analysis of the purified polymer. In the parenthesis, F_{PhDL} value calculated from the monomer conversion (monomer consumption) determined using ^1H NMR is given.



0.587 and $r_{\text{MBDO}} = 1.51$ (60 °C), showing that MBDO has higher reactivity than VAc.³¹

Thermal properties of the obtained DMDL copolymers (Table 1, entries 3 and 5–7) with MMA, St, AN, and VAc were studied (Fig. S17–S20 in ESI†). The T_g values of the copolymers with MMA (104 (±5) °C), St (90 (±5) °C), and AN (76 (±5) °C) were slightly lower than those of homopolymers of MMA (*ca.* 105 °C),⁴⁴ St (*ca.* 90–100 °C),⁴⁴ and AN (*ca.* 95 °C),⁴⁴ while the T_g value of the copolymer with VAc (38 (33–46) °C) was slightly higher than that of a homopolymer of VAc (*ca.* 30 °C),⁴⁴ because the T_g value of a homopolymer of DMDL is 41 °C.

We further expanded the co-monomer scope to *N*-vinylpyrrolidone (NVP) and *N,N*-dimethylacrylamide (DMA) as other families of monomer (Table 1, entries 8 and 9) and functional methacrylates with poly(ethylene glycol) methyl ether (PEGMA), dimethylamino (DMAEMA), butyl (BMA), lauryl (LMA), benzyl (BzMA), ethylhexyl (EHMA), and stearyl (SMA) groups (Table 1, entries 10–16) as well as a functional acrylate with a methoxyethyl group (MEA) (Table 1, entry 17), yielding co-polymers with $M_n = 27\,000$ –91 000 and $\mathcal{D} = 1.54$ –2.22 at $F_{\text{DMDL}} = 10$ –32% (Fig. S21–S30 in ESI†). Such a wide range of amenable monomer families and functional monomers, including hydrophobic (BMA, LMA, BzMA, EHMA, SMA, and MEA), hydrophilic (NVP, DMA, PEGMA, and DMAEMA), and biocompatible (PEGMA and MEA) monomers, are attractive for polymer design by DMDL.

While the homopolymerization of PhDL generated only an oligomer (Table 2, entry 1), PhDL was able to be copolymerized with a methacrylate and an acrylate (Table 2, entries 2–5). We carried out a random copolymerization of PhDL (50 mol%) with MMA (50 mol%) in toluene at 70 °C (Table 2, entry 2), which is the same condition as that in the DMDL/MMA system (Table 1, entry 3) except using PhDL instead of DMDL. After 7 h, the conversions of MMA and PhDL were 93% and 15%, respectively, yielding a random copolymer of MMA and PhDL with $M_n = 15\,000$ and $\mathcal{D} = 1.83$ (Fig. S31 in ESI†). The conversion of PhDL (15%) was smaller than that of DMDL (23%) at similar conversions of MMA (93–99%). In order to increase the conversion of PhDL, we elevated the polymerization temperature from 70 °C to 120 °C (Table 2, entry 3). After 5 h, the conversions of MMA and PhDL were 94% and 33%, respectively, yielding a random copolymer with $M_n = 21\,000$ and $\mathcal{D} = 1.45$. The conversion of PhDL increased by 18% (from 15% to 33%) by elevating the temperature from 70 °C to 120 °C at the similar (93–94%) conversions of MMA.

We also conducted a random copolymerization of PhDL (50 mol%) with BA (50 mol%) in toluene at 70 °C (Table 2, entry 4). After 3 h, the conversions of BA and PhDL were 96% and 39%, respectively, yielding a random copolymer with $M_n = 47\,000$ and $\mathcal{D} = 2.08$ (Fig. S32 in ESI†). At a higher temperature of 120 °C, the conversions of BA and PhDL were 96% and 48%, respectively (Table 2, entry 5), showing an increase in the conversion of PhDL by 9% by elevating the temperature from 70 °C to 120 °C at the same (96%) conversion of BA.

Thus, DMDL and PhDL were able to be copolymerized with a range of vinyl monomers, offering a wide scope of accessible polymers. This feature is in sharp contrast to the limited co-

monomer scope of the previously reported CKHE monomer, *i.e.*, MBDO, which was copolymerized only with VAc.

Hydroxide-assisted degradation of PDMDL

We studied the degradation of PDMDL. We dissolved a PDMDL homopolymer ($M_n = 104\,000$ and $\mathcal{D} = 2.04$ (Table 1, entry 2)) (0.064 g, 2.4 wt%, 1 equiv. of the DMDL monomer unit) in DMSO- d_6 (2.08 g, 77.2 wt%), which was a colorless transparent solution. To this solution, a D₂O solution of 1 M NaOH (0.55 g, 20.4 wt%, 1 equiv. of NaOH) was added at room temperature. The mixture became slightly turbid, because PDMDL chains became globular due to the presence of water (D₂O), but the mixture soon turned reddish brown and transparent upon stirring at room temperature. PDMDL seemed to degrade to small molecules. The reaction mixture was subsequently analyzed using GPC and NMR within 5 h after the reaction started.

In the GPC chromatogram (DMF eluent) (Fig. 3a), no peak was observed in the polymer region (molecular weight >2000), suggesting that PDMDL degraded. (The peak at the molecular weight <2000 observed in Fig. 3a would be ascribed to generated oligomers and/or possible clusters of lithium bromide (LiBr) used in the DMF eluent.) The ¹H (Fig. 3b) and ¹³C (Fig. 3c) NMR spectra also show no polymer (PDMDL) signals (no signal at 2.15–3.25 ppm for the backbone CH₂ protons in the ¹H NMR

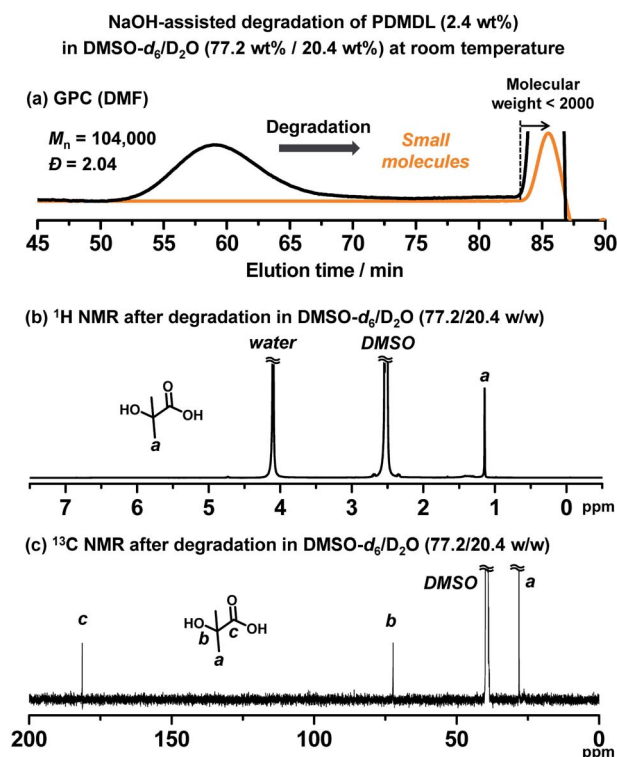


Fig. 3 Degradation of PDMDL ($M_n = 104\,000$, $\mathcal{D} = 2.04$, 2.4 wt%, 1 equiv. of the DMDL monomer unit) using 1 equiv. of NaOH in a mixture of D₂O (20.4 wt%) and DMSO- d_6 (77.2 wt%) at room temperature within 5 h. (a) GPC chromatogram before (black line) and after (orange line) degradation. (b) ¹H NMR and (c) ¹³C NMR spectra of the reaction mixture after degradation (400 MHz for ¹H NMR (100 MHz for ¹³C NMR) in DMSO- d_6 /D₂O).



spectrum and no signal at 174 ppm for the ester carbon of the DMDL unit in the polymer in the ^{13}C NMR spectrum) (the ^1H and ^{13}C NMR spectra of (non-degraded) PDMDL (2.4 wt%) in the studied $\text{DMSO-}d_6/\text{D}_2\text{O}$ (77.2/20.4 wt%) mixed solvent are given in ESI (Fig. S33 and S34 †)). The only species clearly detected was HIBA (Scheme 3) whose CH_3 protons appeared at 1.13 ppm (Fig. 3b) and whose CH_3 carbons, quaternary carbon, and carbonyl carbon appeared at 28, 73, and 181 ppm, respectively (Fig. 3c).

Scheme 4 shows a possible degradation mechanism of PDMDL in the presence of OH^- . It should be emphasized that this scheme is a possible mechanism but is not a definitive one at this moment. Firstly, OH^- attacks the carbonyl carbon of the hemiacetal ester, generating an alkoxide anion (Scheme 4, compound 5). A subsequent elimination can result in either (a) the side-chain cleavage or (b) the main-chain cleavage. The side-chain cleavage can generate a ketone in the backbone chain (compound 6) and HIBA. If the side-chain cleavage is significant, a polyketone can be generated. The main-chain cleavage can generate a carboxylic acid chain-end PDMDL (compound 8) and a carboxylate chain-end PDMDL (compound 9). These two compounds can further generate another form of carboxylic acid chain-end PDMDL (compound 10) and a ketone chain-end PDMDL (compound 11) as well as two molecules of HIBA. If the main-chain cleavage is significant, the polymers (compounds 8–11) will further undergo the main-chain cleavage and degrade into small molecules. A main-chain cleavage mechanism was previously proposed for the degradation of the MBDO polymer, for which the mechanism was studied in an acidic condition. 31

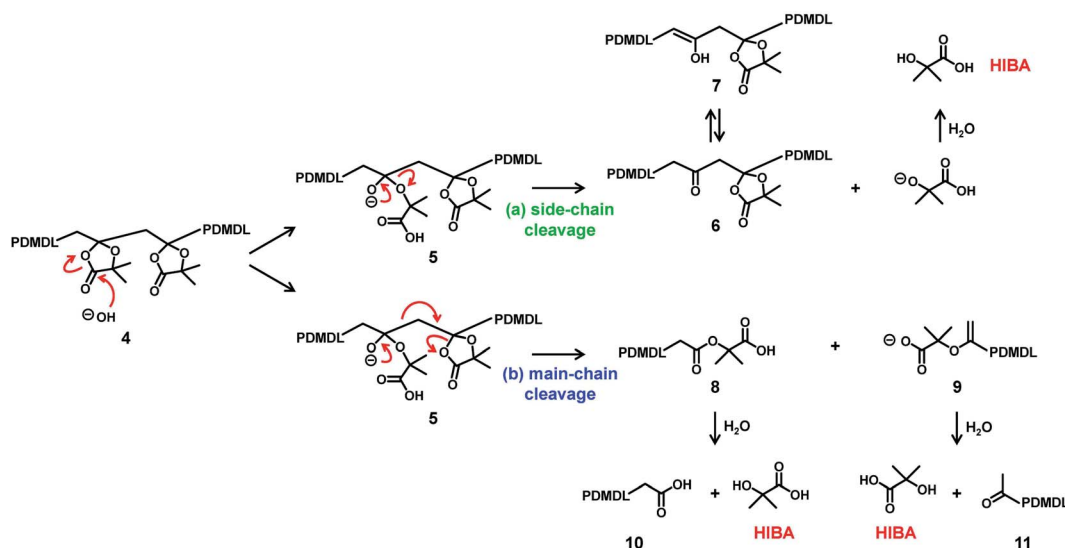
No polymer peak observed in the GPC chromatogram (Fig. 3a) suggests that the main-cleavage operated significantly in the present system so that the polymer chains could degrade into small molecules or oligomers. The side-chain cleavage is not excluded at this moment and might also operate to some extent. HIBA is a product in both pathways. The observation of HIBA in the NMR analysis (Fig. 3b and c) supports the proposed

degradation scheme (either or both of the two pathways). However, the ^1H and ^{13}C NMR analyses detected no other species, and the reason is unclear at the moment. When the main-chain cleavage operates, small molecules or oligomers with carboxylic acid (compounds 8 and 10), carboxylate (compound 9), and/or ketone (compound 11) can be generated. However, the ^{13}C NMR analysis (Fig. 3c) detected no other $\text{C}=\text{O}$ carbons than that of HIBA. When the side-chain cleavage operates, a polyketone (compound 6) or its isomer, *i.e.*, a poly-enol (compound 7) can be generated, and the intramolecular cyclization of polyketone can also generate aromatic species. 45 However, the ^{13}C NMR analysis (Fig. 3c) detected no ketone (>190 ppm), no enol (90–100 ppm), or no aromatic species (110–140 ppm). Small molecules that were possibly generated might be volatile and evaporate during the degradation. Thus, we also conducted the same experiment in a valved (sealed) NMR tube (Fig. S35 in ESI †). However, again, only HIBA was detected. A possible reason for no detection of other species (compounds 6–11) is the solubility of the product. To check this possibility, we studied the degradation of PDMDL in a more hydrophobic solvent (a mixture of $\text{THF-}d_8$ and D_2O instead of a mixture of $\text{DMSO-}d_6$ and D_2O), as described in detail in ESI. The ^1H NMR and ^{13}C NMR spectra of the reaction mixture (Fig. S36 in ESI †) again showed only HIBA even in this more hydrophobic solvent. Thus, the reason for no detection of other species is not clear at this moment, and hence the proposed degradation scheme (Scheme 4) is viewed as a tentative mechanism.

Nevertheless, as an experimental fact, PDMDL clearly degraded in the presence of NaOH . The degradation occurred within 5 h at room temperature or markedly fast (within 30 min) at 50°C , which would be an attractive feature of PDMDL.

Acid-assisted degradation of PDMDL

We also studied the degradation of PDMDL in an acidic condition, using the same condition as previously reported



Scheme 4 Possible mechanisms of hydroxide-assisted degradation of PDMDL.



condition for the acid-assisted degradation of the MBDO polymer.³¹ We dissolved a PDMDL ($M_n = 106\ 000$ and $D = 1.88$) (0.064 g, 2.4 wt%, 1 equiv. of the DMDL monomer unit) in DMSO (2.08 g, 78.7 wt%), which was a colorless transparent solution. To this solution, an aqueous 1 M HCl solution (0.5 g, 18.9 wt%, 1 equiv. of HCl) was added, and the mixture became turbid. The mixture was heated at 80 °C for 24 h, and the generated precipitate was analyzed with GPC (Fig. S37 in ESI†). The precipitate was found to be unreacted PDMDL, and its M_n (= 100 000) and D (= 2.00) values were close to its original values ($M_n = 106\ 000$ and $D = 1.88$). The amount of the precipitate (unreacted PDMDL) was 0.057 g, which is 89% of the original amount (0.064 g). Thus, PDMDL did not degrade in the same condition as the MBDO polymer did, mainly because of the low solubility of PDMDL in the studied acidic condition.

Thus, we used a more hydrophobic solvent (a mixture of DMSO- d_6 (79.4 wt%) and H₂O (1.4 wt%)) to dissolve PDMDL (10.2 wt%) and used an organic acid, *i.e.*, trifluoroacetic acid (9 wt%), that is soluble in this solvent. We heated the mixture at 80 °C for 3 days and observed a decrease in the M_n value by 25%, showing that PDMDL slowly degraded in this acidic condition (Fig. S38 in ESI†). We observed HIBA in this acid-assisted degradation, as in the NaOH-assisted degradation. Furthermore, acetone was observed in the acid-assisted degradation (Fig. S38 in ESI†), which may suggest a chain-end degradation of **11** in the acidic condition (Scheme S1 in ESI†), although this mechanism is a tentative mechanism. While the acid-assisted degradation of the MBDO polymer generated acetic acid,³¹ we did not observe acetic acid in the degradation of PDMDL. The degradation mechanism might be different for the MBDO polymer and PDMDL.

Amine-assisted degradation of PDMDL

The NaOH-assisted degradation was too fast to monitor over time. Hence, we used a milder base, *i.e.*, an amine to probe the degradation of PDMDL. The studied amine was a primary amine, *i.e.*, pentylamine (R-NH₂). We dissolve a PDMDL ($M_n = 39\ 000$ and $D = 1.42$) (0.064 g, 4.0 wt%, 1 equiv. of the DMDL monomer unit) and pentylamine (0.065 g, 4.0 wt%, 1.5 equiv.) in CD₃CN (1.48 g, 92.0 wt%), which was a colorless transparent solution. The solution was heated at 65 °C. Fig. 4a and b show the GPC chromatograms and the plot of M_n vs. the reaction time, respectively. The original M_n (= 39 000) value of PDMDL started to decay ($M_n = 33\ 000$) right after the addition of the amine and became approximately a half ($M_n = 20\ 000$) after the heating for 4 h, approximately a quarter ($M_n = 10\ 000$) for 8 h, and approximately 1/10 ($M_n = 3000$) for 24 h. The polymer degraded to small molecules or oligomers ($M_n < 2000$) for 60 h.

The ¹H (Fig. 4c) and ¹³C (Fig. 4d) NMR spectra of the reaction mixture for 60 h show no polymer (PDMDL) signals (no signal at 2.38–3.13 ppm for the backbone CH₂ protons in the ¹H NMR spectrum or no signal at 174 ppm for the ester carbon of the DMDL unit in the polymer in the ¹³C NMR spectrum) (the ¹H and ¹³C NMR spectra of (non-degraded) PDMDL in the studied CD₃CN are given in Fig. S39 and S40 in ESI†). A possible mechanism is given in Scheme S2 in ESI.† We observed 2-

hydroxy-2-methyl-*N*-pentylpropanamide (HMPPA) (product), HIBA (product), and pentylamine (remaining reactant) (Fig. 4c and d), which were identified from the spectra of the isolated HMPPA, HIBA, and pentylamine (Fig. S41 and S42 in ESI†). The observation of HMPPA supports the proposed amine-assisted degradation scheme (Scheme S2 in ESI†). The degradation was carried out without removing moisture. HIBA (Scheme 4) was generated probably because of the presence of moisture, which might provide hydroxide in the presence of the amine. However, similar to the NaOH-assisted degradation, no other products than HMPPA and HIBA were observed in the ¹H and ¹³C NMR spectra (Fig. 4c and d) for an unclear reason.

Although the degradation mechanism is yet to be definitive, this experiment demonstrated that PDMDL degraded even with a milder base (amine). The facile operation (no need to remove moisture) and mild temperature (65 °C) are attractive.

Degradation of PPEGMA-*r*-PDMDL

Besides the PDMDL homopolymer, we studied the degradation of a PPEGMA-*r*-PDMDL random copolymer, where PPEGMA is poly(ethylene glycol) methyl ether methacrylate. This random copolymer is hydrophilic and soluble in water. We dissolved a PPEGMA-*r*-PDMDL ($M_n = 39\ 000$, $D = 2.22$, and $F_{\text{DMDL}} = 17\%$) (0.06 g, 1 wt%, 1 equiv. of the DMDL monomer unit) in a 0.9 M KOH aqueous solution (6.0 g, 99 wt%, 67 equiv. of KOH) and monitored the degradation at room temperature for 23 h. Fig. 5a and b show the GPC chromatograms and the plot of M_n vs. time. The original M_n (= 39 000) value of the copolymer became approximately a half ($M_n = 21\ 000$) for 1 h, approximately 1/3 ($M_n = 13\ 000$) for 8 h, and approximately a quarter ($M_n = 9300$) for 23 h. To confirm that the observed decrease in the M_n value was not ascribed to the pendant chain cleavage of the PEGMA units, we studied the degradation of a PPEGMA homopolymer ($M_n = 72\ 000$ and $D = 2.82$) in the same condition. After 24 h, the M_n value decreased from 72 000 to only 70 000 (Fig. S43 in ESI†), confirming no significant pendant chain cleavage. Thus, the observed degradation (significant decrease in the M_n value) of the PPEGMA-*r*-PDMDL is ascribed to the main-chain cleavage of the DMDL units in the polymer. The F_{DMDL} value of the PPEGMA-*r*-PDMDL was 17%, meaning that a DMDL unit was incorporated in every approximately 6 PEGMA units on average. Upon the full main-chain cleavage of the DMDL units, the M_n value may decrease to 1800 (6 units of PEGMA). The observed M_n value (=9400) after 23 h of the reaction was larger than 1800, possibly because of a compositional drift during the polymerization (smaller DMDL compositions in the polymers generated at an early stage of polymerization and larger DMDL compositions in those generated at a later stage of polymerization).⁴⁶

Degradation of PLMA-*r*-PDMDL

We also studied the degradation of a PLMA-*r*-PDMDL random copolymer, where PLMA is poly(lauryl methacrylate). This random copolymer is hydrophobic. We used a mixture of methanol and THF to dissolve this hydrophobic copolymer. We dissolved a PLMA-*r*-PDMDL ($M_n = 42\ 000$, $D = 1.76$, and F_{DMDL}



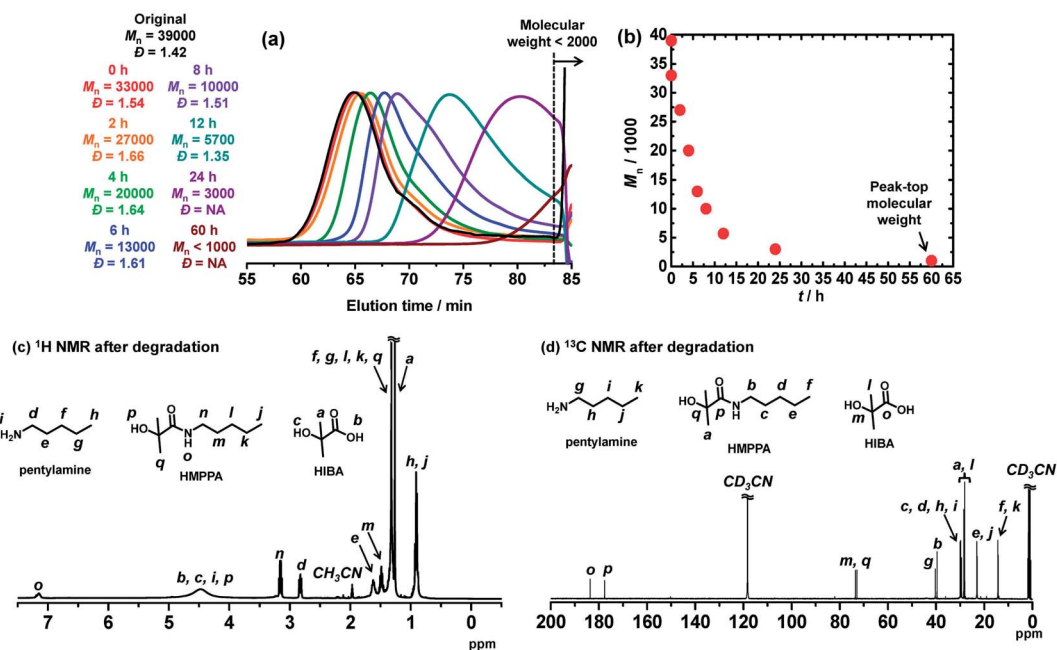
Degradation of PDMDL (4.0 wt%) using pentylamine (4.0 wt%) in CD₃CN (92.0 wt%) at 65 °C.

Fig. 4 Degradation of PDMDL ($M_n = 39\ 000$, $\bar{D} = 1.42$, 4.0 wt%, 1 equiv. of the DMLD unit) using pentylamine (4.0 wt%, 1.5 equiv.) in CD₃CN (92.0 wt%) at 65 °C. (a) GPC chromatograms and (b) plot of M_n vs. degradation time. The 0 h sample in (a) is a sample right after the addition of pentylamine. (c) ¹H NMR and (d) ¹³C NMR spectra of the reaction mixture after degradation for 60 h (400 MHz for ¹H NMR (100 MHz for ¹³C NMR) in CD₃CN).

= 16%) (0.04 g, 2.0 wt%, 1 equiv. of the DMDL monomer unit) in a mixture of a 0.9 M KOH methanol solution (0.16 g, 8.1 wt%, 3.5 equiv. of KOH) and THF (1.78 g, 89.9 wt%) and monitored the degradation at room temperature for 24 h (Fig. 5c and d). The original M_n (= 42 000) value of the copolymer became approximately its 60% ($M_n = 25\ 000$) for 2 h, approximately its 45% ($M_n = 19\ 000$) for 8 h, and approximately its 30% ($M_n = 13\ 000$) for 24 h. Thus, the degradation of the PLMA-*r*-PDMDL successfully occurred.

Recovery of HIBA via NaOH-assisted degradation of PDMDL

HIBA is the starting material (compound 1 ($R_1 = \text{CH}_3$) in Scheme 3) of the DMDL monomer. Hence, we attempted to recover HIBA after the NaOH-assisted degradation of PDMDL,

use the recovered HIBA to synthesize DMDL monomer, and use this monomer to synthesize PDMDL for demonstrating the concept of the recycling of PDMDL (Scheme 3). As mentioned, four of the six carbons of the DMDL monomer are recycled via the recovery of HIBA.

A PDMDL ($M_n = 56\ 000$ and $\bar{D} = 2.47$) (1.0 g, 1 equiv. of the DMDL monomer unit) was dissolved in a mixture of a 0.9 M KOH methanol solution (28 g, 4 equiv. of KOH) and THF (300 g) and was stirred at room temperature for 1 h. The mixture of methanol and THF was used instead of a mixture of DMSO and water (Fig. 3), because DMSO is difficult to evaporate (in the evaporation process below). KOH was used instead of NaOH (Fig. 3), because KOH is soluble in the mixture of methanol and THF. After the reaction, acidic water and diethyl ether were

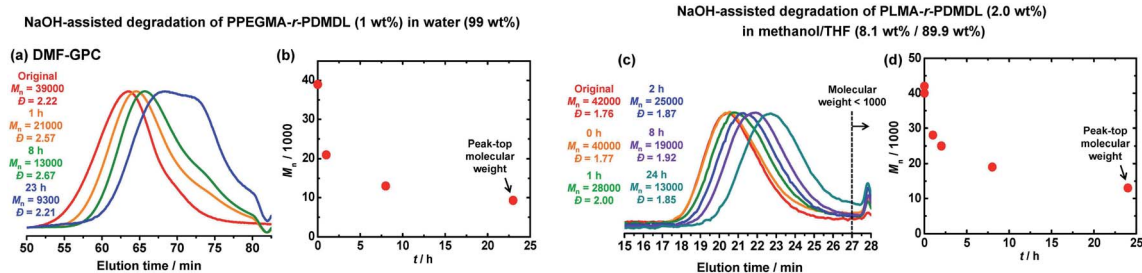


Fig. 5 Degradation of PPEGMA-*r*-PDMDL ($M_n = 39\ 000$, $\bar{D} = 2.22$, and $F_{\text{DMDL}} = 17\%$) (1 wt%) using 67 equiv. of KOH in water (99 wt%) at room temperature; (a) GPC chromatograms and (b) plot of M_n vs. time. Degradation of PLMA-*r*-PDMDL ($M_n = 42\ 000$, $\bar{D} = 1.76$, and $F_{\text{DMDL}} = 16\%$) (1.8 wt%) with 3.5 equiv. of KOH in a mixture of methanol (8.1 wt%) and THF (89.9 wt%) at room temperature; (c) GPC chromatograms and (d) plot of M_n vs. time.



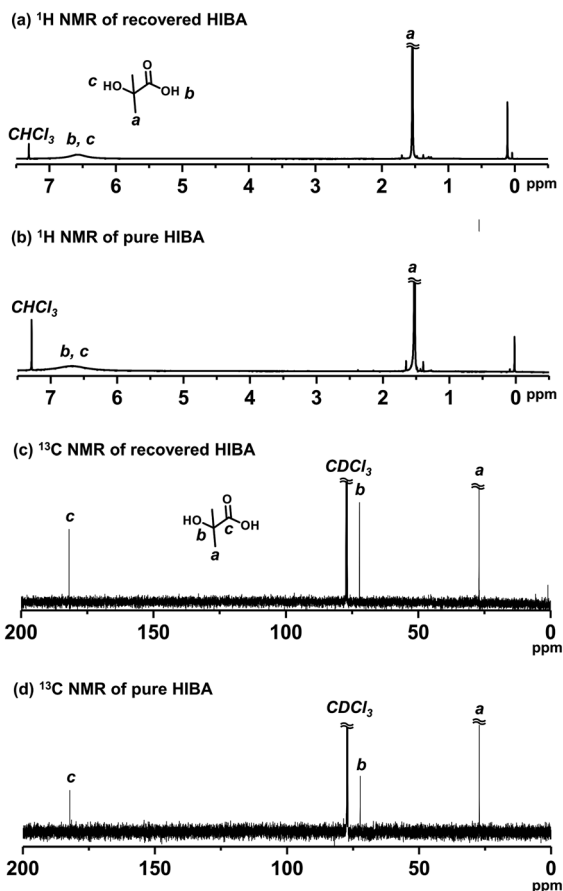


Fig. 6 ^1H NMR spectra of (a) recovered HIBA and (b) pure HIBA. ^{13}C NMR spectra of (c) recovered HIBA and (d) pure HIBA. (400 MHz for ^1H NMR (100 MHz for ^{13}C NMR) in CDCl_3).

added to the reaction mixture. HIBA was more soluble in the organic phase than in the aqueous phase in an acidic condition. After the extraction with diethyl ether three times, the collected organic layer was evaporated. The ^1H and ^{13}C NMR spectra (Fig. 6) show that the compound extracted in the organic layer was nearly pure HIBA (we observed only HIBA). We collected 0.39 g of HIBA. The theoretically recoverable maximum amount of HIBA from 1 g of PDMDL is 0.81 g. Thus, the recovery yield was 48% ($= (0.39 \text{ g}) / (0.81 \text{ g})$).

We used the recovered HIBA to synthesize DMDL (Fig. S45–49 in ESI†). We carried out a polymerization of this DMDL (100 equiv.) with AIBN (0.5 equiv.) at 70 °C for 6 h (in the same condition as Table 1, entry 2) and obtained a PDMDL with $M_n = 52\,000$ and $\mathcal{D} = 2.32$ with an 80% monomer conversion. Thus, we successfully experimentally demonstrated the concept of the chemical recycling of PDMDL.

Conclusions

We successfully synthesized polymers using two new CKHE monomers (DMDL and PhDL). The two monomers were synthesized in high yields (80–90%), which is an attractive feature. DMDL generated homopolymers, and DMDL and PhDL were copolymerized with important families of vinyl monomers,

i.e., methacrylates, acrylates, styrene, acrylonitrile, vinyl pyrrolidinone, and acrylamide, and various functional methacrylates and acrylate, exhibiting a wide scope of the accessible polymers. The obtained homopolymers and random copolymers of DMDL degraded in the presence of a hydroxide (NaOH or KOH) and an amine (pentylamine) at mild temperatures (room temperature to 65 °C). The degradation of the PDMDL homopolymer generated HIBA. We recovered HIBA, used the recovered HIBA to synthesize DMDL monomer, and used this monomer to synthesize PDMDL, demonstrating the concept of the chemical recycling of PDMDL. HIBA is not only an ingredient to regenerate DMDL but also a useful ingredient of other important chemicals such as methacrylic acid, tetramethylglycolide, and chloro- and amino-derivatives of HIBA^{47,48} and can serve for a range of chemical recycling. The degradability and chemical recyclability of PDMDL may be useful for circular materials economy.

Data availability

All supporting data have been included in the ESI.†

Author contributions

X. Y. Oh performed all experiments. A. Goto and X. Y. Oh contributed to the conception of the experiments, discussion of the results and preparation of the manuscript. Y. Ge assisted in monomer synthetic work.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 N. Kamaly, B. Yameen, J. Wu and O. C. Farokhzad, *Chem. Rev.*, 2016, **116**, 2602–2663.
- 2 M. Hong and E. Y. X. Chen, *Green Chem.*, 2017, **19**, 3692–3706.
- 3 E. Guegain, J. Tran, Q. Deguettes and J. Nicolas, *Chem. Sci.*, 2018, **9**, 8291–8306.
- 4 T. P. Haider, C. Volker, J. Kramm, K. Lanfester and F. R. Wurm, *Angew. Chem., Int. Ed.*, 2019, **58**, 50–62.
- 5 K. H. Paek and S. G. Im, *Green Chem.*, 2020, **22**, 4570–4580.
- 6 K. W. Meereboer, M. Misra and A. K. Mohanty, *Green Chem.*, 2020, **22**, 5519–5558.
- 7 A. Barron and T. D. Sparks, *iScience*, 2020, **23**, 101353.



- 8 Y. Xu, S. Sen, Q. Wu, X. Zhong, R. H. Ewoldt and S. C. Zimmerman, *Chem. Sci.*, 2020, **11**, 3326.
- 9 J. Yang and Y. Xia, *Chem. Sci.*, 2021, **12**, 4389.
- 10 A. Tardy, J. Nicolas, D. Gigmes, C. Lefay and Y. Guillaneuf, *Chem. Rev.*, 2017, **117**, 1319–1406.
- 11 A. W. Jackson, *Polym. Chem.*, 2020, **11**, 3525–3545.
- 12 J. Folini, W. Murad, F. Mehner, W. Meier and J. Gaitzsch, *Eur. Polym. J.*, 2020, **134**, 109851.
- 13 T. Pesenti and J. Nicolas, *ACS Macro Lett.*, 2020, **9**, 1812–1835.
- 14 J. Folini, C.-H. Huang, J. C. Anderson, W. P. Meier and J. Gaitzsch, *Polym. Chem.*, 2019, **10**, 5285.
- 15 W. Bailey, Z. Ni and S.-R. Wu, *J. Polym. Sci., Polym. Chem. Ed.*, 1982, **20**, 3021–3030.
- 16 A. Tardy, J.-C. Honoré, D. Siri, J. Nicolas, D. Gigmes, C. Lefay and Y. Guillaneuf, *Polym. Chem.*, 2017, **8**, 5139–5147.
- 17 G. G. Hedir, M. C. Arno, M. Langlais, J. T. Husband, R. K. O'Reilly and A. P. Dove, *Angew. Chem., Int. Ed.*, 2017, **56**, 9178–9182.
- 18 J. Tran, T. Pesenti, J. Cressonnier, C. Lefay, D. Gigmes, Y. Guillaneuf and J. Nicolas, *Biomacromolecules*, 2019, **20**, 305–317.
- 19 J.-B. Lena, A. W. Jackson, L. R. Chennamaneni, C. T. Wong, Y. Andriani, P. Thaniyot and A. M. V. Herk, *Macromolecules*, 2020, **53**, 3994–4011.
- 20 A. W. Jackson, L. R. Chennamaneni and P. Thaniyot, *Eur. Polym. J.*, 2020, **122**, 109391.
- 21 M. C. D. Carter, A. Hejl, S. Woodfin, B. Einsla, M. Janco, J. DeFelippis, R. J. Cooper and R. C. Even, *ACS Macro Lett.*, 2021, **10**, 591–597.
- 22 T.-Y. Zeng, L. Xia, Z. Zhang, C.-Y. Hong and Y.-Z. You, *Polym. Chem.*, 2021, **12**, 165.
- 23 S. Komatsu, T. Sato and A. Kikuchi, *Polym. J.*, 2021, **53**, 731–739.
- 24 W. Bailey, Z. Ni and S.-R. Wu, *Macromolecules*, 1982, **15**, 711–714.
- 25 F. Xie, X. Deng, D. Kratzer, K. C. K. Cheng, C. Friedmann, S. Qi, L. Solorio and J. Lahann, *Angew. Chem., Int. Ed.*, 2017, **56**, 203–207.
- 26 C. Zhu and J. Nicolas, *Polym. Chem.*, 2021, **12**, 594–607.
- 27 W. Bailey, S.-R. Wu and Z. Ni, *Makromol. Chem.*, 1982, **183**, 1913–1920.
- 28 M. R. Hill, E. Guégain, J. Tran, C. A. Figg, A. C. Turner, J. Nicolas and B. S. Sumerlin, *ACS Macro Lett.*, 2017, **6**, 1071–1077.
- 29 J. Tran, E. Guégain, N. Ibrahim, S. Harrisson and J. Nicolas, *Polym. Chem.*, 2016, **7**, 4427–4435.
- 30 E. Guegain, C. Zhu, E. Giovanardi and J. Nicolas, *Macromolecules*, 2019, **52**, 3612–3624.
- 31 A. Kazama and Y. Kohsaka, *Polym. Chem.*, 2019, **10**, 2764–2768.
- 32 R. Friary, *J. Heterocycl. Chem.*, 1978, **15**, 63–64.
- 33 O. Igglessi-Markopoulou, G. Athanasellis, A. Detsi, K. Prousis and J. Markopoulos, *Synthesis*, 2003, 2015–2022.
- 34 L. Zheng, D. Zheng, Y. Wang, C. Yu, K. Zhang and H. Jiang, *Org. Biomol. Chem.*, 2019, **17**, 9573–9577.
- 35 P. J. Ma, F. Tang, Y. Yao and C. D. Lu, *Org. Lett.*, 2019, **21**, 4671–4675.
- 36 M. Haim-Zada, A. Basu, T. Hagigit, R. Schlinger, M. Grishko, A. Kraminsky, E. Hanuka and A. J. Domb, *Biomacromolecules*, 2016, **17**, 2253–2259.
- 37 A. L. J. Beckwith, S. Brumby and C. L. L. Chai, *J. Chem. Soc., Perkin Trans. 2*, 1992, **2**, 2117–2121.
- 38 A. Rudin and P. Choi, in *The Elements of Polymer Science & Engineering*, ed. A. Rudin and P. Choi, Academic Press, 3rd edn, 2013, ch. 9, pp. 402–406.
- 39 S. Agarwal, *Polym. J.*, 2007, **39**, 163–174.
- 40 J. Undin, T. Illanes, A. Finne-Wistrand and A.-C. Albertsson, *Polym. Chem.*, 2012, **3**, 1260–1266.
- 41 H. Wickel, S. Agarwal and A. Greiner, *Macromolecules*, 2003, **36**, 2397–2403.
- 42 J. Huang, R. Gil and K. Matyjaszewski, *Polymer*, 2005, **46**, 11698–11706.
- 43 H. Wickel and S. Agarwal, *Macromolecules*, 2003, **36**, 6152–6159.
- 44 *Polymer Handbook Fourth Edition*, ed. J. Brandrup, E. H. Immergut and E. A. Grulke, Interscience Publisher, New York, 1999.
- 45 Y. Inokuma, T. Yoneda, Y. Ide and S. Yoshioka, *Chem. Commun.*, 2020, **56**, 9079–9093.
- 46 D. Gigmes, P. H. M. Van Steenberge, D. Siri, D. R. D'hooge, Y. Guillaneuf and C. Lefay, *Macromol. Rapid Commun.*, 2018, 1800193.
- 47 T. Rohwerder and R. H. Müller, *Microb. Cell Fact.*, 2010, **9**, 13.
- 48 M. Pirmoradi and J. R. Kastner, *ACS Sustainable Chem. Eng.*, 2016, **5**, 1517–1527.

